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# Kinetics of denaturation and renaturation processes of double-stranded helical polysaccharide, xanthan in aqueous sodium chloride

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## ABSTRACT

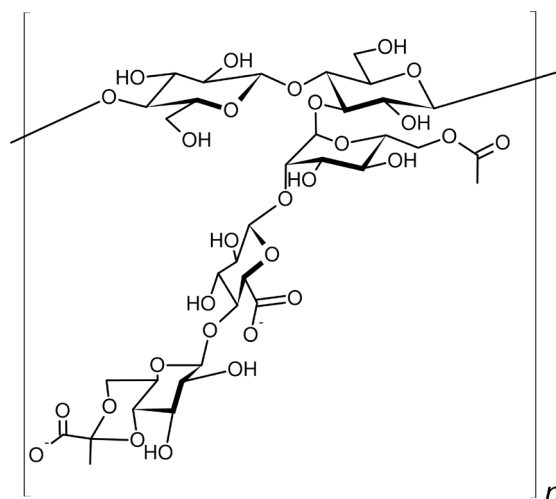
Circular dichroism (CD) and small-angle X-ray scattering (SAXS) measurements were made for three xanthan samples, a double helical polysaccharide, in 5 or 10 mM aqueous NaCl after rapid temperature change to investigate the kinetics of the conformational change between the ordered and disordered states. After the rapid heating, the CD signal mainly reflecting the carbonyl groups on the side chains quickly changed ( $< 150$  s) while the scattering intensity from SAXS around  $q$  (magnitude of the scattering vector) =  $1 \text{ nm}^{-1}$  changed more gradually, reflecting the main-chain conformation. The difference between CD and SAXS implies us the intermediate conformation which can be regarded as a loose double helix. The SAXS profile in the rapid cooling process showed that the loose double helical structure was constructed within 150 s, but the CD signal slowly changed with around 2 days to recover the native tight double helix.

Keywords: double helix; conformational change; small-angle X-ray scattering; circular dichroism; intermediate state

*Abbreviations:*  $c$ , polymer mass concentration;  $c_d$ , denatured concentration; CD, circular dichroism;  $C_s$ , salt concentration;  $d$ , bead diameter;  $D$ , dispersity index;  $L$ , contour length;  $L_w$ , weight-average contour length;  $L_k$ , Kuhn segment length;  $k$ , first-order rate constant;  $K$ , optical constant; MALS, multi-angle light scattering;  $M_L$ , molar mass per unit contour length;  $M_n$ , number-average molar mass;  $M_w$ , weight-average molar mass; PEO, polyethylene oxide;  $P(q)$ , form factor;  $q$ , magnitude of the scattering vector;  $R_g$ , radius of gyration; RI, refractive index;  $R_q$ , Rayleigh ratio; SAXS, small-angle X-ray scattering; SEC, size-exclusion chromatography;  $S(q)$ , structure factor;  $t$ , time;  $V_e$ , retention volume;  $v_p$ , partial specific volume;  $z$ , number of moles of electron per unit mass of the solute;  $\rho_{e,s}$ , electron density of the solvent;  $\Delta I(q)$ , excess scattering intensity;  $\eta_{sp}$ , specific viscosity;  $[\theta]$ , specific ellipticity;  $\lambda_0$ , wavelength in a vacuum;  $\partial n/\partial c$ , refractive index increment.

## 1. Introduction

Xanthan or xanthan gum, for which representative chemical structure is shown in Fig. 1 but the actual chemical structure varies (Kool, Gruppen, Sworn, & Schols, 2013), is an extracellular polysaccharide. It has multiple anionic groups on each side chain. Xanthan is therefore well soluble both in water and saline. It is widely used for food additives (Hublik, 2012; Nishinari, Turcanu, Nakauma, & Fang, 2019) and biomedical applications (Kumar, Rao, & Han, 2018) because dilute solution of xanthan has significantly high viscosity. Since the rheological properties of xanthan in aqueous media are substantially influenced by the chain conformation, its dimensional, hydrodynamic, spectroscopic, and thermodynamic properties have been widely investigated (Dalheim, Christensen, Comesse, & Renou, 2020; Morris, 2019; Rinaudo, 2008; Sato & Matsuda, 2009) to clarify the conformational properties in solution. Even though the main chain of xanthan consists of  $\beta$ -1,4-glucan, xanthan can form antiparallel double helix in the crystal (Okuyama et al., 1980). The double helical chains were also found in the electron micrograph (Stokke, Elgsaeter, & Smidsrod, 1986; Stokke, Smidsrød, & Elgsaeter, 1989) and the atomic force micrograph (Camesano & Wilkinson, 2001; Capron, Alexandre, & Muller, 1998; Moffat, Morris, Al-Assaf, & Gunning, 2016). One of the representative characteristics of xanthan is that the double helical structure can exist in the aqueous media. The double helical xanthan transforms to the single disordered chain in a low ionic strength aqueous solution with raising temperature (Kitamura, Takeo, Kuge, & Stokke, 1991; Liu & Norisuye, 1988a, 1988b; Liu, Sato, Norisuye, & Fujita, 1987). The ordered double helical structure has significantly high chain stiffness, for which Kuhn's segment length  $L_k$  was estimated to be  $240 \pm 40$  nm (Sato, Norisuye, & Fujita, 1984a) in terms of the wormlike chain model (Kratky & Porod, 1949; Yamakawa & Yoshizaki, 2016). This value is much larger than the disordered xanthan ( $L_k = 42$  nm) (Liu et al., 1987). Furthermore, the double-helical hairpin like conformation can be formed in the renatured process on the basis of the molar mass and the intrinsic viscosity in aqueous NaCl (Capron, Brigand, & Muller, 1997). This hairpin like conformation can also be found by the microscopic observations (Capron et al., 1998; Moffat et al., 2016; Stokke et al., 1989). Furthermore, more complicated aggregates are found by means of light scattering (Matsuda, Biyajima, & Sato, 2009; Matsuda, Okumura, & Tasaka, 2018; Matsuda, Sugiura, Okumura, & Tasaka, 2016; Merino-González & Kozina, 2017; Papagiannopoulos, Sotiropoulos, & Radulescu, 2016) and atomic force microscopy (Ikeda, Gohtani, Nishinari, & Zhong, 2012; Matsuda, Sugiura, Mays, & Tasaka, 2015; Teckentrup et al., 2017) depending on the renatured condition. The resulting conformation influences their rheological behaviors (Choppe, Puaud, Nicolai, & Benyahia, 2010; Oviatt & Brant, 1994; Papagiannopoulos, Sotiropoulos, & Pispas, 2016). Conformational change of xanthan has also recently been studied by some modern method with continuous change of ionic strength (Savi, de Freitas, Sasaki, Koop, & Silveira, 2015) and the combination of viscosity and infrared absorption at low salt concentrations (Brunchi, Avadanei, Bercea, & Morariu, 2019).



**Fig. 1.** Chemical structure of xanthan.

The kinetics of the conformational change may be an important research topic taking again into consideration that the obtained renatured conformation can be controllable by the solvent and temperature condition in the denature-renature processes. Indeed, optical rotation after abrupt change of the salt concentration revealed that the renature process was the first order reaction (Norton, Goodall, Frangou, Morris, & Rees, 1984; Norton, Goodall, Morris, & Rees, 1980). More recently, relatively slow change of the viscoelasticity was found after abrupt temperature changes (Bercea & Morariu, 2020). However, kinetics of formation and deformation of the double stranded conformation has not been fully understood.

Recent development of synchrotron-radiation X-ray sources and detectors allows us to make time-resolved small-angle X-ray scattering (SAXS) measurement with high accuracy (Otsubo & Terao, 2021). Analyses of the SAXS data yield the content of the double stranded conformation which cannot be observed in the previous studies because circular dichroism (CD) and optical rotation can be affected by the side chain conformation and rheological data reflects the intermolecular association and entanglement. In this study, we therefore attempted to make both time-resolved SAXS and CD measurements for relatively low molar mass xanthan for which weight-average molar mass  $M_w$  ranges between 300 and 500 kg mol<sup>-1</sup> after rapid temperature change to clarify the kinetics of the temperature-induced conformational change. Synchrotron radiation CD measurements in the wavelength range down to the ultraviolet region were also examined to explore the local conformation after the abrupt temperature change.

