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# A Study of the Photo-Catalytic Character of Plasma Sprayed TiO<sub>2</sub> Coatings †

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## Abstract

*Two types of rutile and anatase powders are used for the deposition of TiO<sub>2</sub> coatings. The effects of plasma spraying conditions on the structure of TiO<sub>2</sub> coatings are investigated in order to clarify controlling factors of the phase formation and to aim at the development of effective photo-catalyst TiO<sub>2</sub>. It is found that the amount of anatase TiO<sub>2</sub> in the coating is influenced by spray parameters. The decrease of the heat input to spray droplet and an increase in the cooling speed during droplet deposition will increase the amount of the anatase TiO<sub>2</sub> in the coating. As a photo-catalyst, the coating deposited under limited plasma power using anatase powder is effective for the decomposition of acetaldehyde gas.*

**KEYWORDS :** (Plasma Spray) (Photo-Catalyst) (TiO<sub>2</sub> Coating) (Acetaldehyde) (Decomposition)

## 1. Introduction

Owing to its unique semi-conductor characteristics, TiO<sub>2</sub> oxide is increasingly of interest as a promising photo functional material, which can be applied to solar cells, water decomposition, photo assisted decomposition of harmful organic substances and removal of harmful foul gases by decomposition, especially with increasing attention concerning environment protection[1]. TiO<sub>2</sub> presents three different types of crystal structures. These are rutile type, anatase type and brookite type. Generally, the rutile type of TiO<sub>2</sub> is thermally stable. The anatase TiO<sub>2</sub> is metal-stable, which will be transformed to rutile structure after an annealing treatment[2]. As a photo-catalyst, anatase TiO<sub>2</sub> is more effective[3]. Therefore, TiO<sub>2</sub> photo-catalyst is generally prepared by low temperature processes such as dipping and spraying using the microfine powders of anatase structure, and sol-gel method as well.

Plasma spraying has been used effectively to produce ceramic coatings. Generally, the meta-stable phase can be deposited because of the inequilibrium characteristics of rapid cooling after a droplet impacts on a surface[4]. Therefore, despite the high temperature characteristics of ther-

mal plasma as a heat source, it was found that the anatase TiO<sub>2</sub> can be formed in plasma-sprayed TiO<sub>2</sub> coatings although the amount of the anatase phase in the coating is much limited[5].

The present study aims to clarify controlling factors of the phase transformation in the deposition of TiO<sub>2</sub> coating and the development of anatase-TiO<sub>2</sub> dominant coatings. TiO<sub>2</sub> coatings are sprayed using two TiO<sub>2</sub> powders with rutile and anatase structures, and the effects of plasma spraying conditions on the TiO<sub>2</sub> coating structure are investigated.

## 2. Materials and experimental procedures

### 2.1 Materials

Two types of TiO<sub>2</sub> powders with rutile and anatase crystal structures were used in the experiments for plasma spraying. The commercial rutile TiO<sub>2</sub> powder of a mean grain size of 27μm consisted of a fraction of magneli phases besides the main rutile phase. Therefore, the powder was annealed at 1270K for 10.8ks in the ambient atmosphere before spraying. After the annealing treatment, the structure of the powder became the single rutile phase. The anatase

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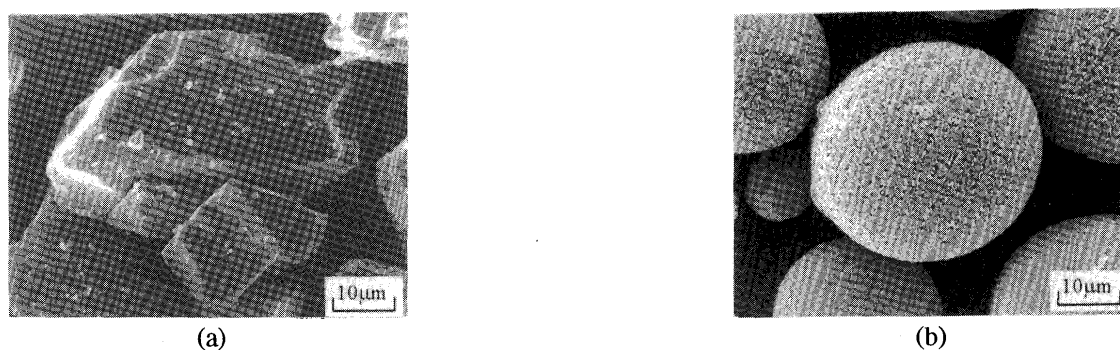


Fig. 1 The morphologies of powders of rutile TiO<sub>2</sub> (a) and anatase TiO<sub>2</sub> (b).

