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<td>Kikuchi, Yasushi; Matsuda, Fukuhisa; Okabe, Tomoyosi</td>
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Weldability of High Nitrogen Austenitic Stainless Steel Report 1
~ Nitrogen Absorption of 316L Type Stainless Steel Welds in Extreme Environments ~

Yasushi KIKUCHI*, Fukuhiya MATSUDA**, and Tomoyosi OKABE***

Abstract
Nitrogen absorption of 316L weld metal by high pressure GMA(constant arc length condition) process was investigated. The welding atmospheres were prepared by pressurized N2 and N2-CO2 gas mixture. Also, the behavior of nitrogen in remelted weld metal containing high nitrogen and change of microstructure of welded were observed. The nitrogen content increased with increasing partial pressure of N2. 0.60wt.% of nitrogen was absorbed in weld metal at 6.0MPa N2 atmosphere. But this was very lower content than equilibrium solubility of 316L stainless steels at close to melting point. Nitrogen content in weld metal under pressurized N2-CO2 mixture atmosphere was different from pure N2 atmosphere. The gas composition consists of PN2: 0.35MPa:Po2:0.05MPatotal pressure : 0.4MPa, it showed that maximum nitrogen content of 0.56wt.%.

KEY WORDS:(Nitrogen) (Absorption) (Stainless Steel) (Gas-Metal Reaction)

1. Introduction
The structural materials are demanded the properties of nonmagnetic, high resistance for corrosion, wear and high strength according to the development of the electronics, the precision and the cryogenic industry. Recently a new material of cold-rolled austenitics stainless steel which contained over 1% of nitrogen is candidate[1].

The characteristics for remelting and solidification of the new material aimed welding, however, dose not studied sufficiently yet.

The purpose of dioxide gas mixture atmospheric GMA method was discussed. The results of nitrogen content (n) were discussed referring to thermodynamical equilibrium date also.

2. Experimental Procedure
The chemical compositions of steel used was shown in Table 1
The size of specimen is 12mm in thickness, 100mm in width and 200mm in length. The gas metal arc (GMA) process was used in high pressure atmosphere. The consumable electrode wire with 1.6mm diameter was used also.

Table 1 Chemical composition of base metal and electrode wire (mass%)

<table>
<thead>
<tr>
<th>Material</th>
<th>C</th>
<th>Si</th>
<th>Mn</th>
<th>P</th>
<th>S</th>
<th>Cr</th>
<th>Ni</th>
<th>Mo</th>
<th>N</th>
<th>O</th>
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<tr>
<td>SUS316L</td>
<td>0.017</td>
<td>0.65</td>
<td>1.21</td>
<td>0.03</td>
<td>0.02</td>
<td>17.01</td>
<td>12.26</td>
<td>2.07</td>
<td>0.018</td>
<td>0.004</td>
</tr>
<tr>
<td>SUS316L-5mm</td>
<td>0.018</td>
<td>0.50</td>
<td>1.82</td>
<td>0.019</td>
<td>0.021</td>
<td>19.24</td>
<td>12.71</td>
<td>2.42</td>
<td>0.05</td>
<td>0.012</td>
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Figure 1 shows the stainless steel chamber with 1.8m3 volume capacity for the high pressure welding.

Fig. 1 Arrangement of equipment

*** Graduate student, Osaka University (KAWASAKI Steel Co. Ltd)
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The atmospheric pressure in the chamber can be changed from about 15Pa to 6.5MPa. The head of an automatic MIG, TIG and Sub-margined arc welding machine are able to mount in this chamber.

Bead on plate welding performed by means of direct current power source which has an open circuit voltage of 80V (1000A). The chamber was evacuated to a pressure until under 100Pa and then filled with the desired gas pressure and gas compositions.

Surface ground and cleaned test specimens were set in the chamber.

The distance between the contact tip and the specimen surface was 35mm. The welding current was about 200A. Travel speed of specimen was 3.3 mm/s. The visual arc length was approximately 10mm.