## 2. Materials & methods

### 2.1. Xanthan samples and their characterization

Three low molar mass xanthan samples were prepared in the following manner. A xanthan sample (TCI, Tokyo, Japan) was dissolved in 100 mM aqueous NaCl to obtain the solution with the polymer mass concentration  $c = 10$  mg mL<sup>-1</sup>. The solution was sonicated with a Branson Sonifier Advanced 450 (20 kHz) in an ice bath for 6 h to reduce the chain length. The resultant solution was poured into a large amount of acetone to precipitate the xanthan sample. The obtained sample was designated as **X461k**. This xanthan sample was further purified by the fractional precipitation with 100 mM aqueous NaCl as a solvent and acetone as a precipitant after the counter ion was completely changed to sodium in the manner reported previously (Sato, Norisuye, & Fujita, 1984b). The appropriate middle fractions, **X341k** and **X400k**, were also chosen for this study. <sup>1</sup>H

NMR measurements were examined for the D<sub>2</sub>O solution of **X341k** to determine the acetate/pyruvate ratio as 1.5 with the literature method (Brunchi et al., 2019).

Size-exclusion chromatography (SEC) equipped with a multi-angle light scattering photometer (MALS) and a differential refractive index (RI) detector was carried out for the three samples in aqueous NaCl with the salt concentration  $C_s = 100$  mM to determine  $M_w$ , the radius of gyration  $R_g$ , and the dispersity index  $\mathcal{D}$  defined as the ratio of  $M_w$  to the number-average molar mass  $M_n$ . A GPC-101 chromatography system (Shodex, Japan), a DAWN HELEOS II MALS detector (Wyatt technology Co., USA, the wavelength  $\lambda_0$  in a vacuum was 658 nm), and a RI detector were used with serially connected SEC columns, that is, a guard column (OHpak SB-G, Showa Denko) and two SEC columns (OHpak SB-806M-HQ, Showa Denko) set in a column oven at 40 °C. The universal calibration method (Grubisic, Rempp, & Benoit, 1967; Pasch, 2012) was also applied for the RI chromatogram to confirm the validity of  $M_w$  with the standard polyethylene oxide (PEO) samples (Tosoh, Japan) ranging in  $M_w$  from 24 to 790 kg mol<sup>-1</sup>; the literature relationship between the intrinsic viscosity and  $M_w$  for PEO (Kawaguchi, Imai, Suzuki, Miyahara, & Kitano, 1997) and for xanthan (Sato, Kojima, Norisuye, & Fujita, 1984) were utilized to calculate molar mass at each elution volume. The SEC-MALS measurement was also made for the solution treated at 85 °C for 30 min and at 20 °C for 48 h in 5 or 10 mM aqueous NaCl with the denatured concentration  $c_d$  of about 3 or 6 mg mL<sup>-1</sup> to estimate the molecular characteristics of renatured samples. The column temperature was set to be 30 °C. The refractive index increment  $\partial n/\partial c$  at  $\lambda_0 = 658$  nm was estimated from the literature values (Liu et al., 1987; Sato, Norisuye, et al., 1984b) in 10 and 100 mM aqueous NaCl. The  $\partial n/\partial c$  value in 5 mM aqueous NaCl was assumed to the same as that at  $C_s = 10$  mM because of substantially no  $C_s$  dependence of  $\partial n/\partial c$ .

Aqueous NaCl with  $C_s = 5$  and 10 mM was chosen as the solvents for the following SAXS, CD, and viscosity measurements. Each xanthan sample was dried in a vacuum at least 12 hours before solution preparation. Polymer mass concentration  $c$  of xanthan was determined from the weight ratio of xanthan to the solution with the solvent density.

## 2.2. Circular dichroism

CD measurement was performed for the three xanthan samples in 5 mM and 10 mM aqueous NaCl with the CD spectropolarimeter of BL-12 in HiSOR (Hiroshima, Japan) for which details are available elsewhere (Matsuo & Gekko, 2013) and/or a Jasco J720WO spectropolarimeter to determine the specific ellipticity  $[\theta]$ , the ellipticity divided by the path length and  $c$ . A specially designed cell with CaF<sub>2</sub> windows (Matsuo, Sakai, Matsushima, Fukuyama, & Gekko, 2003) was used for the former equipment to evaluate the ellipticity down to  $\lambda_0 = 180$  nm. The path length was set as 0.1 mm with an appropriate spacer. Two rectangular quartz cells with the path lengths of 2 or 5 mm were used for the latter equipment. For the rapid temperature change experiment, we installed the cell into the cell holder controlled at target temperature. Temperature inside the cell became the targeted temperature within 150 s for the apparatuses.

## 2.3. Small-angle X-ray scattering

SAXS measurements with rapid temperature change were carried out at the BL40B2 beamline in SPring-8 and at the BL-6A beamline (Shimizu et al., 2013) in KEK-PF. A quartz capillary cell with the diameter of 2.0 mm $\phi$  was fixed in an aluminum block. The same capillary cell was used for the solution, solvent, and the reference solution, that is, a standard PEO sample **SE-5** (Tosoh, Japan) with the  $M_w$  and  $\mathcal{D}$  being 37.6 or 44.2 kg mol<sup>-1</sup> and 1.03 or 1.17 both depending on the lot number, respectively. The wavelength  $\lambda_0$  was chosen as 0.10 and 0.15 nm

for SPring-8 and KEK-PF, respectively. The camera length, that is, the distance from the sample to the detector, was 4.2 m and 2.6 m for the two beamlines. Dectris PILATUS3 2M and PILATUS 1M detectors were used to determine the scattering intensity as a function of the magnitude  $q$  of the scattering vector with a SAnGLer circular averaging software (Shimizu et al., 2016). Silver behenate was used to determine the  $q$  value of each pixel on the detectors. The temperature was controlled by a Peltier element. The excess scattering intensity  $\Delta I(q)$  was evaluated by the numerical difference of scattering intensities between the solution and the solvent; note that each scattering intensity was calibrated by the intensity of the direct beam detected at the lower end of the solution cell to compensate both the intensity fluctuation of the incident light and the transparency of the solution. The ratio  $R_q/K$  of the Rayleigh ratio  $R_q$  to the optical constant  $K$  was estimated from the following equation (Glatter & Kratky, 1982; Ishida, Yoshida, & Terao, 2019)

$$\frac{R_q}{K} = M_{w,\text{PEO}} \left( \frac{\Delta z_{\text{PEO}}}{\Delta z} \right)^2 \left[ \frac{c_{\text{PEO}}}{\Delta I_{\text{PEO}}(q)} \right]_{c \rightarrow 0, q \rightarrow 0} \Delta I(q) \quad (1)$$

where the subscript PEO denotes the value for the standard PEO sample and  $\Delta z$  is defined as

$$\Delta z = z - v_p \rho_{e,s} \quad (2)$$

Here,  $z$ ,  $v_p$ , and  $\rho_{e,s}$  are the number of moles of electron per unit mass of the solute, the partial specific volume, and the electron density of the solvent, respectively. These values were calculated from the density of the solution and the solvent.

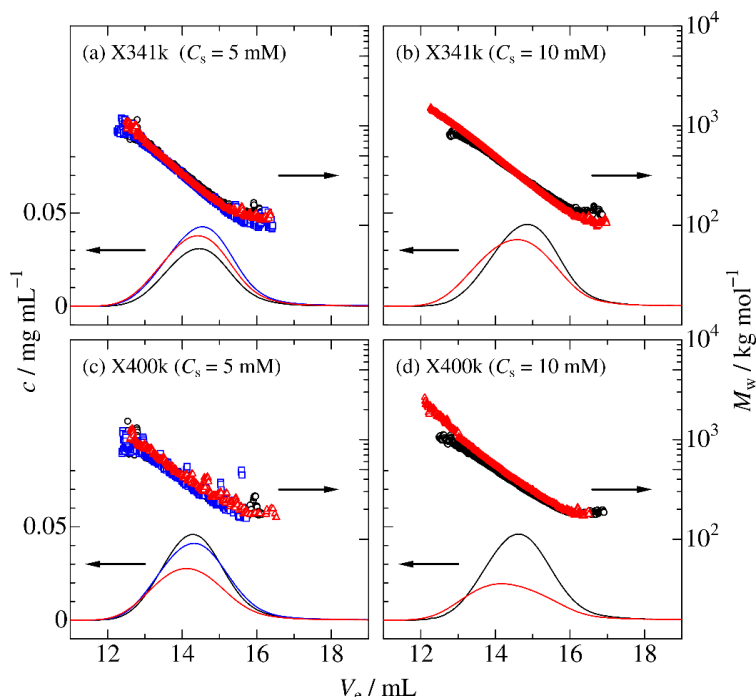
#### 2.4. Viscometry

The specific viscosity  $\eta_{sp}$  in the renatured process for **X341k** in 10 mM aqueous NaCl was measured at 20 °C with a Ubbelohde type viscometer. Just after the solution with  $c = 6 \text{ mg mL}^{-1}$  was treated at 80 °C for 1 h and abruptly cooled to 17 °C, it was diluted to  $c = 1 \text{ mg mL}^{-1}$ . Then, the solution was poured into the viscometer thermostated at 20 °C.

