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Excitation Dynamics in Disubstituted Polyacetylene

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We studied the excitation dynamics in films of disubstituted polyacetylene, a degenerate groundstate conjugated polymer, using psec transient and steady-state spectroscopies. The polymer is found to support charged and neutral topological soliton excitations concurrent with a strong intrinsic photoluminescence band with quantum efficiency, $\eta \approx 50\%$. This leads to stimulated emission in thin films and lasing in cylindrical μ cavities. The seeming contradiction of a degenerate ground-state polymer with high η is explained by the lowest excited-state ordering. [S0031-9007(99)09122-X]

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Recent advances in understanding the excited electronic states in π -conjugated polymers have led the research community to recognize the importance of the excited state ordering. The photoluminescence (PL) quantum efficiency η and the resonant third-order optical properties of these materials are determined by the relative energy and symmetry of a subset of the excited states, including a series of singlet excitons with odd (mB_u) and even (kA_{e}) parity lying below the continuum threshold [1]. If the lowest A_g exciton $(2A_g)$ is below the lowest $1B_u$ exciton, $E(2A_g) < E(1B_u)$, as in polymers with small effective dimerization δ , then η is small because of the dipole forbidden character of the lowest singlet [2]. Conversely, for $E(2A_{g}) > E(1B_{u})$, as in polymers with large δ , η is large and these polymers might be considered as active materials for displays and laser-action applications.

Among the polymers with extremely weak PL are transpolyacetylene $[t-(CH)_x]$ [3], its monosubstituted derivatives [4], polyheptadiyne [5], and polygranniline [6], which are considered nonluminescent conjugated polymers (non-LCP). These polymers have a degenerate ground-state (DGS) structure [Figs. 1(a) and 1(b)] and as such support soliton (S) excitations, both neutral [S^0 , Fig. 1(c)] and charged [S^{\pm} , Fig. 1(d)]. Based on this correlation, it is usually assumed that DGS polymers cannot exhibit PL. This was traditionally explained by the soliton model, in which the primary excitations are soliton-antisoliton $(S\overline{S})$ pairs [7,8]. However, when electron correlations are taken into account [9,10], the weak PL in these DGS polymers can be alternatively explained by simply assuming $E(2A_g) < E(1B_u)$. This is certainly true for t-(CH)_x, where $E(2A_g) = 0.96 \text{ eV}$ and $E(1B_u) = 1.5 \text{ eV}$ [11]. In addition, it is also known that energy relaxation is larger in non-LCP compared to that in LCP [12]. This gives rise to a larger PL Stokes shift in DGS polymers. For example,

in *t*-(CH)_{*x*} it was determined [11,13] that the weak IR PL peak [emitted from the relaxed $1B_u$ state, denoted hereafter as $(1B_u)_r$] is at 1.24 eV, compared to $E(1B_u) = 1.5$ eV; this amounts to a 0.26 eV Stokes shift. Thus, it may be possible to find DGS polymers with strong PL, provided that either $E(1B_u) < E(2A_g)$ or $E(1B_u)_r < E(2A_g)$. However, so far none have been found.



FIG. 1. (a),(b) The two degenerate ground-state structures (*A* and *B*) of PDPA, where R_1 and R_2 are phenyl derivative side groups. (c),(d) Neutral S^0 and charged S^{\pm} soliton excitations, respectively. (e),(f) The energy levels and optical transitions of S^0 , S^{\pm} , and polaron excitations, respectively; U_{eff} is the effective electron correlation energy and HOMO (LUMO) is the highest occupied (lowest unoccupied) molecular orbital.

In this Letter we discover a new, strongly luminescent DGS polymer, a representative of the novel family of disubstituted polyacetylenes $(CR_1CR_2)_x$ (Fig. 1), where R_1 and R_2 are phenyl derivatives (CH₃, C₂H₅, and C₆H₁₃), denoted hereafter as PDPA [14]. In particular, we have investigated the excitation dynamics of a PDPA derivative, where R_1 is a phenyl group and R_2 is a phenyl group attached to butyl [Fig. 2(b), inset] referred to as PDPA*n*Bu. In our studies we have employed steady-state and ps transient spectroscopies, such as photoinduced absorption (PA), PL, and doping-induced absorption (DA). We show that PDPA-nBu supports both charged and neutral soliton excitations, making it a member of the DGS polymer class. On the other hand, we discovered that it also shows a strong, intrinsic PL band with $\eta \simeq 50\%$, which may be used to produce microlasers (μ lasers) in the green spectral range.

Doping was performed either by exposing the film to iodine vapor forming $(polymer)^+I_3^-$ or, electrochemically, by forming $(polymer)^+BF_4^-$. PA and PL were excited by either an Ar⁺ laser at 458 nm or a monochromatized xenon lamp, and probed in the spectral range from 0.1 to 3.4 eV. In order to determine the spin state



FIG. 2. Evolution of DA spectra of PDPA-*n*Bu upon electrochemically doping with BF_4^- (a) and by vapor doping with I_3^- (b). The inset of (a) shows the difference in DA spectra and the inset to (b) shows the PDPA-*n*Bu repeat units. δS^+ and the IRAV transitions are assigned.

of the various photoexcitations, we used the PA detected magnetic resonance (PADMR) technique [15], with μ waves at ~3 GHz. Magnetic field-swept *H*-PADMR spectra (at fixed wavelength λ) are presented, as well as λ -PADMR spectra (at fixed *H* while sweeping λ).