The nitrogen and oxygen content of deposited metals were analyzed by leco equipment.

Surface appearance was checked and microstructure of weld metal was observed by optical microscope or SEM. The lattice parameter measurement and Vicker's hardness test were applied to weld metals.

3. Experimental Results and Discussion

3.1 Nitrogen absorption under high pressure pure nitrogen

The surface appearances of weld metal obtained in various nitrogen pressure from 0.1MPa to 6MPa using GMA welding are shown in Fig. 2. There is no defect on the bead surface as a blowhole or crack.

Its geometry seems smooth and without peculiar shape. Arc voltage is varied by pressure of atmosphere from about 68V at 0.1MPa to about 90V at 6.0MPa. Constant arc length condition by means of drooping characteristic power source was used in this work.

So, the effect of pressure appears in the arc voltage change directly. Because, the density of nitrogen gas increases and cooling effect of the arc column increase also. The arc is constricted, consequently, arc voltage raises remarkably. The nitrogen content (N) of weld metal as a function of ambient nitrogen pressure are shown in Fig. 3. N increases with increasing of nitrogen pressure and N shows approximately 0.6 mass% at 6MPa. N content reported by T-Kuwana et.al 2) is plotted in the same figure (base metal 304 stainless steel). Comparing with T-Kuwana's result, the present result shows higher nitrogen content. This reason can predict as follow. Constant arc voltage control condition had been applied by T-Kuwana's work. In his code, it was observed that arc length decreased with increase of pressure. It means that reaction time
molten metal and nitrogen has become much shorter. But, in case of constant arc length condition like this work, nitrogen absorption was not hindered by the change of arc length resulting high pressure arc atmosphere.

Thermodynamical discussion on the nitrogen absorption of the stainless steel weld metal in high pressure GMA welding is considered as follows.

Usually, an equilibrium state, the relationship between the nitrogen absorption in liquid iron or liquid stainless steel is described by the following equation.

\[ \frac{1}{2} N_2 = N \text{(in liquid iron)} \]

\[ K = \frac{[\%N (\text{iron})]}{\sqrt{P_{N_2}}} \cdots (1) \]

\[ \log K = -\frac{188}{T} - 1.248 \]

\[ \frac{1}{2} N_2 = N \text{(in liquid stainless steel)} \]

\[ K = \frac{[\%N (\text{SS})] f_{N}^{(S.S)}}{\sqrt{P_{N_2}}} \cdots (2) \]

Where \( K \) : the equilibrium constant

\( \%N_{(SS)} \) : the solubility of nitrogen in liquid stainless steel%

\( f_{N}^{(SS)} \) : the activity coefficient of nitrogen in liquid stainless steel

\( (\approx f_{N} \cdot f_{N(Cr)} \cdot f_{N(Ni)} \cdot f_{N(Si)} \cdot f_{N(Mn)} \cdots) \)

\( f_{N} = 1 \)

\( f_{N(M)} \) : the effect of alloying element M on the activity coefficient of nitrogen

\( P_{N_2} \) : the nitrogen partial pressure (at)

\( a_{N} \) : activity of nitrogen

from equation (1) and (2), equation (3) is given.

\[ \frac{\%N_{(SS)}}{f_{N}^{(SS)}} = \cdots (3) \]

\( f_{N}^{(SS)} \) in equation (3) is expressed by Wagner’s equation as follows.

\[ f_{N}^{(SS)} = e_{N}^{(Cr)} \cdot \frac{[\%Cr]}{e_{N}^{(NO)} \cdot \%Ni} \]

\[ + e_{N}^{(Mo)} \cdot \%Mo + e_{N}^{(Mn)} \cdot \%Mn \cdots (4) \]

where \( e_{N}^{(M)} \) : \( \bar{\partial} \log f_{N}^{(M)} / \partial \%M \) is an interaction parameter.