### 3. Results & discussion

#### 3.1. Molar mass of as-prepared and renatured xanthan samples

The obtained SEC-MALS data, the retention volume  $V_e$  dependence of  $c$  and  $M_w$ , are shown in Fig. 2. The averaged  $M_w$  and  $D$  values calculated from the data for each sample are summarized in Table 1. The evaluated  $M_w$  values for the as-prepared samples were consistent with those from the universal calibration method within the experimental error ( $\pm 5\%$ ) while the  $D$  values are larger, that is 1.6, 1.6, and 1.8 for **X341k**, **X400k**, and **X461k**, respectively. This is probably because of insufficient resolution of the current column set for our xanthan samples to evaluate  $M_n$  accurately, suggesting the actual  $D$  values may be between those from the two methods. The resulting  $M_w$  values for **X341k** at  $C_s = 5 \text{ mM}$  are smaller than that in 100 mM aqueous NaCl, indicating partly unfolded structure in the low ionic strength solvent. The molar mass values of the renatured samples are substantially the same as or somewhat larger than the as-prepared sample except for **X341k** at  $C_s = 5 \text{ mM}$ , suggesting the existence of double helical aggregates composed of more than two xanthan molecules as is found in the previous research (Matsuda et al., 2009). Indeed, the gyration radii  $R_g$  data for some renatured samples were appreciably smaller than that for the native samples with the same  $M_w$  as illustrated in Fig. S1. Somewhat larger  $\eta_{sp}/c$  value for the renatured solution as shown in Fig. S2 supports this suggestion. This figure also shows that almost no time dependence of  $\eta_{sp}/c$  was observed in the range of  $t < 1000 \text{ s}$  after the cooling.



**Fig. 2.** Retention volume  $V_e$  dependence of the polymer mass concentration  $c$  (curves) and the weight-average molar mass  $M_w$  (symbols) for (a) **X341k** at  $C_s = 5$  mM, (b) **X341k** at  $C_s = 10$  mM, (c) **X400k** at  $C_s = 5$  mM, and (d) **X400k** at  $C_s = 10$  mM. Black, blue, and red symbols (or curves) denote the data for the as-prepared solution, the renatured solution at  $c_d$  between 3.0 and 3.1 mg mL<sup>-1</sup>, and the renatured solution at  $c_d$  between 5.4 and 6.4 mg mL<sup>-1</sup>, respectively.

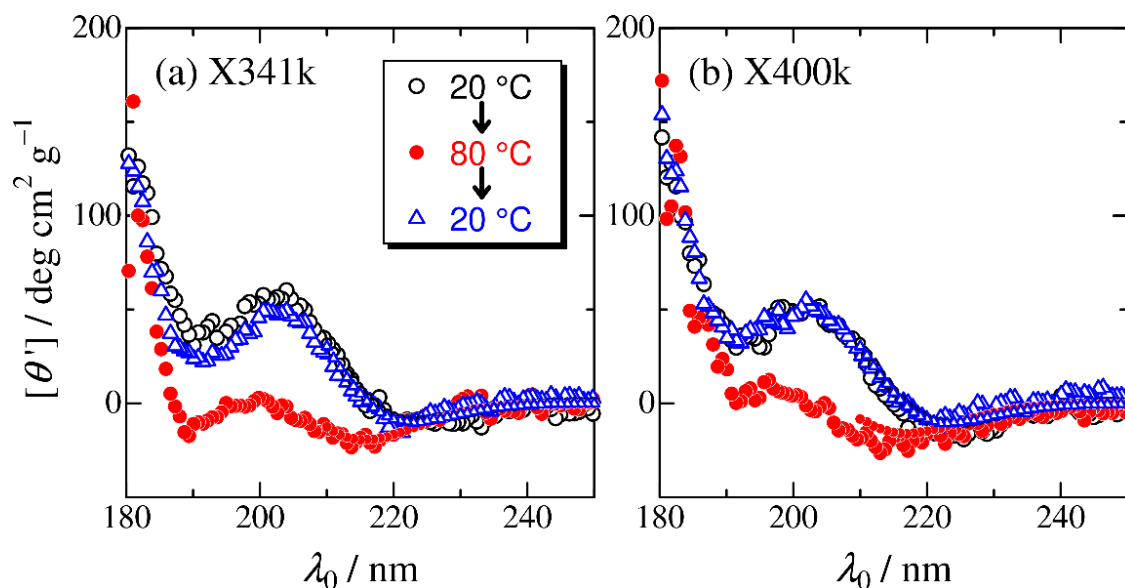
**Table 1**  
Molar masses of xanthan samples

Sample	$c_d$ / mg mL <sup>-1</sup>	$C_s$ / mM	$M_w$ / kg mol <sup>-1</sup>	$\mathcal{D}$
<b>X341k</b>	As Prepared	100	341	1.3 (1.6 <b>a</b> )
	As Prepared	10	331	1.2
	6.1	10	413	1.4
	As Prepared	5	298	1.2
	3.1	5	285	1.3
	6.4	5	322	1.4
<b>X400k</b>	As Prepared	100	400	1.2 (1.6 <b>a</b> )
	As Prepared	10	390	1.2
	5.4	10	573	1.4
	As Prepared	5	384	1.2
	3.0	5	404	1.2
	6.2	5	534	1.3
<b>X461k</b>	As Prepared	100	461	1.1 (1.8 <b>a</b> )

**a:** From universal calibration.

### 3.2. Circular dichroism spectra including vacuum ultraviolet region

CD spectra for the xanthan samples in 10 mM aqueous NaCl are illustrated in Fig. 3, in which the data obtained from the two apparatuses are substantially the same for the overlapped  $\lambda_0$  range between 210 and 250 nm. It was difficult to acquire the data below  $\lambda_0 = 180$  nm even by using the synchrotron radiation owing to the absorption of NaCl. The specific ellipticity  $[\theta']$  for the disordered xanthan at 80 °C in the range between  $\lambda_0 = 187$  and 218 nm are appreciably smaller than those for the ordered xanthan at 20 °C. This  $\lambda_0$  range is suitable to observe the conformational change of xanthan because of smaller difference in the vacuum ultraviolet region below  $\lambda_0 = 187$  nm. Essentially the same CD spectra were found at 20 °C both before and two days after the heating, indicating that local optically active structure of renatured xanthan is substantially the same as that for the as-prepared solution.



**Fig. 3.** CD spectra for (a) **X341k** and (b) **X400k** in 10 mM aqueous NaCl. Polymer mass concentration was adjusted around  $6 \text{ mg mL}^{-1}$ . Black unfilled circles, red filled circles, and blue triangles indicate the data for native sample at 20 °C, denatured sample at 80 °C, and renatured sample (at 20 °C) incubated for 2 days after cooled from 80 °C. Large and small symbols, data from HiSOR and the lab equipment, respectively.

### 3.3. Temperature-induced conformational change detected by CD

The temperature of order-disorder conformational change was estimated by the CD measurement with raising temperature as illustrated in Fig. S3 for **X341k** and **X461k** in aqueous NaCl. The specific ellipticity  $[\theta']_{210}$  at  $\lambda_0 = 210$  nm and at  $C_s = 10$  mM decreases rapidly between 50 and 65 °C both for the samples, indicating that the current xanthan samples have the completely ordered structure at 20 °C and the mostly disordered conformation at 80 °C. Lower conformational change temperature between 45 and 60 °C was observed in 5 mM aqueous NaCl.