The transient PL and lasing measurements were obtained using a mode-locked Nd:YAG regenerative laser amplifier, tripled at 3.45 eV, with 120 ps pulse duration and up to 100 μ J energy per pulse. For the PL decay we used a streak camera with 10 ps time resolution. Neat thin films of PDPA-*n*Bu were spun cast on quartz or single crystal potassium bromide substrates from polymer solutions in chloroform. The polymer μ -ring lasers were obtained by dipping commercially available optical fibers into *saturated* chloroform solutions, forming thin polymer rings around the glass cylindrical core. The emitted light was collected in the plane of the μ -ring with a spectral resolution of ≈ 1 Å.

One way to identify DGS polymers is by studying their elementary excitations. Degenerate ground-state polymers support topological soliton excitations (S^0, S^{\pm}) which are in the form of a domain wall separating the two DGS structures [16]. These quasiparticles may be identified by their reversed spin-charge relationship and optical transitions [3]. S^0 carries spin- $\frac{1}{2}$, whereas S^{\pm} are spinless. Also, both S^0 and S^{\pm} show a single absorption band, δS , below the optical gap. The energy difference $\delta S^0 - \delta S^{\pm}$ is the soliton effective correlation energy U_{eff} [Fig. 1(e)] [17]. Excitations in non-DGS polymers, on the contrary, may be accommodated as charged polarons (P^{\pm}) [3], which carry spin- $\frac{1}{2}$ and have *two* absorption bands below the gap [Fig. 1(f)] [18].

We show that PDPA-nBu supports soliton excitations and therefore has DGS structure. Figure 2 shows the evolution of the absorption spectrum $\alpha(\omega)$ upon BF₄⁻ (a) and I_3^- (b) doping. Upon BF₄ doping, only a *single* dopinginduced absorption band at ≈ 1.2 eV appears and the interband absorption bleaches. From the difference in DA spectra in Fig. 2(a) (inset), it is seen that the total oscillator strength is conserved upon doping [3], so that an additional DA band below 0.8 eV is improbable. This is verified upon I_3^- doping, using a Fourier-transform infrared spectrometer down to 500 $\rm cm^{-1}$. As can be seen in Fig. 2(b) there is only a single DA band at $\approx 1.2 \text{ eV}$, accompanied by an infrared active vibration (IRAV) band at ≈ 0.15 eV, which is a clear signature for doping-induced charged excitations [19]. Using an in situ ESR spectrometer, we found that only seven spins (at $g \approx 2$) per 1000 electron charges were created upon doping, showing that the doping-induced charged species are practically spinless. Then, based on the reversed spin-charge relationship and the single DA band found below the gap, even at low doping levels, we conjecture that S^+ excitations are formed in PDPA-*n*Bu upon *p*-doping.

Soliton excitations can also be created in PDPA-nBu upon photoexcitation, as shown in Fig. 3. Pristine



FIG. 3. Photoexcitations in PDPA-*n*Bu. (a),(b) The PA spectra of pristine and photo-oxidized films, respectively, where the transitions δS^{\pm} and δS^{0} are assigned. Insets of (a) and (b) show, respectively, the excitation dependence spectra of δS^{0} and δS^{\pm} . (c) λ -PADMR spectrum at H = 990 G. Inset: *H*-PADMR spectrum probed at 1.7 eV.

PDPA-nBu films show a single long-lived PA band at $\approx 1.7 \text{ eV}$ [Fig. 3(a)] with no correlated photoinduced IRAV, indicating neutral photoexcitations. Comparing it with the DA band shown above, we estimate a low quantum efficiency (QE) of $\approx 0.2\%$. Furthermore, PADMR studies of this sample reveal that this long-lived PA band is associated with spin- $\frac{1}{2}$ excitations. The *H*-PADMR spectrum [Fig. 3(c), inset] measured at 1.7 eV shows a correlated spin doublet at $H_0 = 990$ G $(g \approx 2)$; the λ -PADMR [Fig. 3(c)], measured at H_0 , tracks this PA band. Similar to t-(CH)_x [18], there is also a negative PADMR tracking δS^+ in this film. Therefore, the longlived PA in pristine PDPA-nBu films is due to neutral species with spin- $\frac{1}{2}$, and may be identified as the δS^0 transition [Fig. 1(e)]. The δS^0 QE per absorbed photon [Fig. 3(a), inset] basically shows a step-function response with very low QE at the optical band edge, indicating that S^0 photogeneration in pristine films is small and correlated with thermalized excitons.