TiO<sub>2</sub> powder, which has a mean grain size of 34µm, was an agglomerated powder with the fine TiO<sub>2</sub> powders of 0.2µm in original grain size. The morphologies of two powders were illustrated in Fig. 1. Stainless steel was used as a substrate.

## 2.2 Plasma spraying

The TiO<sub>2</sub> coatings were deposited by a Metco 9MB plasma spray system. Argon was used as a primary plasma gas and hydrogen was used as the secondary gas. Table 1 illustrates the primary spray conditions. During the spraying, the flow of hydrogen gas and arc current were changed to investigate the effect of spray conditions on the structure of deposited TiO<sub>2</sub> coatings.

## 2.3 Characterization of the coating

The structure of the coating was characterized by X-ray diffraction analysis. The relative content of anatase TiO<sub>2</sub> in the coating was estimated from the ratio of diffraction intensity of the anatase (101) peak with the total intensity of the anatase (101) and rutile (101) peaks. Regarding the orientation of the rutile TiO<sub>2</sub> in the coating, the above estimation utilized the intensity of the (101) peak of rutile TiO<sub>2</sub> instead of the main peak (110).

The microstructure of the TiO<sub>2</sub> coating was examined by optical microscopy and electron scanning microscopy.

Table 1 Plasma spray conditions

Primary plasma gas (Ar)	
Pressure	0.7 MPa
Flow rate	$7.88 \times 10^{-4} \text{ m}^3/\text{s}$
Secondary plasma gas (H <sub>2</sub> )	
Pressure	0.4 MPa
Flow rate	$0 - 1.97 \times 10^{-4} \text{ m}^3/\text{s}$
Spray distance	100 mm
Arc current	400 - 900 A
Arc voltage	37 - 72 V

## 2.4 Characterization of the acetaldehyde gas decomposition by TiO<sub>2</sub> coating

The TiO<sub>2</sub> coating as a semi-conductor photo-catalyst was characterized through the decomposition test of acetaldehyde. Fig. 2 shows schematically the experimental set-up. The volume of the container is about 2l. After the container was filled with a certain concentration of the acetaldehyde, the ultraviolet lamp was switched on to generate the light of 360nm wave length towards the sample surface at an intensity of 1mW/cm<sup>2</sup>. Afterward, the decay in the concentration of the acetaldehyde was measured with a gas detector after a certain time interval.

The experiment results showed that the decay of the concentration of the acetaldehyde follows the exponential rule, e.g. :

$$N = N_0 \exp(-t / t_0)$$

Where,  $N$  and  $N_0$  is the concentration at time  $t$  and initial concentration, respectively;  $t_0$  is a constant related to the decay speed. The lower the value  $t_0$ , the more rapid the decomposition of acetaldehyde. Therefore, the  $t_0$  was used as the characteristic decay time to evaluate the effectiveness of the coating for decomposition.

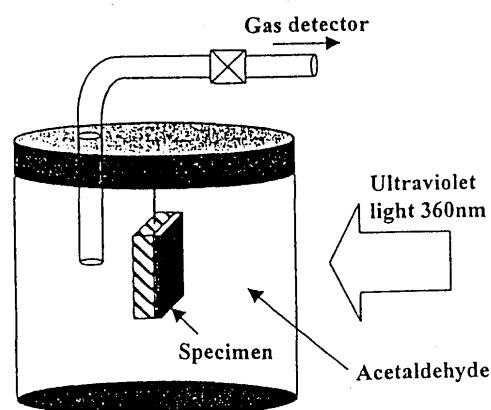


Fig. 2 Schematic diagram of photo-catalytic experimental set-up.

