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# Improvement of Plasma-Sprayed $\text{ZrO}_2$ Coatings by $\text{Cr}_2\text{O}_3$ Penetration and Evaluation of Cavitation Erosion <sup>†</sup>

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## Abstract

*As plasma sprayed  $\text{ZrO}_2$  coatings, like all ceramics coatings, include micro cracks and non-bonded part between flattened  $\text{ZrO}_2$  particles, such properties as mechanical strength, wear resistance, cavitation erosion are greatly reduced. The possibility of improving the cavitation erosion of  $\text{ZrO}_2$  coatings by penetration treatment with  $\text{Cr}_2\text{O}_3$  was examined. It was found that the hardness of  $\text{ZrO}_2$  coatings penetrated with  $\text{Cr}_2\text{O}_3$  increased greatly, compared with as-sprayed  $\text{ZrO}_2$  coatings. The evaluation of cavitation erosion of  $\text{ZrO}_2$  coatings was investigated by magnetostrictive cavitation tests at a frequency of 20 kHz and at a horn amplitude of 50  $\mu\text{m}$ . The cavitation erosion property was improved greatly by  $\text{Cr}_2\text{O}_3$  penetration treatment of the  $\text{ZrO}_2$  coating.*

**KEY WORDS :** (Plasma-spray) ( $\text{ZrO}_2$ ) (Coating) ( $\text{Cr}_2\text{O}_3$ ) (Sintering) (Erosion) (Cavitation) (Penetration)

## 1. Introduction

Coatings formed by plasma spraying, etc. have uncombined portions and cracks. These can be penetrated by the solvent method but coatings treated by this method generally cannot withstand cavitation for long times under severe conditions of ultrasonic vibration. It is also reported that fusion/thermal diffusion and mechanical compaction methods<sup>1-2)</sup> and coating bond improving methods<sup>3-5)</sup> of plating uncombined portions with Cu or filling pores with  $\text{Cr}_2\text{O}_3$  are in use.

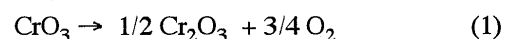
The penetration method to fill pores and microcracks in coatings with  $\text{Cr}_2\text{O}_3$  prevents the propagation and growth of cracks under stress condition and increases corrosion resistance in corrosive environments by preventing liquid from infiltrating into pores and cracks.

Wear rates and the mercury method are used to evaluate coating structures and properties<sup>3)</sup> but ultrasonic vibration has hardly been used for evaluating damage to thermal sprayed coatings up to the present. In such circumstances an attempt was therefore made to fill pores and cracks of sprayed  $\text{ZrO}_2$  coatings with  $\text{Cr}_2\text{O}_3$ , heat treat the coatings and subject them to cavitation tests.

As a result, cavitation erosion was reduced significantly. The test results are reported.

## 2. Experimental procedure

Test specimens were prepared by plasma-coating 16 mm (0.63in)-diameter steel bars with a 500  $\mu\text{m}$  thick  $\text{ZrO}_2$  layer. The specimen was penetrated as shown in the flowchart of **Figure 1**. Specifically it is immersed in  $\text{CrO}_3$  liquid and burned to fill uncombined portions of  $\text{ZrO}_2$  and pores with  $\text{Cr}_2\text{O}_3$  which is converted from Chromic anhydride  $\text{CrO}_3$  (which has the highest production rate of  $\text{Cr}_2\text{O}_3$  of chromates).



It is reported that  $\text{CrO}_3$  is converted to  $\text{Cr}_2\text{O}_3$  via  $\text{Cr}_3\text{O}_8$  and  $\text{Cr}_2\text{O}_5$  according to quation (1) at a temperature over 673K.

The specimen was immersed in chromic anhydride  $\text{CrO}_3$  for 40 sec heated to 723K for 60 min in an electric furnace open to the atmosphere and cooled down in the furnace. This immersion and heating process was repeated eight times. Specimens with a  $\text{Cr}_2\text{O}_3$  production of about 8 mg/cm<sup>2</sup> were adopted for testing<sup>4)</sup>.

