<table>
<thead>
<tr>
<th>Title</th>
<th>Interfacial Structure and Reaction Mechanism of SiC/V Joints (Materials, Metallurgy &amp; Weldability)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Author(s)</td>
<td>Fukai, T.; Naka, M.; Schuster, J. C.</td>
</tr>
<tr>
<td>Citation</td>
<td>Transactions of JWRI. 1997, 26(1), p. 93-98</td>
</tr>
<tr>
<td>Version Type</td>
<td>VoR</td>
</tr>
<tr>
<td>URL</td>
<td><a href="https://doi.org/10.18910/12134">https://doi.org/10.18910/12134</a></td>
</tr>
<tr>
<td>rights</td>
<td></td>
</tr>
<tr>
<td>Note</td>
<td></td>
</tr>
</tbody>
</table>
Interfacial Structure and Reaction Mechanism of SiC/V Joints†

T. Fukai*, M. Naka** and J. C. Schuster***

Abstract

SiC was bonded to SiC using V foil at temperatures ranging from 1473K to 1673K for 1.8 to 64.8ks and 30MPa in vacuum. Interfacial reactions and microstructures were investigated using or electron probe microanalyser and X-ray diffractometer. SiC begins to join with V at temperatures above 1473K. At 1573K for 1.8 and 7.2ks, granular V2C phase formed at the V side of the reaction zone, while the layer V3Si phase is formed at the interface between V2C and SiC. The same reaction structure can be observed at 1473 and 1673K for 1.8ks. Hexagonal V5Si3C_x phase was formed at the interface between V3Si and SiC at 1573K for 14.4 and 21.6ks, and the interface structure of the joint became SiC/V5Si3C_x/V3Si/V2C+V IV. This complete diffusion path is correlated with the corresponding Si-V-C phase diagram. At the longer bonding time of 64.8ks, V was completely consumed, and the joint showed the layer structure of SiC/V3Si/V2C/V3Si/V5Si3C_x/SiC.

KEY WORDS: (Reaction Mechanism) (Interfacial Structure) (Ceramic) (Metal) (Diffusion) (SiC) (V) (Vanadium Carbide) (Vanadium Silicide)

1. Introduction

Silicon carbide is one of the candidates for high temperature structure components used up to 1273K with its remarkable features such as light weight, high strength and good resistance to wear. Some structural applications of ceramics require a reliable joining method with a metallic component. Compared to other joining methods such as brazing, solid state bonding provides high temperature strength in the interfacial region. However, for effective control of the interface structure, study of the phase reaction between the ceramic and the metal is necessary. While the interfacial structure, phase reaction, and the formation of titanium carbide and silicide at the SiC/Ti interface have been investigated in detail(1-5), few reports exist on the SiC/V system and V is a high melting metal like titanium and also reactive with SiC. According to Schuster(7), V5Si3C_x, V5Si3, V3Si, and V2C are formed in the reaction zone at temperature 1573K, but this report did not characterize the microstructures of the observed phases of the system. The aim of the present work is to examine the interfacial structure, phase reactions, and diffusion path between SiC and V.

2. Experimental Procedure

The materials used were cylindrical SiC rods of 6 mm diameter and 4 mm thickness, containing a few percent of alumina as sintering aid, and V foils of 25 µm thickness of 99.9 mass % purity. SiC /V/SiC couples were bonded under vacuum below 1.33mPa in a high frequency furnace equipped with a graphite tube. The applied bonding pressure was 30MPa. The bonding temperature ranged from 1473K to 1673K. The bonding time extended between 1.8 and 64.8ks. The phases occurring in the reaction zone were characterized by electron probe microanalysis (EPMA), and identified by x-ray diffraction with Cu Kα diffraction.

† Received on May 19, 1997
* Graduate Student
** Professor
*** Associate Professor, University of Vienna

Transactions of JWRI is published by Joining and Welding Research Institute of Osaka University, Ibaraki, Osaka 567, Japan.
3. Results and Discussion
3.1 Interfacial structure

At a constant bonding temperature of 1573K, the change in microstructure and reaction phases at the interface between SiC and V were examined by changing bonding time. The microstructure and distribution of elements for a joint bonded at 1573K for 1.8ks are shown in Figs.1 and 2 respectively. V reacts with SiC, and forms two kinds of phase. The layer phase adjacent to SiC and a granular phase adjacent to V were identified as $V_3Si$ and $V_2C$, respectively as shown in Fig.2. The microphotograph of SiC/V/SiC joint bonded at 1573K for 7.2ks is shown in Fig. 3. Vanadium still remains at the central part of the joint. The reaction phases at the interface grew with increasing bonding time from 1.8ks to 7.2ks. One is $V_3Si$ and other is $V_2C$. The layer of $V_3Si$ phase which has a thickness of 3μm is adjacent to SiC. The granular compound of $V_2C$ which has thickness of 13μm is adjacent to V. These microstructures can be clearly seen in Fig.4, which shows the distribution of the C, Si and V elements across the interface from A to B in Fig.3. This analytical result agrees with the X-ray diffraction results shown in Fig.5. The same reaction phase can be formed in the SiC/V/SiC bonded at the conditions of 1473K for 1.8ks and 7.2ks, and 1673K for 1.8ks.