### 3. Results and discussion

#### 3.1 The structure of the TiO<sub>2</sub> coating deposited by rutile powder

Fig. 3 shows the typical cross-sectional microstructure and X-ray diffraction pattern of a TiO<sub>2</sub> coating sprayed by rutile powder under an arc current of 600A and H<sub>2</sub> flow of 7.88×10<sup>-4</sup> m<sup>3</sup>/s.

From the X-ray diffraction pattern it is clearly recognized that the coating consists of rutile TiO<sub>2</sub> and anatase TiO<sub>2</sub>. Regarding the rutile structure of the starting powder and the thermal stability of rutile TiO<sub>2</sub> at high temperature, it can be considered that the formation of the anatase TiO<sub>2</sub> is related to the rapid cooling of spray droplets inherent to the thermal spray process. In order to clarify the forming process of anatase TiO<sub>2</sub> in the coating, the effects of spray parameters on the anatase TiO<sub>2</sub> in the deposited coating were investigated.

#### 3.2 Effect of H<sub>2</sub> flow rate on the formation of anatase-TiO<sub>2</sub> in the deposited coating

Fig. 4 shows the X-ray diffraction patterns of the TiO<sub>2</sub> coatings deposited under different H<sub>2</sub> flow rates. It is evident that with an increase in H<sub>2</sub> flow rate the intensity of the anatase TiO<sub>2</sub> decreases. Fig. 5 illustrates the anatase ratio in TiO<sub>2</sub> coatings estimated from the relative intensity

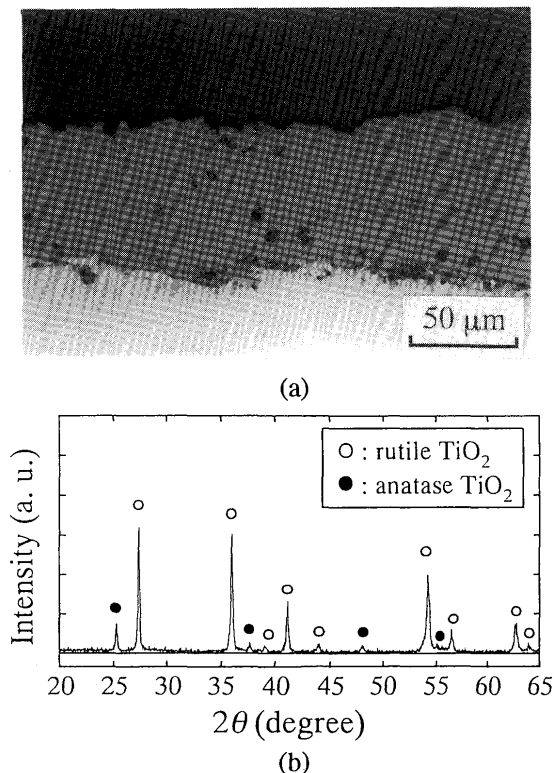


Fig. 3 Microstructure (a) and X-ray diffraction pattern (b) of plasma-sprayed TiO<sub>2</sub> coating.

of the X-ray diffraction peaks. The result clearly shows that the increase in H<sub>2</sub> gas flow has the effect of suppressing the formation of the anatase phase in the TiO<sub>2</sub> coatings. This may be because an increase of the H<sub>2</sub> gas flow into the plasma improves the heating of plasma jet to the spray powder, and subsequently raises the temperature of droplet solidification. Therefore, the result suggests that a decrease in the droplet temperature during plasma spraying enhances the formation of the anatase TiO<sub>2</sub>.