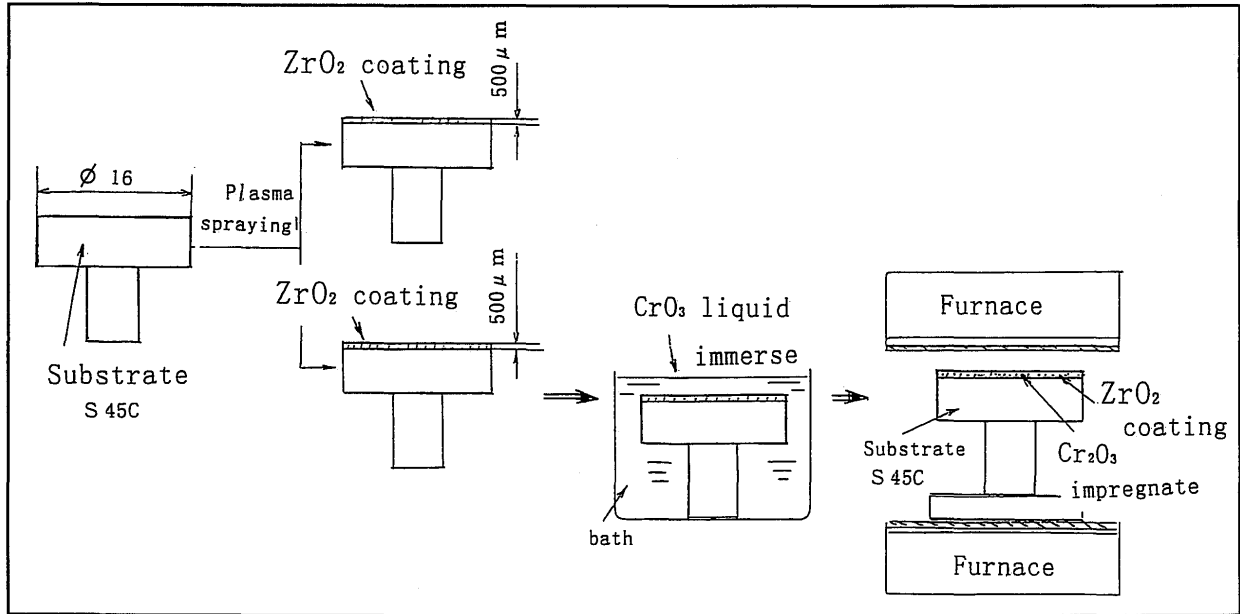
The cavitation tester used is the magnetostrictive vibration type and has a high-frequency power of 1.2 kW and a resonance frequency of  $19.5 \pm 0.5$  kHz. The vibration amplitude used in the test was kept constant at 50  $\mu\text{m}$ . The amplitude was measured with a noncontact displacement

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**Fig. 1** Dipped in  $\text{CrO}_3$  solution to penetrate, then baked to melt  $\text{CrO}_3$  and impregnate the uncombined part of  $\text{ZrO}_2$  as diagrammatically illustrated.

**Table 1** Chemical compositions and size of powders.

Material	composition (wt%)					Size ( $\mu\text{m}$ )
	$\text{ZrO}_2$	$\text{YO}_3$	$\text{Al}_2\text{O}_3$	$\text{SiO}_2$	$\text{Fe}_2\text{O}_3$	
$\text{ZrO}_2$ -8% $\text{Y}_2\text{O}_3$	—	8.36	0.02	0.09	0.07	10~45