It can be known that Cr, Ni, Mo and Mn content of weld metal by chemical analysis. \( f_{N}^{(M)} \) (activity coefficient) is determined from the figure in an equilibrium study by Humbert 4) and works of Phelke et al. 5) so, \( f_{N}^{(S.S)}_{1873K} = 0.181 \)

An equilibrium solubility of 316L stainless steel at 1873K is given in equation (5)

\[ \%N_{(SS)}_{1873K} = 5.52 \cdot \sqrt{P_{N_2}} \cdots (5) \]

The equation (5) means that the equilibrium solubility of nitrogen in the liquid stainless weld metal is 5.52 time that in pure iron weld metal.

But, nitrogen content of 316L stainless steel weld metal in the result, is much lower content. This experimental result means that in GMA welding, reaction time between molten weld metal and nitrogen gas is much shorter than that of equilibrium solubility measurements.

On the other hand, it has been reported that the solubility of nitrogen molten stainless steel decrease with increasing of temperature.

Using Nelson's equation 6), solubility of nitrogen as function of temperatures is calculated as follows. Relation between the activity coefficient of nitrogen and temperature is described by the following equation (6).

\[ \log f_{N}^{(SS)} = \frac{1873}{T} \left( \sum \log f_{N}^{(M)}_{1873K} \right) \cdots (6) \]

From equation (5) and (6), equation (7) is given.

\[ \%N_{(SS)} = \frac{K_{SS1873}}{f_{N}^{(SS)}_{1873}} \cdots (7) \]

where \( \log K = -\frac{188}{T} - 1.248 \)

\( f_{N}^{(SS)}_{1873K} = 0.181 \)

Using equation (7), solubilities of nitrogen from 1873K to 2273K were calculated. Obtained nitrogen content as a function of the square root of nitrogen pressure are shown in Fig. 4.

Results of this work are shown by open circles in the same figure. Nitrogen content obtained by this experiment, can not attain the equilibrium solubility at higher pressure because of shortage of reaction time. At \( \sqrt{p_{N_2}} \) above 1MPA, liner relationship is observed between nitrogen content of weld metal and nitrogen pressure.

Apparently K calculated is 0.268. Relation between weld metal nitrogen contents and base metal
nitrogen contents are shown in Fig. 5. Nitrogen pressure of GMA welding were changed from 0.1MPa to 6.0 MPa. From Fig. 5, it can be estimated that minimum nitrogen pressure for GMA required by base metal nitrogen content.

Also, about 0.8 % nitrogen weld metal is given by 6MPa nitrogen pressure. In the weld metal, no micro void or crack were observed.

### 3.2 Nitrogen absorption under high pressure nitrogen -carbon dioxide gas mixture.

It was reported 7) that nitrogen absorption of weld metal is accelerated by presence of carbon dioxide gas in nitrogen shielding gas on GMA process. So, in order to higher nitrogen contained stainless steel weld metal, GMA welding was carried out in pressurized N2-CO2 gas mixture.

Figure 6 shows the relationship between nitrogen contents of weld metal and CO2 % in welding atmosphere. Welding was carried out from 0.4 MPa to 6 MPa using a tured gas.

An acceleration of the nitrogen absorption of weld metal by presence of CO2 shows at 0.4MPa N2-CO2 gas mixture.

But, at above 2MPa, addition of CO2 into arc atmosphere is of no effect. But, at above 2 MPa, addition of CO2 into arc atmosphere is of no effect. Weld metal surface was covered by oxide film and bead wide slightly increased by CO2 addition.
Fig. 7 Effect of nitrogen pressure in N₂-CO₂ mixture on the nitrogen contents of weld metal

Relation between nitrogen contents and nitrogen pressure of N₂-CO₂ mixture atmosphere is shown in Figure 7. Partial pressure of CO₂ is marked in the same figure. An accelerative effects of the nitrogen absorption by CO₂ presence are observed at less than about 1MPa nitrogen and less than 0.1MPa CO₂ (1st period). Ambient nitrogen pressure not be less than 2MPa, an accelerative effect has not been observed. (2nd period).

Welding was carried out on the water cooled Copper plate. By this method, droplets which reacted to arc atmosphere could be collected.