#### 3.3 Effect of plasma arc current on the formation of anatase-TiO<sub>2</sub> in the TiO<sub>2</sub> coating

The effect of H<sub>2</sub> addition into the plasma gas on the X-ray diffraction results of the sprayed coatings implies that the high heat input into spray powder will suppress the formation of the anatase TiO<sub>2</sub> in the coating. Regarding the

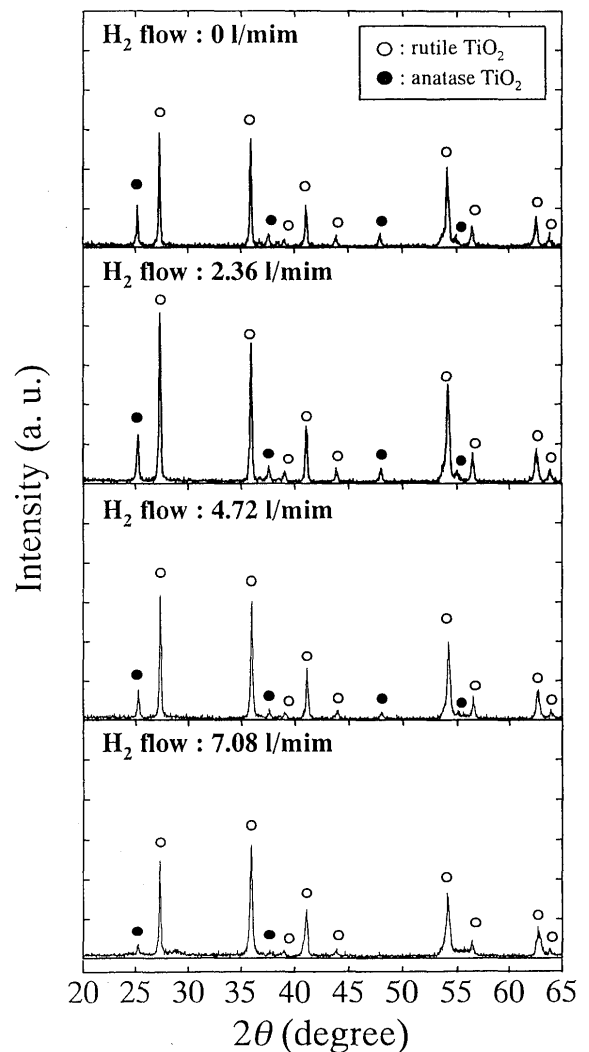


Fig. 4 X-ray diffraction patterns of TiO<sub>2</sub> coatings sprayed at different H<sub>2</sub> flow rates (Arc current : 600A).

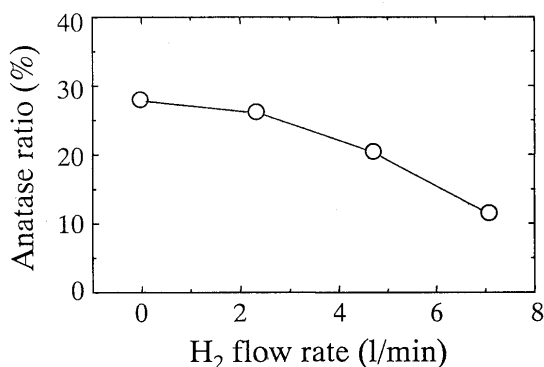


Fig. 5 Effect of H<sub>2</sub> flow rates on the anatase ratio in the TiO<sub>2</sub> coating (Arc current : 600A).

heating of the powders by the plasma jet, the plasma power is one of the important parameters. Fig. 6 illustrates the effect of the plasma arc current on the relative anatase ratio in the TiO<sub>2</sub> coating. During the deposition the argon plasma was used without H<sub>2</sub> addition. It is seen that with an increase in plasma arc current (e.g. plasma power) the content of the anatase phase in the coating decreases.

### 3.4 Effect of substrate cooling on the formation of anatase-TiO<sub>2</sub> in the TiO<sub>2</sub> coating

The effect of the power and H<sub>2</sub> addition into plasma gas revealed that the formation of the anatase phase is closely related to the heating of the droplet and subsequent cooling processes. A lower heat input to the spray powder and rapid cooling of the droplet will enhance the formation of the anatase phase in the TiO<sub>2</sub> coating. To confirm the effect of the cooling of the droplet on the formation of anatase TiO<sub>2</sub>, a TiO<sub>2</sub> coating was deposited onto a copper substrate of 3mm in thickness and the effect of the cooling from the back of the substrate by running water was examined.