However, when the pristine film was exposed to either moderate laser light or sun light for a few hours at ambient atmosphere, photo-oxidation-related impurities caused dramatic changes in the PA spectrum [Fig. 3(b)]. In addition to the δS^0 PA band, the spectrum also contains a PA band at ~1.1 eV together with a correlated IRAV at ~0.13 eV. The similarity with the DA spectra in Fig. 2 leads us to identify the PA band at 1.1 eV with δS^{\pm} and conclude that S^{\pm} are also photogenerated in photooxidized films, with QE ~ 0.1%. The excitation dependence of δS^{\pm} QE is shown in the inset of Fig. 3(b). Similar to δS^0 QE spectrum in Fig. 3(a), the δS^{\pm} QE spectrum here also abruptly increases at the optical band edge, indicating that S^{\pm} photogeneration in photo-oxidized films is a competing process to S^0 photogeneration, and is also correlated with thermalized excitons. Moreover, using the relation $U_{\rm eff} = \delta S^0 - \delta S^{\pm}$ [17] [Fig. 1(e)] and $\delta S^0 = 1.7$ eV and $\delta S^{\pm} = 1.1$ eV, we find $U_{\rm eff} = 0.6$ eV.

PDPA-*n*Bu films also show a strong PL band in the green/yellow spectral range, with full width at half maximum (FWHM) of ≈ 100 nm [≈ 0.4 eV, see Fig. 4(a)] that has already been used for the fabrication of light-emitting diodes with remarkable electron/photon efficiency [20].



FIG. 4. (a) PL spectrum in PDPA-*n*Bu film; the insets show the excitation dependence of the PL QE and the PL transient decay, respectively. (b) Laser modes in a PDPA-*n*Bu μ -ring (shown in the inset) wrapped around a 128 μ m diameter optical fiber core. The spectral resolution is ≈ 1 Å.

The 0-0 PL transition [or $(1B_u)_r$] is at 2.5 eV, indicating a moderately large Stokes shift (≈ 0.25 eV), comparable to that found in *t*-(CH)_x [13]. The absolute PL QE, excited at $\hbar \omega = 2.7$ eV and measured with an integrating sphere, was found to be $\eta \approx 50\%$. The η excitation spectrum [Fig. 4(a), inset] is similar to the δS^0 QE spectrum [Fig. 3(a)] and shows a step-function increase at 2.5 eV which, however, has high QE, indicating that the PL is an intrinsic process.

We also measured the PL decay at low excitation intensity [Fig. 4(a), inset], and found an exponential decay with lifetime $\tau \approx 600$ ps. In order to find the radiative lifetime τ_r , we use the following relation [21]: $\eta = b\tau/\tau_r$, where *b* is the $(1B_u)_r$ photogeneration quantum yield. Assuming $b \approx 1$ [21], and using the experimental values $\eta = 50\%$ and $\tau \approx 0.6$ ns, we find $\tau_r \approx 1.2$ ns. This is close to τ_r of poly(p-phenylene vinylene) (PPV) based polymers [21], showing that the $1B_u$ exciton transition dipole moment and wave function extent in PDPA-*n*Bu are similar to those in PPV. This is also in agreement with the intrinsic character of the PL mechanism in PDPA-*n*Bu.

Neat PDPA-*n*Bu films show amplified spontaneous emission at high excitation intensity *I* above a threshold intensity, $I_{th} \simeq 1 \ \mu J/mm^2$ [22], similar to I_{th} found in PPV films. Under these conditions, the PL band shows a dramatic spectral narrowing from FWHM of 100 nm (0.4 eV) at low *I* to less than 10 nm (40 meV) at $I > I_{th}$. The process can be used to produce μ -lasers, provided a feedback mechanism exists. This is shown in Fig. 4(b), where we show the emission spectrum of a PDPA-*n*Bu μ -ring laser deposited around an optical fiber with a core diameter of 128 μ m [Fig. 4(b), inset]; I_{th} in this case was extremely small, of order 10 nJ/pulse [23], again in agreement with PPV μ -lasers. The laser resonant modes are resolution limited (<1 Å), showing a relatively high quality factor *Q* of order 5000.

In order to explain the seeming contradiction in the photoexcitation properties of PDPA-*n*Bu that are mainly excitons with high η , but also form neutral and charged solitons, with low QE, we note that there are two possible relaxation channels in DGS polymers following excitations into the $1B_u$ [10]. These are the ionic channel, in which $1B_u$ relaxes to $(1B_u)_r$, and the covalent channel, in which $1B_u$ relaxes to $2A_g$ by emitting an odd parity phonon [10]. Excitations in the ionic channel may luminesce, whereas excitations in the covalent channel may decompose into a pair of triplets [24], which further decompose into a quartet of S^0 excitations [25]. The QE of these two channels depends on the order of the lowest lying excitonic states. If $E(2A_g) < E(1B_u)$, the covalent channel dominates. This may be the case in t-(CH)_x [11], where S^0 are photogenerated with high QE [15] and PL is extremely weak [13]. On the other hand, if $E(2A_g) > E(1B_u)$ or $E(2A_g) > E(1B_u)_r$, the ionic channel dominates and the polymer may show PL with high QE, as is the case here. However, whether $E(2A_g) > E(1B_u)$ or $E(2A_g) > E(1B_u)_r$ in PDPA-*n*Bu remains to be determined in future work.

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