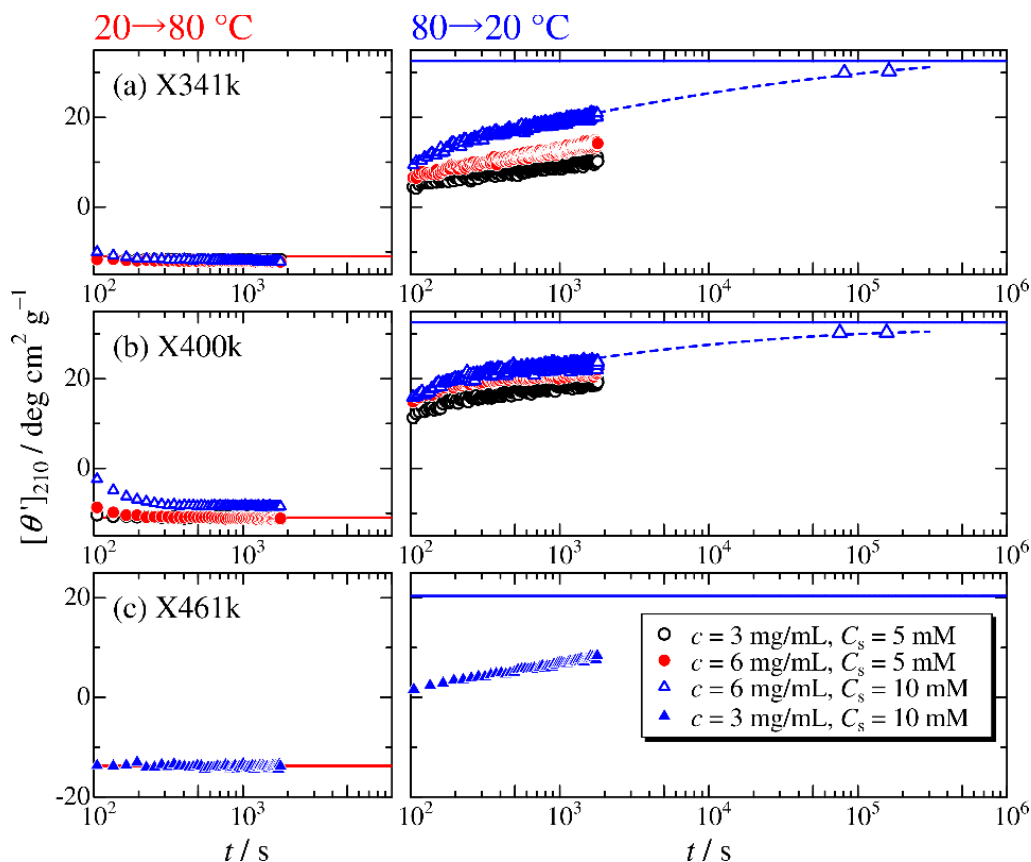
### 3.4. Time course of CD after rapid heating or cooling

Time-resolved data of  $[\theta']_{210}$  after the rapid heating from 20 °C to 80 °C are displayed in Fig. 4 for the three xanthan samples along with those for the rapid cooling from 80 °C to 20 °C. The time zero is defined as the time when the cell was installed in the thermostatic cell holder and the temperature of the inside solution became substantially the same temperature ( $< \pm 1$  °C) as the



target temperature at most 150 s; noted that it was less than 30 s to pass through the conformational change temperature. The  $[\theta']_{210}$  value for the rapid heating reaches the completely denatured value (red line) at most 200 s, indicating that the denature process observed by  $[\theta']_{210}$  is faster than or comparable to the heating rate.

On the other hand, gradual increase of  $[\theta']_{210}$  was observed for the rapid cooling process. The  $[\theta']_{210}$  data at 1800 s are still smaller than that for the ordered structure while they mostly reach the blue line for the as-prepared solution after 80,000 s (22 h) and 160,000 s (44 h) later, clearly indicating that the renature process observed by  $[\theta']_{210}$  is much slower than the cooling rate. Slightly smaller  $[\theta']_{210}$  at the longest  $t$  than the as-prepared sample can be attributed to the branching conformation (Matsuda et al., 2009) at which the xanthan chains cannot form the double helical structure properly. Similar tendency was also observed at  $\lambda_0 = 200$  nm for **X341k** and **X400k** in 10 mM aqueous NaCl (Fig. S4). The first-order plot, that is,  $\ln\{[\theta']_{210}(\infty) - [\theta']_{210}(t)\} / \{[\theta']_{210}(\infty) - [\theta']_{210}(0)\}$  vs  $t$ , assuming that  $[\theta']_{210}(t)$  at  $t = 44$  h is referred to as  $[\theta']_{210}(\infty)$ , is shown in Fig. S5. The data points follow a straight line while the intercepts at  $t = 0$  are appreciably smaller than zero, suggesting that the conformational change is faster at higher temperature, or a certain faster process exists in the disorder-order conformational change. We estimated the first-order rate constant  $k$  of this slow process to be  $6 \times 10^{-5} \text{ s}^{-1}$  and  $4 \times 10^{-5} \text{ s}^{-1}$  for **X341k** and **X400k**, respectively.



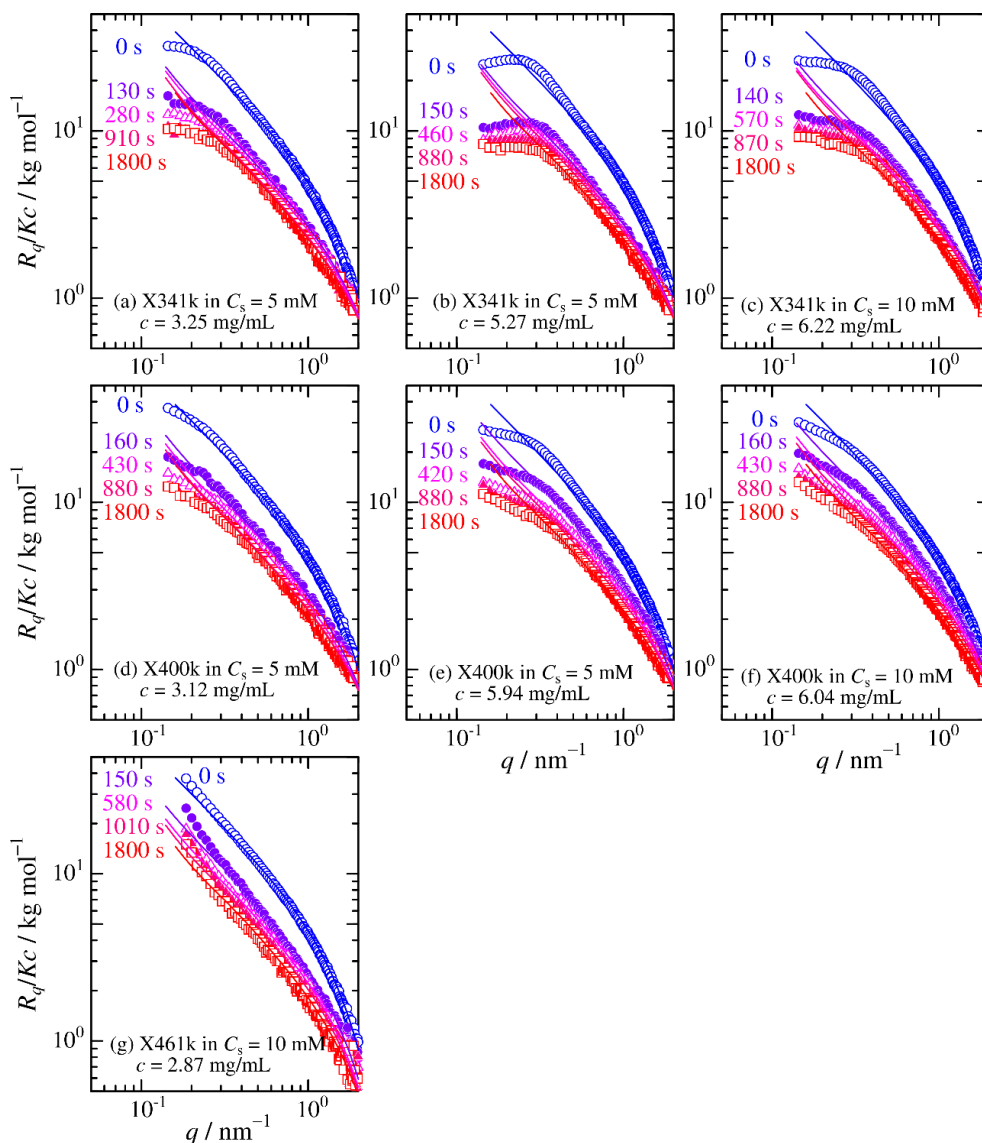
**Fig. 4.** Time course of  $[\theta']_{210}$  for (a) **X341k**, (b) **X400k**, and (c) **X461k** in aqueous NaCl. Left, abruptly heated from 20 to 80 °C. Right, abruptly cooled from 80 to 20 °C. Black unfilled circles, red filled circles, blue unfilled triangles, and blue filled triangles denote  $C_s = 5$  mM and  $c = 3$  mg

$\text{mL}^{-1}$ ,  $C_s = 5 \text{ mM}$  and  $c = 6 \text{ mg mL}^{-1}$ ,  $C_s = 10 \text{ mM}$  and  $c = 6 \text{ mg mL}^{-1}$ , and  $C_s = 10 \text{ mM}$  and  $c = 3 \text{ mg mL}^{-1}$ , respectively. Solid blue lines, the  $[\theta']_{210}$  value for the as-prepared solution at  $20 \text{ }^\circ\text{C}$ .