Fig. 7 illustrates the X-ray diffraction patterns of TiO<sub>2</sub> coatings deposited on the copper substrate both with, and without water cooling compared with that of the coating deposited on the stainless steel. It can be seen that the coatings sprayed on the copper substrate contain a little more anatase phase and the cooling of the copper substrate tends to increase the anatase ratio in the TiO<sub>2</sub> coating.

Therefore, it is clear that an increase in the cooling speed of the droplet after it impacts on a substrate lead to an increase of the anatase phase in the coating. On the other hand, it is also evident that the maximum relative content of the anatase TiO<sub>2</sub> phase in a plasma-sprayed TiO<sub>2</sub> coating is around 30 to 35% by using rutile powder.

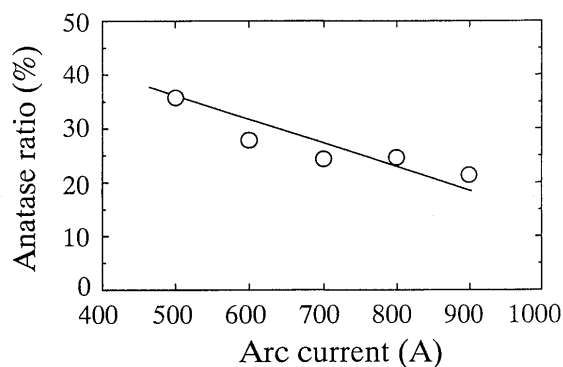


Fig. 6 Effect of arc current on the anatase ratio in the TiO<sub>2</sub> coating.

### 3.5 Effect of powder types on the formation of anatase-TiO<sub>2</sub> in the deposited coating

Fig. 8 illustrates the X-ray diffraction patterns of TiO<sub>2</sub> coatings sprayed with Ar plasma at different arc currents using agglomerated anatase powder. The estimation of the anatase ratios of the coatings deposited at the arc currents of 400, 500 and 600A yielded about 32%, 18% and 16%, respectively. It is clear that the arc current exerts the same effect on the anatase ratio as in the coatings deposited with rutile powder. The maximum anatase ratio yielded a similar value to that using rutile powder. An annealing treatment of the anatase powder above 1223K revealed that after annealing the anatase phase in the original powder is changed to rutile phase. Therefore, it can be considered that the anatase phase in the coating deposited using the anatase powder is primarily formed through the same process as with rutile powder when the powder is completely melted during the spraying, because the heating and subsequent melting of the powder leads to the formation of liquid droplets of the same structure despite original structure of the powders.

### 3.6 The decomposition characteristics of acetaldehyde by plasma-sprayed TiO<sub>2</sub> coating

Fig. 9 illustrates the decay characteristics of the acetaldehyde concentration by TiO<sub>2</sub> coating catalyst-assisted decomposition. The coatings were deposited under different arc currents using anatase powder. It was found that using the coating deposited under low arc current (which contains a little more anatase TiO<sub>2</sub>) the decomposition of the acetaldehyde is much more rapid compared with the coating deposited under a high arc current. The relationship between the anatase ratio, arc current and decay time of