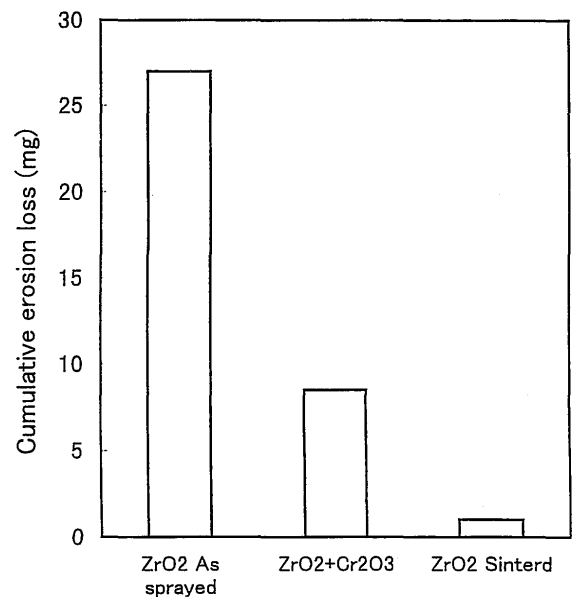
meter and a dial gauge. As shown in Fig. 1 an unsealed specimen namely a 16 mm (0.63 inch) -diameter steel bar with  $\text{ZrO}_2$  coating and a penetrated specimen, whose uncombined  $\text{ZrO}_2$  portions and pores were filled with  $\text{CrO}_3$  and burned were mounted on the vibrator with screws. Distilled water was used as the testing water and kept at  $293 \pm 2\text{K}$ . Spray  $\text{ZrO}_2$  powder, including 8%  $\text{Y}_2\text{O}_3$  by weight, was used for spraying.

The composition of the powder is shown in Table 1.

### 3. Results and discussion

#### 3.1 Relationship between erosion and elapsed time

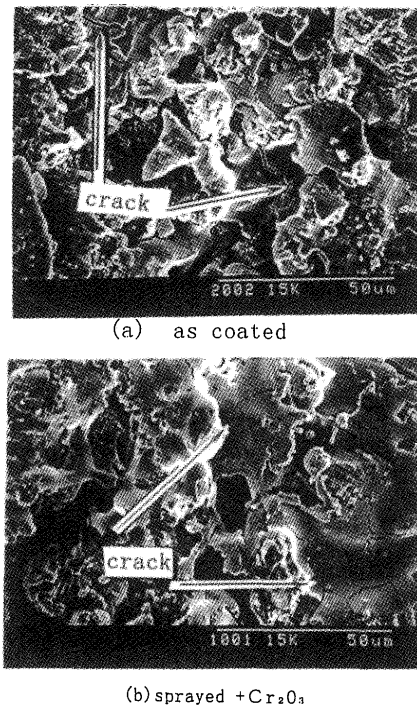
Figure 2 shows the relationship between erosion and elapsed time. When an unsealed specimen as sprayed was compared with a specimen penetrated with  $\text{Cr}_2\text{O}_3$ , the erosion of the latter was reduced to one fourth of that of the former. The  $\text{Cr}_2\text{O}_3$  - penetrated specimen showed no remarkable change in the roughness of the eroded surface. In Figure 3 the sprayed surface of the unsealed specimen in (a) is compared with that of the  $\text{Cr}_2\text{O}_3$  penetrated one in (b)



**Fig. 2** Growth of cavitation erosion as a function of time.

by using SEM observation.

There are many cracks about  $25 \mu\text{m}$  long and about  $0.3 \mu\text{m}$  wide traversing layered flat particles in (a) and also a lot of cracks in all directions in (b)<sup>2)</sup> thus revealing remarkable difference between (a) and (b) in SEM image. The  $\text{Cr}_2\text{O}_3$  - penetrated specimen tended to be tinged with a dark tone of color due to the heating effect.