### 3.5. Time-dependent scattering intensities for xanthan solutions

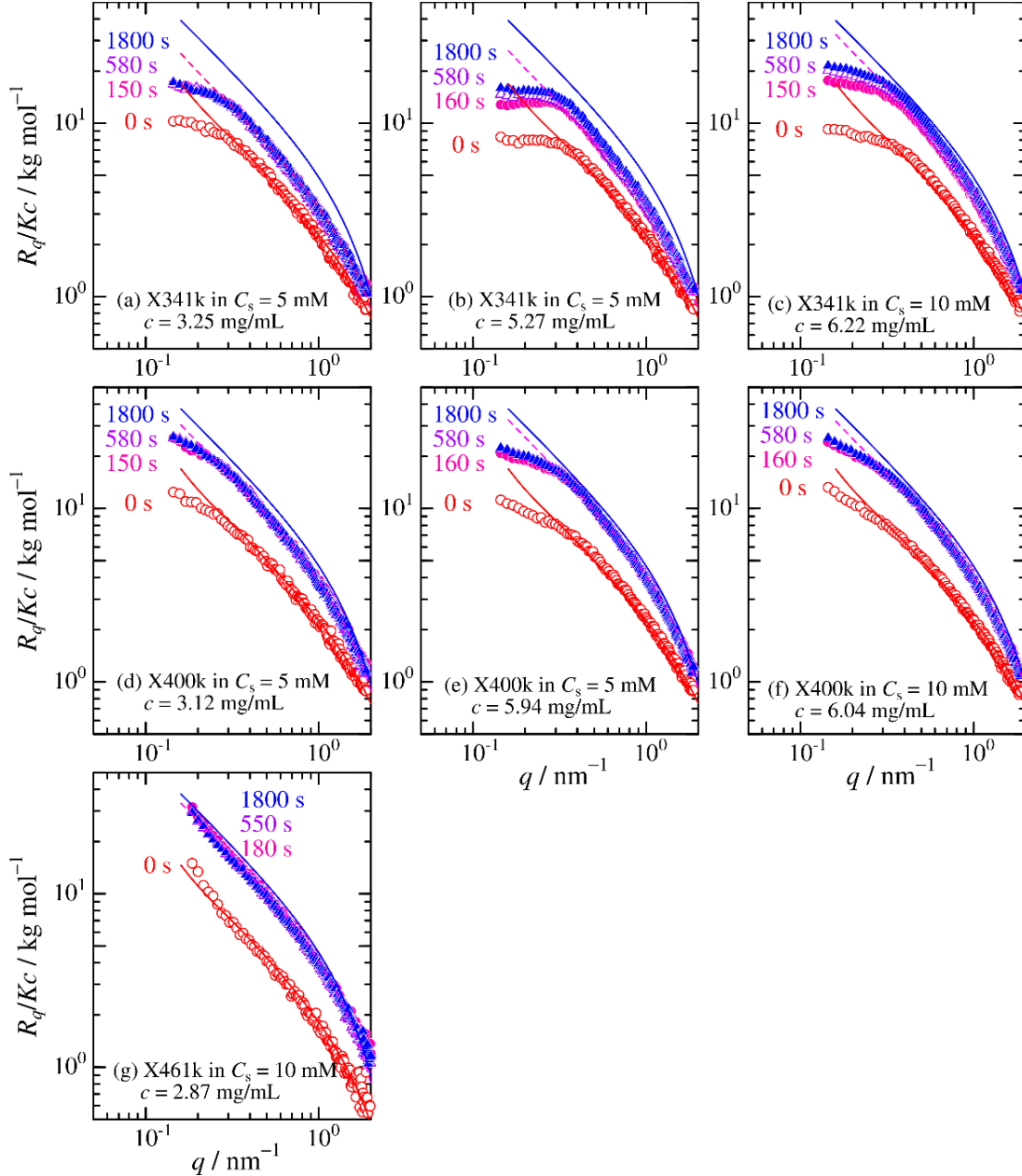
The reduced scattering intensity  $R_q/Kc$  is plotted against the magnitude  $q$  of the scattering vector in Fig. 5, in which doubly extrapolated value to  $c = 0$  and  $q = 0$  is the weight-average molar mass  $M_w$  of the solute. Taking into consideration that the  $qR_q/Kc$  value for the thin rod approaches the asymptotic value (W Burchard, 1994; Holtzer, 1955) of  $\pi M_L$  with  $M_L$  being the molar mass per unit contour length,  $R_q/Kc$  at the middle  $q$  region mainly reflects the  $M_L$  value which is the key factor to determine the single or double stranded structures; it is noted that the chain thickness may affect the  $R_q/Kc$  data at high  $q$  region and the structure factor  $S(q)$ , the chain stiffness ( $L_k$ ), and  $D$  influence the  $R_q/Kc$  only in the low  $q$  region. Indeed, we accurately determined the  $M_L$  value for semiflexible polysaccharides including amylose, cellulose (Jiang, Kitamura, Sato, & Terao, 2017) and single stranded schizophyllan (Tomofuji, Yoshiba, Christensen, & Terao, 2019) from the SAXS intensity in the middle  $q$  range.

Fig. 5 indicates the scattering data for xanthan samples in solution at  $80 \text{ }^\circ\text{C}$  at different  $t$  after the capillary cell was installed into the thermostatic cell holder. It takes similar time to attain the target temperature to that for our CD equipment ( $< 150 \text{ s}$ ). It should be noted that the data at  $t = 0$  were measured at  $20 \text{ }^\circ\text{C}$  without heat treatment. In contrast to the above-mentioned CD data, appreciable time dependent scattering intensity was observed after  $t = 150 \text{ s}$ , clearly indicating that the time course of the main-chain conformation observed by SAXS is different from the side-chain conformation from CD.



**Fig. 5.** Double logarithmic plots of  $R_q/Kc$  vs  $q$  for indicated samples and concentrations after abrupt heat treatment from 20 °C to 80 °C. Blue circles, data for native xanthan at 20 °C, referred to as  $t = 0$  s. The other data are those after abruptly heat to 80 °C. Blue and red curves for  $t = 0$  s and 1800 s are calculated in terms of the wormlike chain model with the parameters in Table 2. Other curves are the theoretical values calculated from eq 6 with  $\alpha$  in Fig. 7.

SAXS profiles for the rapid cooling process from 80 °C to 20 °C are shown in Fig. 6. The scattering intensities after cooling are quite larger than that at  $t = 0$  which was observed at 80 °C after the solution was kept at 80 °C at least 30 min. Interestingly, substantially the same scattering profiles were observed between 150 s and 1800 s, indicating that the conformational change from the disordered to ordered states is essentially faster than the other processes. This is completely opposite tendency comparing with those from CD.



**Fig. 6.** Double logarithmic plots of  $R_q/Kc$  vs  $q$  for indicated samples and concentrations after abrupt cool treatment from 80 °C to 20 °C. Red circles, data for completely denatured xanthan at 80 °C, referred to as  $t = 0$  s. The other data are the data after abruptly heat to 80 °C. Red and blue curves are the same as that in Fig. 5. Dashed curves are the theoretical values calculated from eq 6 with  $\alpha$  in Fig. 7.

### 3.6. Analysis in terms of the wormlike chain

To analyze the above-mentioned SAXS data, we utilized the touched-bead wormlike chain model. The reduced scattering intensity  $R_q/Kc$  can be written as (W. Burchard & Kajiwara, 1970; Yamakawa & Yoshizaki, 2016)

$$\frac{R_q}{Kc} = M_w P(q) S(q) \quad (3)$$

where

$$P(q) = 9 \left( \frac{2}{qd} \right)^6 \left( \sin \frac{qd}{2} - \frac{qd}{2} \cos \frac{qd}{2} \right)^2 P_{\text{thin}}(q) \quad (4)$$