The microstructure and distribution of elements after bonding at 1573K for 14.4ks are shown in Figs.6 and 7. The reaction zone as shown in Fig.6 consists of two layers together with a granular compound. EPMA analysis in Fig.7 shows that the first layer of thickness 3μm is adjacent to SiC, containing on average 30.0at% Si and 17.5at% C, giving the composition of $V_5Si_3C_x$, and that the second layer of a thickness 2.5μm has an average composition of 63.0at% V and 21.0at% Si, giving the composition of $V_3Si$. Granular $V_2C$ also appears in the V, and this $V_2C$ must be adjacent to $V_3Si$. X-ray analysis of the fracture surface of the joint (Fig.8) indicates the formation of hexagonal $V_2C$ with lattice parameters $a = 0.2904\text{nm}$ and $c = 0.4523\text{nm}$, of cubic $V_3Si$ with $a = 0.4734\text{nm}$, and hexagonal $V_5Si_3C_x$ with $a = 0.7161\text{nm}$, $c = 0.4762\text{nm}$. Thus, at bonding conditions of 1573K and 14.4ks, the reaction phases formed established the complete diffusion path between SiC and V as SiC/$V_5Si_3C_x$ / $V_3Si$ / $V_2C$ / $V_2C+V$ / V. This diffusion path is described by the tie line connecting SiC and V in the corresponding V-Si-C ternary phase diagram as shown in Fig.9.

**Fig.1** Microphotograph of SiC/V/SiC joint bonded at 1573 K for 1.8ks.

**Fig.2** Elemental distribution in SiC/V/SiC joint bonded at 1573 K for 1.8 ks.
**Fig. 3** Microphotograph of SiC/V/SiC joint bonded at 1573 K for 7.2 ks.

**Fig. 4** Elemental distribution in SiC/V/SiC joint bonded at 1573 K for 7.2 ks.

**Fig. 5** X-ray diffraction pattern of the interface of SiC/V/SiC joint bonded at 1573 K for 7.2 ks.

**Fig. 6** Microphotograph of SiC/V/SiC joint bonded at 1573 K for 14.4 ks.

**Fig. 7** Elemental distribution in SiC/V/SiC joint bonded at 1573 K for 14.4 ks.

**Fig. 8** X-ray diffraction pattern of the interface of SiC/V/SiC joint bonded at 1573 K for 14.4 ks.
At the longer bonding time of 64.8ks at the bonding temperature of 1573K, the microstructure of SiC / V / SiC joint is shown as in Fig. 10. V at the central part of the joint was completely consumed, and changed to V2C. The joint is composed of the layer structure as SiC / V5Si3C / V3Si / V2C / V3Si / V5Si3C / SiC, as shown in the distribution of elements from A to B in the joint in Fig. 11.

Fig. 9 Diffusion path between SiC and V on the V-Si-C ternary phase diagram at 1573K.

Fig. 10 Microphotograph of SiC/V/SiC joint bonded at 1573 K for 64.8 ks.

3.2 Reaction mechanism

Fig. 12 shows the change in schematic structure of the interface between SiC and V bonded at 1573K for the bonding time from 1.8 to 64.8ks. The reaction phases are V5Si3C and V3Si together with granular V2C. The reaction mechanism between SiC and V is discussed using the present results. The phases of V5Si3C and V3Si grow with increases in the bonding time. The V2C of the central part of V grows with the bonding time. The phases at the interface between SiC and V couples are formed by the following reactions:

\[
\begin{align*}
\text{SiC} + 5\text{V} &\rightarrow \text{V}_3\text{Si} + \text{V}_2\text{C} \\
4\text{SiC} + 14\text{V} &\rightarrow 5\text{V}_5\text{Si}_3\text{C}_x + 3\text{V}_3\text{Si} + 3\text{V}_2\text{C}
\end{align*}
\]

First, V reacts with SiC at the interface, and forms V2C and V3Si. Carbon diffuses faster than silicon in V, and V2C is formed adjacent to the V, and V3Si is formed adjacent to SiC which are represented by eq. (1) and Fig. 12a and b. The V still remains at the central part of the joint and carbon and silicon continue to diffuse from SiC to V. Then, granular V2C is grown and V5Si3C is formed adjacent to the SiC and these are represented by eq.(2), and Fig. 12c and d.

At longer bonding times, V at the central part of joint is completely consumed, and the central part of the the V changes to V2C or VC, which are represented by Fig. 12 e and f.
**Fig. 12** Change in microstructure of the interface between SiC and V bonded at 1573 K for the bonding time from 1.8 ks to 64.8 ks.
4. Conclusions

The solid state bonding of SiC to SiC using V foil was conducted at the bonding conditions of 1373 to 1673K, 1.8 to 64.8ks and 30MPa under a vacuum below 1.33mPa. Interfacial reactions at the interface between SiC and V were studied using elemental analyses and X-ray diffraction analyses.

Reaction at the SiC/V interface was observed at the bonding temperature of 1573K for the bonding time at 1.8ks, two kinds of reaction phases are formed in the interface between SiC and V. The layer V₃Si phase is formed adjacent to SiC and granular V₂C is formed adjacent to the V. After bonding for 14.4ks at the bonding temperature of 1573K, the reaction zone adjacent to SiC consist of two reaction layers together with granular V₂C. The first layer is V₅Si₃Cₓ and the second layer is V₃Si, and granular V₂C is grown in the V.

Thus, the complete diffusion path was established as SiC / V₅Si₃Cₓ / V₃Si / V₂C / V₂C+V / V. At the bonding time of 64.8ks and a temperature of 1573K, the central part of the V was completely, consumed and changed to V₂C or VC.

References