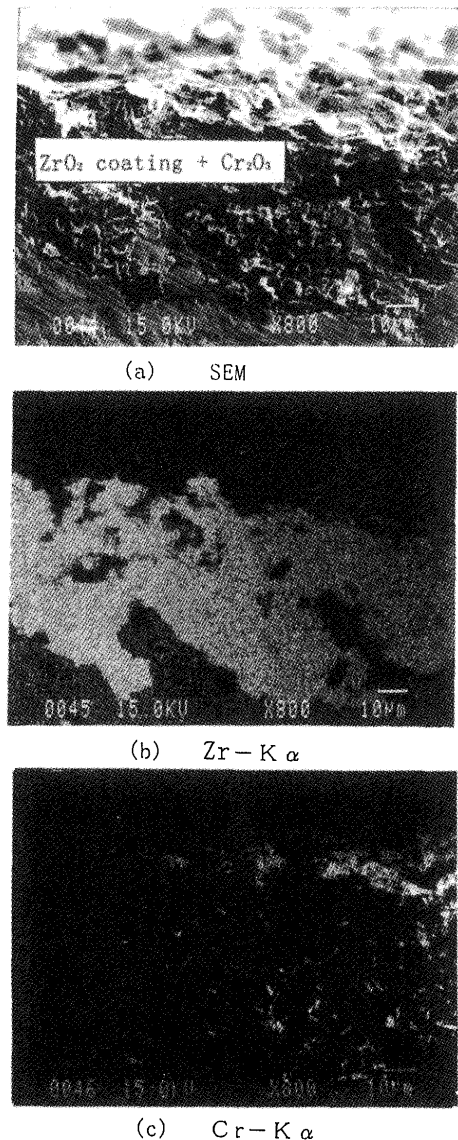


**Fig. 3** SEM images of (a) as sprayed and (b) impregnated surfaces of thermal sprayed coatings.

### 3.2 EPMA analysis of portions filled with $\text{Cr}_2\text{O}_3$

An SEM observation of the cross section of portions filled with  $\text{Cr}_2\text{O}_3$  is shown in **Figure 4** (a), a  $\text{ZrK}_\alpha$  characteristic X-ray image in (b) and a  $\text{CrK}_\alpha$  characteristic image in (c). As shown in (c), it is suggested that Cr is present between layered flat particles and in boundaries in all directions <sup>5)</sup>. **Figure 5** presents an EPMA line analysis of  $\text{Cr}_2\text{O}_3$  having infiltrated into  $\text{ZrO}_2$  coating. The almost continuous detection of  $\text{Cr}_2\text{O}_3$  in (b) proves that the penetrating treatment filled layered flat particle boundaries with  $\text{Cr}_2\text{O}_3$ . **Figure 6** shows SEM images of the surface of layered flat particles. Fig. 6 (a) indicates flat particles as large as about 50  $\mu\text{m}$  on the initial surface. Fig. 6 (b), showing a  $\text{CrK}_\alpha$  characteristic X-ray image, proves that Cr is present in particle boundaries and especially pores.

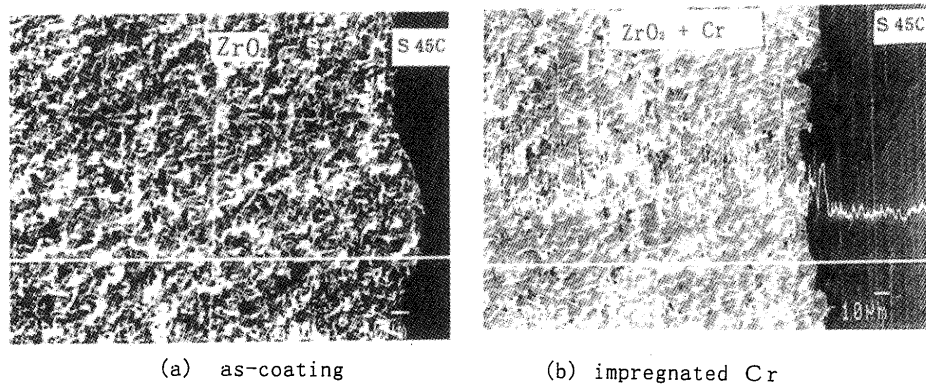
Fig. 6 (c) and (c') present  $\text{ZrK}_\alpha$  characteristic X-ray images. Fig. 6 (a') indicates that only 10 min after the test started, due to the action of cavitation impingement, particles began to be broken down in boundaries while large flat particles were reduced in size, and tended to become smaller on average. Fig. 6 (b)', presenting a  $\text{CrK}_\alpha$  characteristic X-ray image, demonstrates that, though 10 min