Here,  $d$  is the bead diameter and  $P_{\text{thin}}(q)$  is the form factor of the thin wormlike chain, which is calculated with the characteristic function  $I(qL_k; u/L_k)$  by

$$P_{\text{thin}}(q) = \frac{2}{L^2} \int_0^L (L-u) I(qL_k; u/L_k) du \quad (5)$$

with  $L$  and  $L_k$  being the contour length and the Kuhn segment length (a measure of the chain stiffness), respectively. The integral was numerically calculated with the Nakamura-Norisuye expression (Nakamura & Norisuye, 2004, 2008) of  $I(qL_k; u/L_k)$ . Since  $qP_{\text{thin}}(q)$  reaches an asymptotic value of  $\pi/L_w$  with  $L_w$  being weight-average contour length at high  $q$  limit (Holtzer, 1955; Yamakawa & Yoshizaki, 2016) and  $S(q)$  can be regarded as unity except for the low- $q$  region,  $qR_q/Kc = qS(q)P_{\text{thin}}(q)M_w$  for the thin wormlike chain reaches  $\pi M_L$ ; note that  $M_L = M_w/L_w$ . Assuming previously estimated chain stiffness both for the ordered (Sato, Norisuye, et al., 1984a) and the disordered (Liu et al., 1987) conformations, that is,  $L_k = 240$  and  $42$  nm, respectively, and  $M_w$  values determined by SEC-MALS, the other two parameters,  $M_L$  and  $d$ , are unequivocally determined for equilibrium data designated as 0 s both in Figs. 5 and 6 by means of the curve fitting procedure (see also Figs S6 and S7 for the enlarged view of the middle  $q$  range). Substantially no concentration dependence was obtained for the two parameters,  $M_L$  and  $d$ , suggesting the assumption of  $S(q) = 1$  is reasonable. The evaluated  $M_L$  values for the ordered and disordered structures are consistent with the literature values (Matsuda et al., 2009). The parameters are summarized in Table 2. The theoretical values reproduce the experimental data at 0 s in Figs. 5 and 6 except for the low  $q$  region ( $q < 0.3 \text{ nm}^{-1}$ ) where  $S(q)$  may not be ignorable there because of the repulsive interactions between xanthan chains. Somewhat smaller  $R_q/Kc$  in the cooling process (Fig. 6) at finite  $t$  is likely because of the partly unfolded conformation. The discrepancy is more significant for **X341k** in 5 mM aqueous NaCl. This is reasonable because the lower  $M_w$  value in the solvent than those for the as-prepared solutions as shown in Table 1. We note that substantially the same theoretical curves will be obtained even if we assumed the half molar mass for the disordered state or quasi star-like conformation with short rod region at 80 °C reported previously (Liu et al., 1987) because the chain architecture affects the scattering intensity only in the low- $q$  region; the form factor of the wormlike star chain can be calculated by the Huber and Burchard expression (Huber & Burchard, 1989) as described in our recent report for semiflexible star polymers (Hasegawa, Nagata, Terao, & Suginome, 2017).

**Table 2.**

Wormlike chain parameters of the xanthan samples

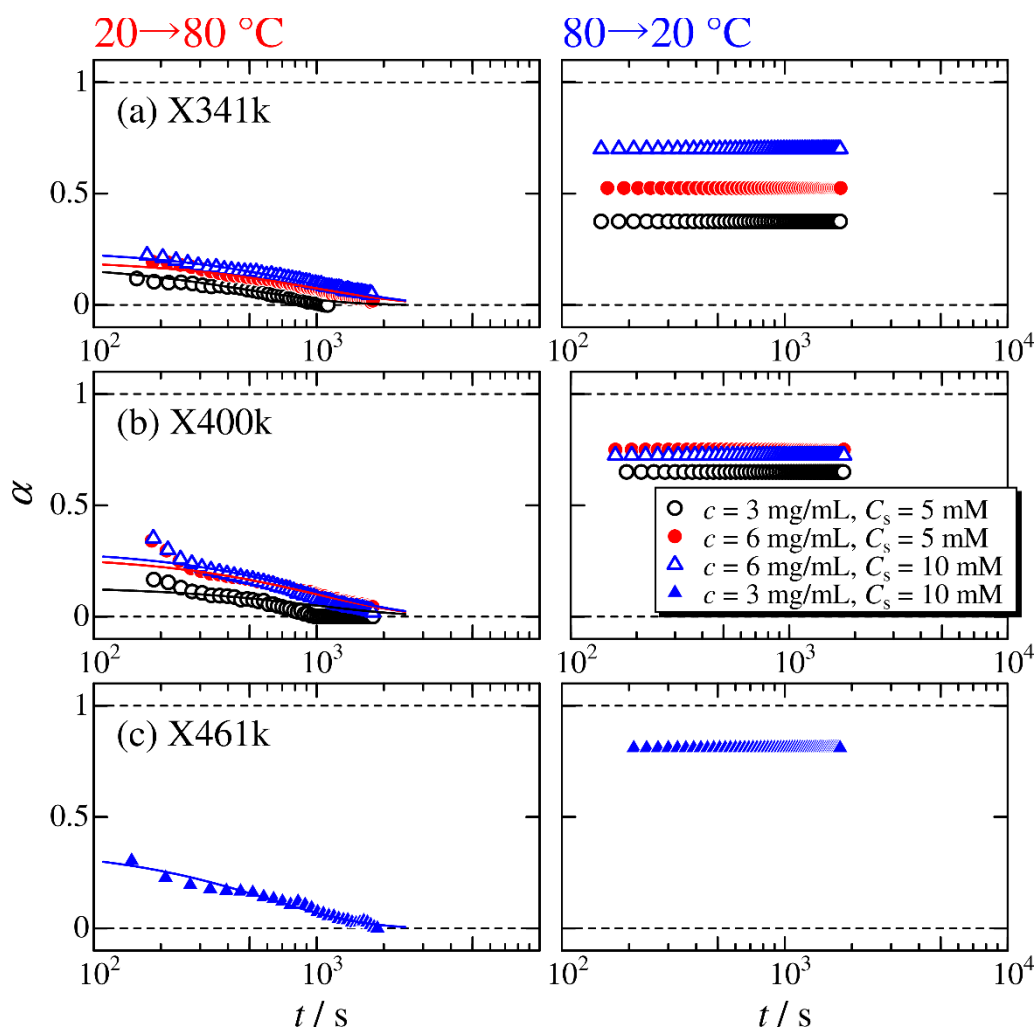
Sample	State	$M_w / \text{kg mol}^{-1}$	$L_K / \text{nm}$	$M_L / \text{kg mol}^{-1} \text{ nm}^{-1}$	$d / \text{nm}$
<b>X341k</b>	ordered <b>b</b>	341 <b>a</b>	240 <b>a</b>	$2.02 \pm 0.12$	$2.3 \pm 0.1$
	disordered <b>c</b>	341 <b>a</b>	42 <b>a</b>	$0.81 \pm 0.17$	$1.6 \pm 0.1$
<b>X400k</b>	ordered <b>b</b>	400 <b>a</b>	240 <b>a</b>	$1.99 \pm 0.05$	$2.3 \pm 0.1$
	disordered <b>c</b>	400 <b>a</b>	42 <b>a</b>	$0.81 \pm 0.03$	$1.6 \pm 0.1$
<b>X461k</b>	ordered <b>b</b>	461 <b>a</b>	240 <b>a</b>	$1.93 \pm 0.08$	$2.4 \pm 0.2$
	disordered <b>c</b>	461 <b>a</b>	42 <b>a</b>	$0.70 \pm 0.02$	$2.1 \pm 0.2$

**a:** Assumed, **b:** For as-prepared samples at 20 °C, **c:** kept at 80 °C for  $t = 1800$  s.

As mentioned above, scattering data in the heating process below  $t = 1000$  s reflects the mixture of the ordered and disordered conformations. When we assume the two-state model in which completely ordered (O) and disordered (D) chains,  $R_q/Kc$  can be written as

$$\frac{R_q}{Kc} = S(q)[\alpha M_{w,O} P_O(q) + (1 - \alpha) M_{w,D} P_D(q)] \quad (6)$$

where  $\alpha$  is the degree of dissociation and subscripts O and D denote the ordered and disordered states. Further assuming  $S(q) = 1$  and that the parameters for the ordered and disordered conformations are the same as those in Table 2 at 20 and 80 °C, respectively, the theoretical  $R_q/Kc$  values with an adjustable parameter  $\alpha$  fit the experimental data in substantially the same  $q$  range described above. The obtained  $\alpha$  values are plotted against  $t$  in Fig. 7. In the heating process,  $\alpha$  gradually decreases with increasing  $t$ . This may be regarded as the first order process taking into consideration that this is the unfolding process of the double helix. Indeed, the plots of  $\ln \alpha$  vs  $t$  (see Fig. S8) follow a straight line. Upward deviation in the initial time range implies a certain faster process for the order-disorder conformation change. The first-order reaction rate  $k$  estimated from the straight line in Fig. S8 ranges between  $1 \times 10^{-3}$  and  $2 \times 10^{-3} \text{ s}^{-1}$  for all the samples. On the contrary, no time dependence was observed for the cooling process, indicating that double stranded conformation is recovered quickly. The  $\alpha$  values are however smaller than unity for the renatured samples. This is most likely because the complete double helical conformation cannot be formed by the hairpin like conformation or the branching aggregates as in the case of the CD data or partly unfolded conformation due to low ionic strength.