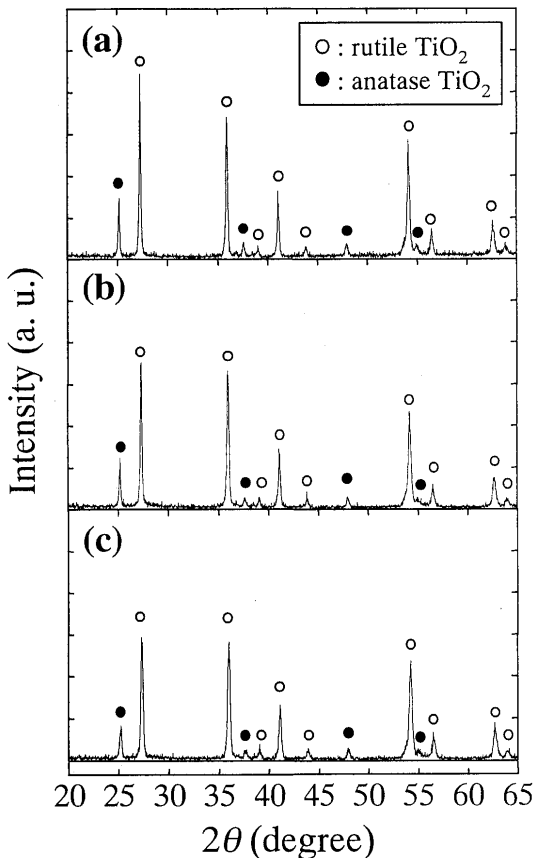


Fig. 7 X-ray diffraction patterns of TiO<sub>2</sub> coatings.  
 (a) Cu substrate with water cooling,  
 (b) Cu substrate without water cooling,  
 (c) Stainless steel substrate.

acetaldehyde gas, shown in Fig. 10, clearly revealed that the decay time of the acetaldehyde gas by the coatings deposited under low arc currents of 400A and 500A is less by one order than that using the coating deposited under the current of 600A.

It was also noticed that although the anatase ratios in the above mentioned three coatings have no significant difference, the decay time is much different. The examination of the surface morphology of the coatings as shown in Fig. 11 clearly shows that under low arc current the surface of the coating has a similar morphology with that of the powder, which evidently shows that the powder is only partially melted. However, the surface of the coating deposited under a high arc current (plasma power) presents a more typical thermally sprayed coating morphology which suggests the significant melting of the spray powder. From these results it can be suggested that, under partially melting, although the anatase phase is still formed during droplet cooling, the subsequent anatase phase will be distributed preferably near the particle surface. As the result, the anatase

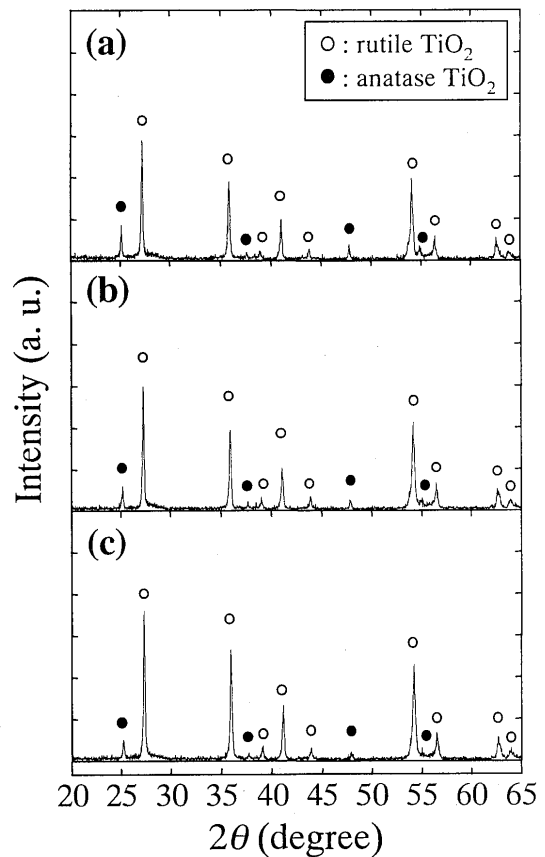


Fig. 8 X-ray diffraction patterns of TiO<sub>2</sub> coatings sprayed under different arc currents using the anatase powder.  
 (a) 400 A, (b) 500 A, (c) 600 A.

phase can directly act as the catalyst to promote the decomposition of acetaldehyde.

#### 4. Conclusions

TiO<sub>2</sub> coatings deposited by plasma spraying using two powders of rutile and anatase structures, the effects of plasma arc current and the addition of H<sub>2</sub> gas into the plasma on the formation of the anatase TiO<sub>2</sub> in the coating are investigated.