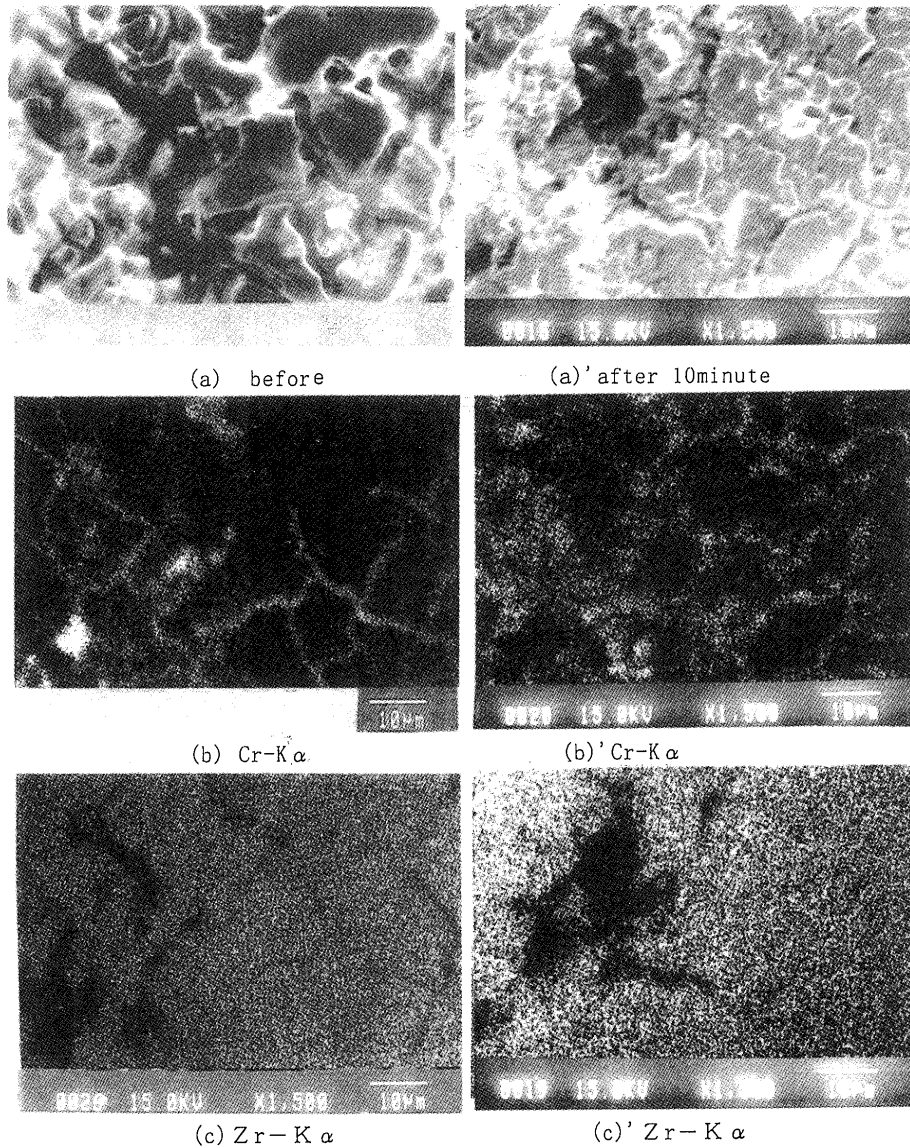
**Fig. 4** Section of  $\text{Cr}_2\text{O}_3$ -filled specimen taken (a) in SEM and with characteristic X-rays of (b)  $\text{Zr}-\text{K}_\alpha$  and (c)  $\text{Cr}-\text{K}_\alpha$ .

passed since cavitation had occurred, Cr remained without being removed by breakage, thereby preventing large eroded particles from being separated. The  $\text{Cr}_2\text{O}_3$  penetration was proved to produce a significant effect on the decrease of erosion. **Figure 7** shows changes in hardness in relation to penetration with  $\text{Cr}_2\text{O}_3$ .