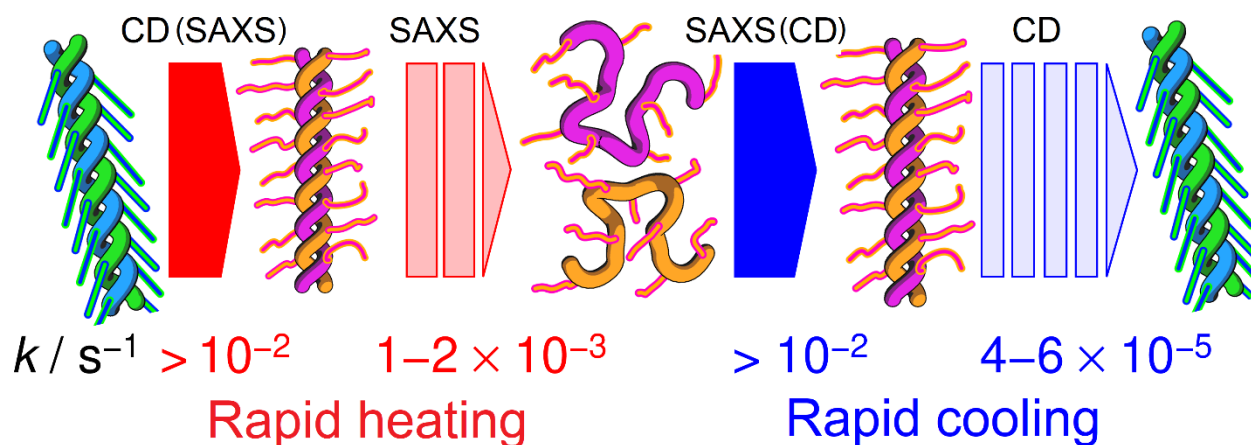


**Fig. 8.** Time course of  $\alpha$  for (a) X341k, (b) X400k, and (c) X461k in aqueous NaCl. Left, abruptly heated from 20 to 80 °C. Right, abruptly cooled from 80 to 20 °C. Black unfilled circles, red filled circles, blue unfilled triangles, and blue filled triangles denote  $C_s = 5$  mM and  $c = 3$  mg mL<sup>-1</sup>,  $C_s = 5$  mM and  $c = 5 - 6$  mg mL<sup>-1</sup>,  $C_s = 10$  mM and  $c = 6$  mg mL<sup>-1</sup>, and  $C_s = 10$  mM and  $c = 3$  mg mL<sup>-1</sup>, respectively. Solid curves indicate the first order reaction with the same rate constant shown in Fig. S8.

### 3.7. Kinetics of thermal conformational change

The current analysis of the time-resolved CD and SAXS measurements clearly showed that both denature and renature processes consist of at least two steps, and hence the existence of some intermediate species as illustrated in Fig. 8. If we consider the CD signals mainly reflect the chiral order of the carbonyl groups on the side chains of xanthan c.f. Fig. 1, the initial change after the rapid heating is mainly the melting of the side chain conformation, while the double helical structure remains. This can be regarded as loose double helix. It takes 30 min to attain the complete disordered state with the first-order reaction rate of  $1 - 2 \times 10^{-3}$  s<sup>-1</sup>. On the other hand, formation of the double helical structure is faster than the temperature changing (< 150 s) after rapid cooling. This double helix may be somewhat looser than the native state because the CD signal was still changing at  $t = 150$  s. The reaction to attain the native double helix obey the first

order rate constant of  $4 - 6 \times 10^{-5} \text{ s}^{-1}$ , then more than 1 day was needed to recover similar CD signal as that for the as-prepared solution. According to the previous reports (Morris, 2019; Norton et al., 1984), the molar mass of xanthan recorded by light scattering with raising temperature decreased at lower temperature than those for the reduction of helix content estimated from optical rotation, suggesting a single helix conformation as an intermediate state. On the one hand, in the case of rapid temperature change of the current study, we found that the quick formation of the (loose) double helical structure, which may also be important to understand the temperature and solvent dependent conformation and aggregation of xanthan.



**Fig. 8.** Schematic representation of denaturation and renaturation processes of xanthan in aqueous NaCl.

#### 4. Conclusions

Time-resolved measurement of CD and SAXS were made for the three xanthan samples ranging in  $M_w$  between 300 and 500  $\text{kg mol}^{-1}$  in aqueous NaCl at  $C_s = 5 \text{ mM}$  and  $10 \text{ mM}$ . Appreciably different kinetics was found between the ellipticity and the scattering profile in the middle  $q$  region, clearly indicating the intermediate species, loose double stranded conformation, in which the side chain is still disordered. This loose double helix was found both after the rapid heating and cooling processes. The current finding may be applicable for the higher  $M_w$  xanthan samples whereas the SAXS measurement becomes more difficult owing to higher melting temperature in aqueous NaCl solutions. The variety of higher order structure of xanthan reported in the literatures (Capron et al., 1997; Matsuda et al., 2009) may be related to the kinetics. In other words, not only static properties including temperature, concentration, chain length, but also kinetic properties such as temperature changing rate may play an important role for the higher order structure and rheological properties of renatured xanthan. The current method can be applied for the other multiple helical polysaccharides (Meng, Lyu, Xu, & Zhang, 2020) to clarify the kinetic behavior to form the higher order structures.

#### Declaration of Competing Interest

The authors reported no declarations of interest.

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with the approval of the Japan Synchrotron Radiation Research Institute (JASRI) (Proposal Nos. 2017B1062, 2018B1088, 2019A1072, and 2019B1113). CD spectroscopy was carried out with the approval of the Hiroshima Synchrotron Radiation Center (HiSOR) of Hiroshima University (Proposal No. 19BU011). This work was partly supported by JSPS KAKENHI Grant Numbers JP17K05884, JP18H02020, and JP20H02788.

## Appendix A. Supplementary data

Supplementary material related to this article can be found in the online version, at doi:

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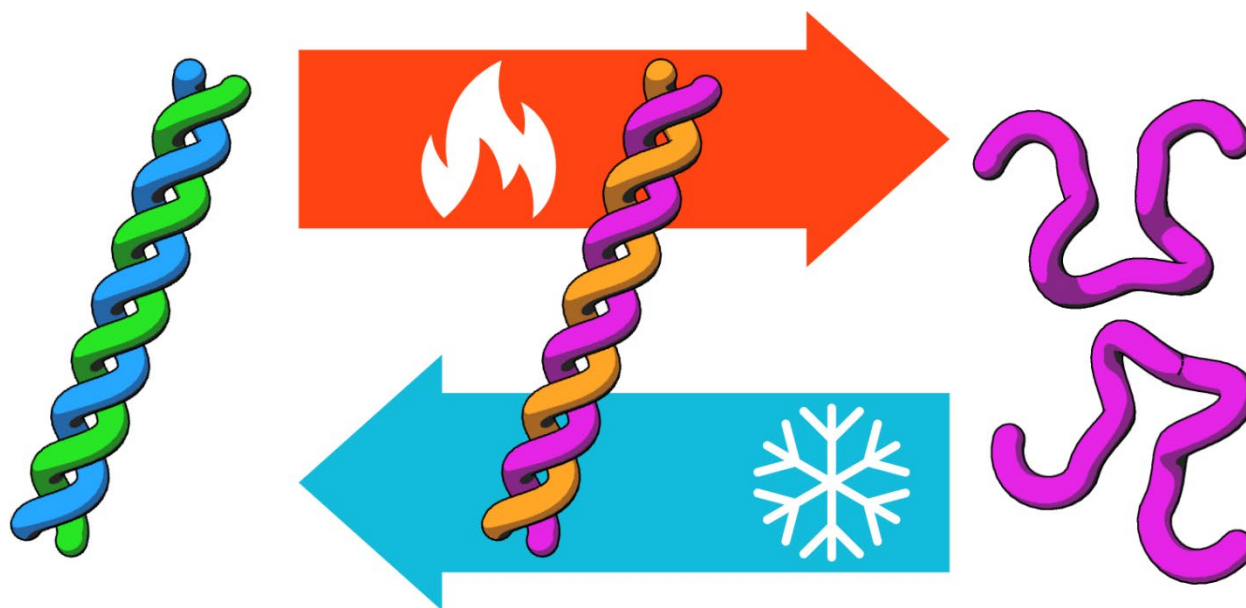
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### Graphical abstract