The experimental results clearly show that the plasma spraying process prefers to deposit a TiO<sub>2</sub> coating with rutile dominant structure despite the structure of the starting powder. The amount of the anatase TiO<sub>2</sub> in the coating is influenced by spray conditions. An increase in plasma arc current and the addition of H<sub>2</sub> gas into the plasma tend to suppress the formation of the anatase-TiO<sub>2</sub> through the increase of heat input to the spray droplets. Moreover, an increase in the cooling speed during droplet deposition will increase the amount of the anatase-TiO<sub>2</sub> in the coating. The photocatalytic characteristics of the TiO<sub>2</sub> depend on the amount

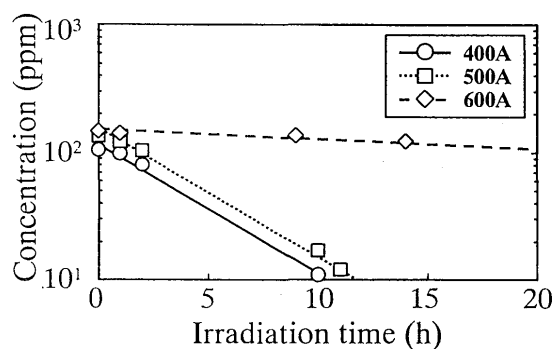


Fig. 9 The decay characteristic of the acetaldehyde concentration by TiO<sub>2</sub> photo-catalyst coatings sprayed under different arc currents.

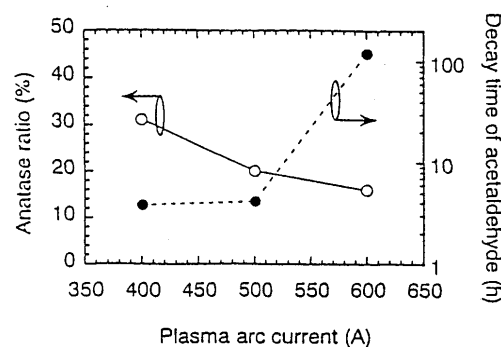


Fig. 10 Effect of plasma arc current on the anatase ratio in TiO<sub>2</sub> coatings and decay time of acetaldehyde irradiated by ultra violet light.

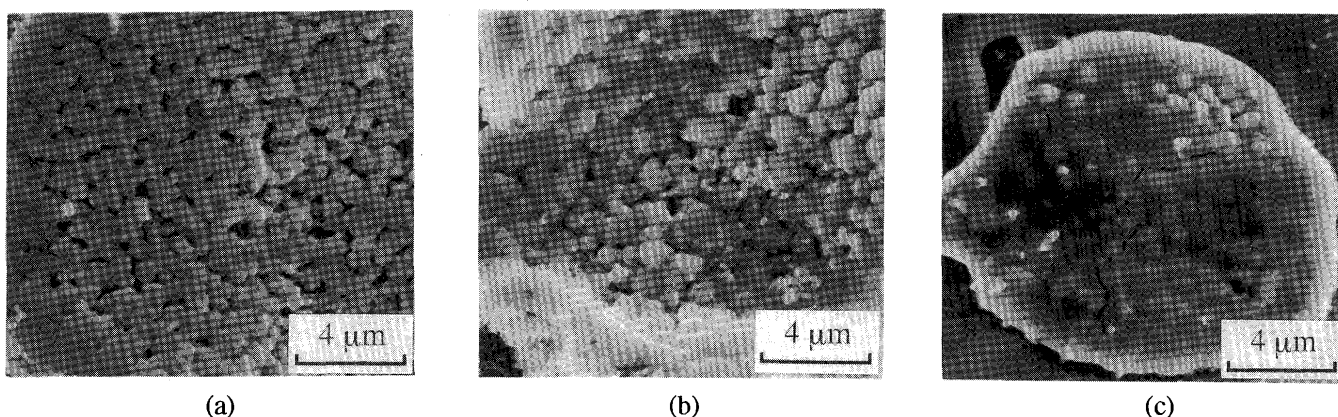


Fig. 11 Surface morphologies of TiO<sub>2</sub> coatings sprayed at different arc current. (a) 400 A, (b) 500 A, (c) 600 A.

of anatase-TiO<sub>2</sub> and also the surface morphology of the coating. It is found that the decay time of acetaldehyde concentration using the coating deposited with partially melted agglomerated anatase powder is one order less than that of the coating deposited with well-melted droplets.

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