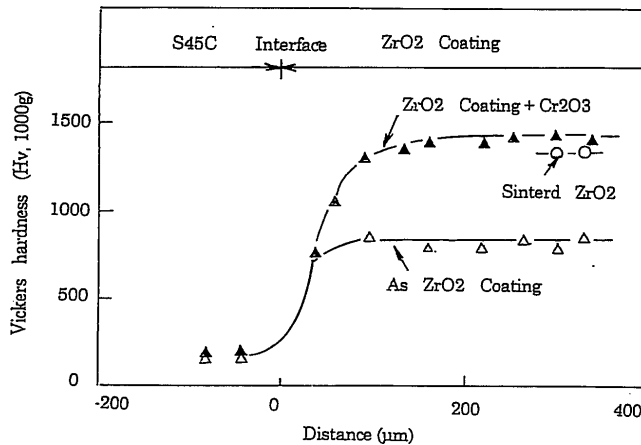
The  $\text{ZrO}_2$  coating penetrated with  $\text{Cr}_2\text{O}_3$  in (b) was 50% harder than the unsealed  $\text{ZrO}_2$  coating in (a). This is supposed to be due to the formation of a dense  $\text{Cr}_2\text{O}_3$  layer <sup>6)</sup>. It is considered that this effect produces a delay of the crack propagation and abrasion.



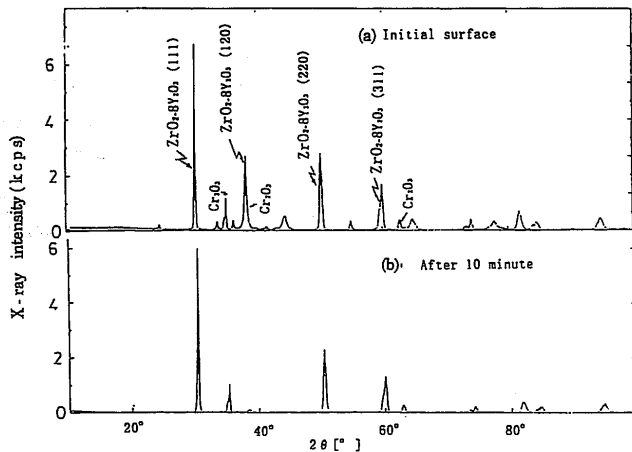
**Fig. 5** EPMA line analysis for (a) as sprayed and (b)  $\text{Cr}_2\text{O}_3$  - filled samples.



**Fig. 6** SEM before image (a) , after 10 minute (a)', characteristic X-ray image with  $\text{Cr-K}\alpha$ , (b). (b)',(c) and (c)' show the images taken with  $\text{Zr-K}\alpha$ .