Relation between nitrogen contents droplets and nitrogen pressure is shown in Figure 8. Open circle means in N₂-CO₂ gas mixture and closed circle is pure N₂ respectively. In N₂-CO₂ gas mixture, the droplets absorbed a lot of nitrogen from atmosphere during the transfer process in the arc.

It was stimulated by CO₂ coexisting N₂. It is commonly produced in the arc by the dissociation of CO₂ into carbon monoxide and oxygen in the arc. Oxygen has strong effect on molten metal.

It is known that oxide film formed on the molten droplet surface retards nitrogen desorption. Also, the effect of NOx produced in arc atmosphere estimated. Surface activation effect of oxygen is important.

2nd period in Fig. 7, nitrogen contents of weld metal in high pressure N₂-CO₂ atmosphere are lower than that of pure N₂ atmosphere.

It is considered that nitrogen absorption into weld metal retards in this atmosphere rather than acceleration by CO₂.

Weld metal oxygen content was analyzed. It was approximately 0.06-0.08%. As stated above, oxygen in the molten metal disturbs strongly the adsorption and desorption rate of nitrogen. In this case, it is estimated that nitrogen absorption from weld pool retards by formed oxide film on the surface.

The effect of oxygen content on the nitrogen contents of weld metal was examined. Different oxygen content base metals were used.

Relation between nitrogen contents of weld metal and nitrogen pressure is shown in Fig. 9. It is clear that high oxygen base metal (0.07wt%) shows lower nitrogen contents. So, nitrogen absorption from weld pool surface is depressed by covered oxide film on weld pool.
3.3 Properties of high nitrogen weld metal

Figure 10 and 11 indicate the microstructures of weld metal.

Weld metals considered from an austenite single phase. Nitrides are observed in 0.68 and 0.84\% nitrogen contained weld metal. As shown in Fig. 11, flake like nitrides are estimated as a CrN. The hardness of weld metal were measured by Vicker’s hardness tester with the load of 200gf. As shown in Figure 12, the hardness increased with increasing of nitrogen content.

For example, weldmetal contained about 0.6\% nitrogen indicates beyond 250. Since the hardening behavior seems to relate solid solution hardening by nitrogen. So, the variation of lattice parameter were measured by X-ray diffraction method as shown Figure 13. The lattice parameter increases with increasing of nitrogen contents. Solved nitrogen makes strain in austenite.

The increase of nitrogen content is well correspond to the change of lattice parameter. Therefore, it is considered that most of nitrogen exists as solid solution.
in weld metal. Hardening behavior resulted by interstitial type nitrogen. But, in case of beyond about 0.7% nitrogen, it is estimated that the possibility of precipitation hardening by Cr2N.

4. Conclusions

Nitrogen absorption of 316L weld metal by high pressure GMA (constant arc length condition) process was investigated. The welding atmosphere was prepared by pressurized N2 and N2 and N2:CO2 gas mixture. Bead on plate welding was carried out in pressurized N2 and N2:CO2 atmosphere of 0.1 MPa-6.0MPa. The results are summarized as follows.

(1) 0.6wt% of nitrogen was absorbed in weld metal at 6.0MPa pure N2 welding atmosphere.

(2) Nitrogen content of weld metal is very lower content than equilibrium solubility of 316L stainless steel at close to melting point.

(3) Nitrogen content in weld metal under pressurized N2:CO2 mixture atmosphere is different from pure N2 atmosphere.

(4) The gas composition consist of PN=0.35MPa-PCO2=0.05MPa (total pressure=0.4MPa), it shows that maximum nitrogen content of 0.56wt%.

(5) Over 2MPa addition of CO2, decreased nitrogen content of weld metal.

(6) About 0.84 wt% nitrogen content was obtained by remelting process under high pressure.

(7) The lattice parameter and Vicker’s hardness number were increased linearly with increasing of nitrogen content.

(8) The nitrides were observed over 0.65wt% nitrogen weld metals.

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