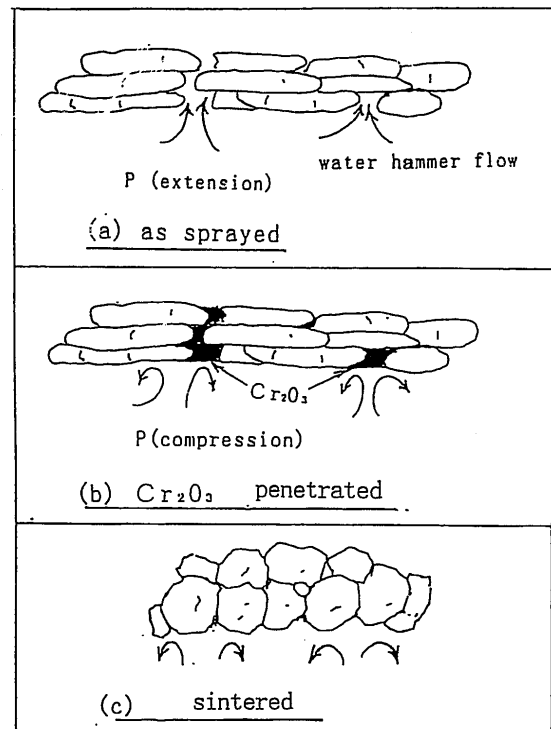
**Fig. 7** Vickers hardness distribution of cross-section of  $\text{ZrO}_2$  coating as-sprayed, heat-treatment with  $\text{Cr}_2\text{O}_3$  - filled and sintered zirconium.



**Fig. 8** XRD for the sprayed  $\text{ZrO}_2$  coating (1) before and (2) after 10 minutes of the erosion test.

### 3.3 X-ray diffraction of eroded $\text{ZrO}_2$ spray coating

Figure 8 illustrates the X-ray diffraction pattern of the eroded surface of sprayed  $\text{ZrO}_2$  coating before (1) and after the erosion (2). No remarkable change can be found at the diffraction angle of  $\theta$  even when comparison is made before the erosion starting and 10 min after it. There was a tendency for X-ray intensity (cps) to decrease in the profile 10 min after the erosion. This is supposed to be due to the disorder of sprayed  $\text{ZrO}_2$  coating crystals as a result of being subjected to severe cavitation, specifically repeated Hertzian stress and stress due to cavitation impingement<sup>7)</sup>. Figure 9 presents a model for the erosion of the layered flat particle portion of a sprayed  $\text{ZrO}_2$  coating due to ultrasonic vibration. A process is macroscopically analyzed in



**Fig. 9** Mechanism model for the erosion that may proceed under ultrasonic vibration of sprayed  $\text{ZrO}_2$  in the layer of flattened particles.

which a specimen in (a) hits the liquid surface in accordance with cavitation vibration when the vibrator descends. Then, a water hammer force  $P$  (compressive force) acts on the whole surface of the specimen and exercises a breaking effect due to Hertzian stress on layered particle boundaries, particle junctions, uncombined particles, microcracks and pores. It is supposed that the penetrating effect of  $\text{Cr}_2\text{O}_3$  on uncombined portions and microcracks was effective in preventing the above stress effects in (b). For the above reasons, the penetrating treatment using  $\text{Cr}_2\text{O}_3$  prevents the propagation and growth of cracks under various stress conditions and contributes to the increase of corrosion resistance by preventing corrosive liquid from infiltrating into pores and cracks. Thus, it was proved that the penetrated specimen was the propagation and growth of cracks under various stress conditions and contributes to the increase of corrosion resistance by preventing corrosive liquid from infiltrating into pores and cracks. Thus, it was proved that the penetrated specimen was improved about 70% in cavitation resistance as compared with the unsealed one.

#### **4. Conclusions**

Thermal sprayed  $\text{ZrO}_2$  coatings generally contain pores and microcracks. The pores and microcracks can be filled with  $\text{Cr}_2\text{O}_3$  by penetrating treatment, thus making it possible to produce a denser sprayed coating. The properties of the  $\text{Cr}_2\text{O}_3$  layer was evaluated and the following conclusions were obtained. Penetrating treatment using  $\text{Cr}_2\text{O}_3$  can fill densely pores, microcracks and uncombined portions with  $\text{Cr}_2\text{O}_3$ . The penetrated  $\text{ZrO}_2$  coating was about 50% harder than the unsealed one. The sprayed  $\text{ZrO}_2$  coating penetrated with  $\text{Cr}_2\text{O}_3$  was about 70% higher in cavitation resistance than the unsealed one.

#### **Acknowledgement**

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