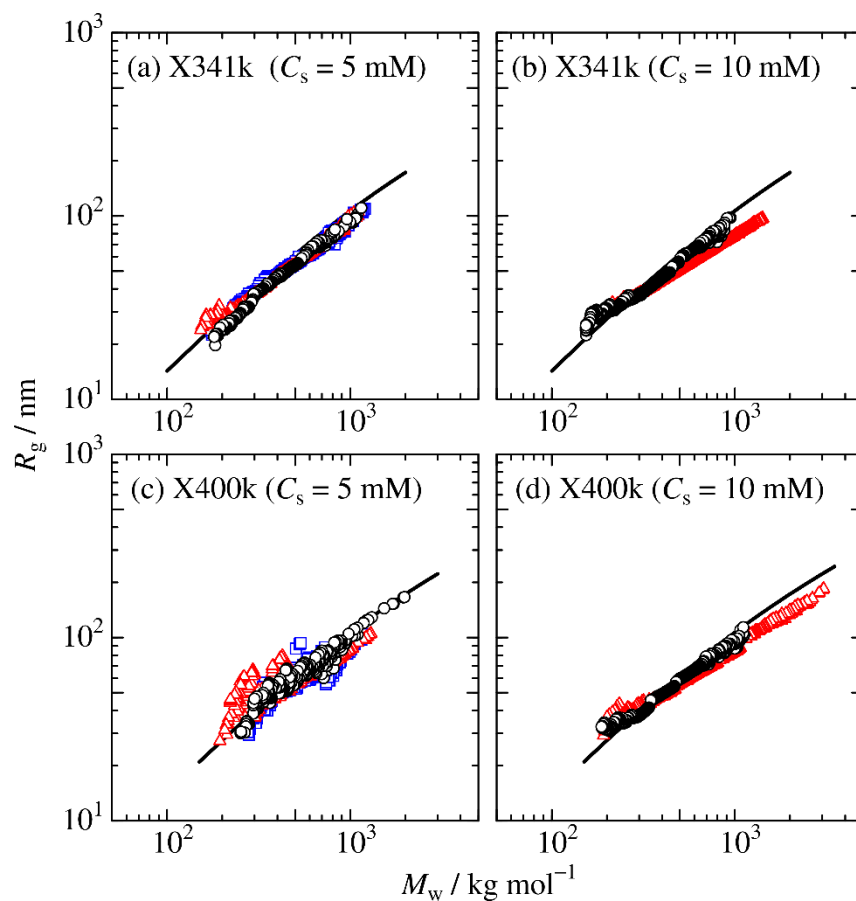


## Kinetics of Denaturation and Renaturation Processes of Double-Stranded Helical Polysaccharide, Xanthan in Aqueous Sodium Chloride

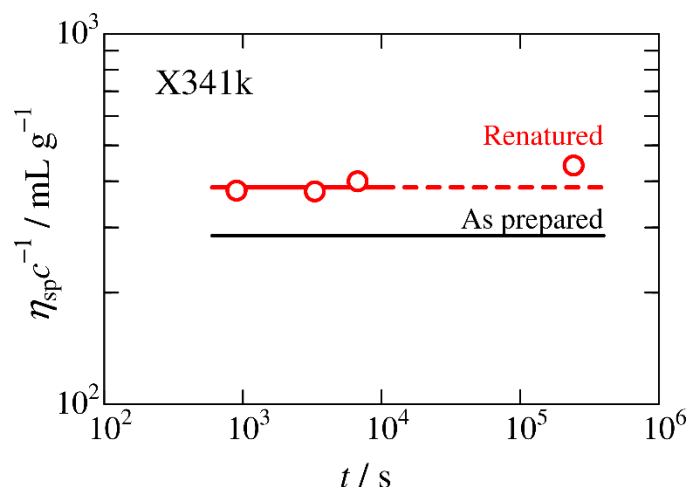
Yu Tomofuji<sup>a</sup>, Koichi Matsuo<sup>b</sup>, and Ken Terao<sup>a,\*</sup>

<sup>a</sup>Department of Macromolecular Science, Graduate School of Science, Osaka University, 1-1, Machikaneyama-cho, Toyonaka, Osaka 560-0043, Japan.

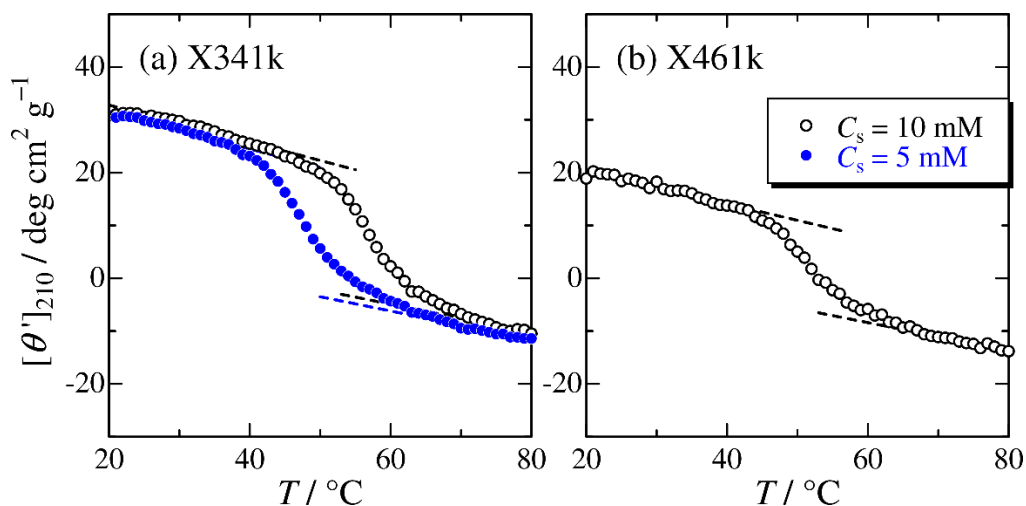
<sup>b</sup>Hiroshima Synchrotron Radiation Center, Hiroshima University, Kagamiyama, Higashihiroshima, Hiroshima 739-0046, Japan.



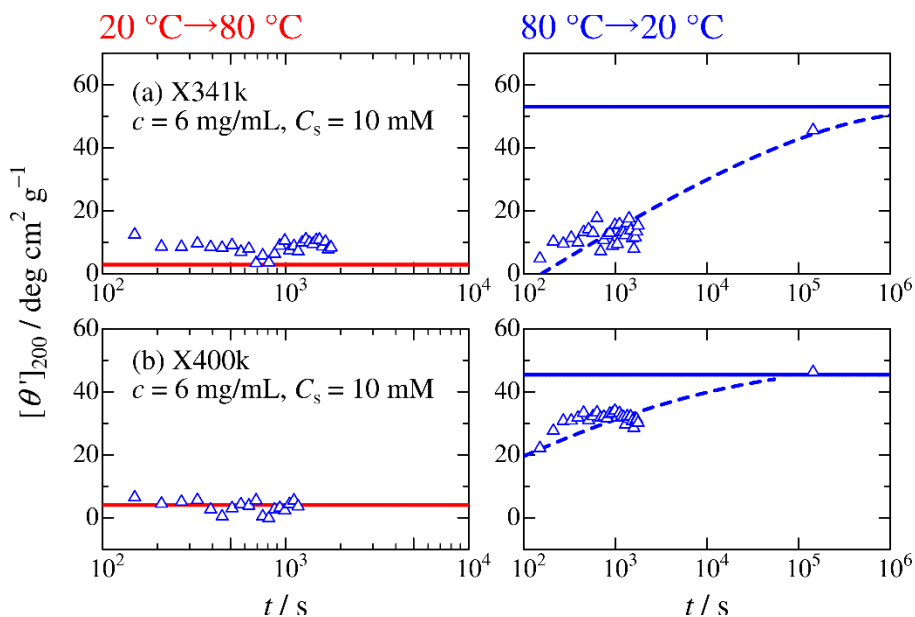
**Fig. S1.** Weight-average molar mass  $M_w$  dependence of the radius of gyration  $R_g$  for (a) X341k at  $C_s = 5 \text{ mM}$ , (b) X341k at  $C_s = 10 \text{ mM}$ , (c) X400k at  $C_s = 5 \text{ mM}$ , and (d) X400k at  $C_s = 10 \text{ mM}$ . Black circles, blue squares, and red triangles indicate the data for the as-prepared solution, the renatured xanthan sample with  $c_d = 3.0 - 3.1 \text{ mg mL}^{-1}$ , and those with  $c_d = 5.4 - 6.4 \text{ mg mL}^{-1}$ . Solid lines indicate the literature values.<sup>1</sup>



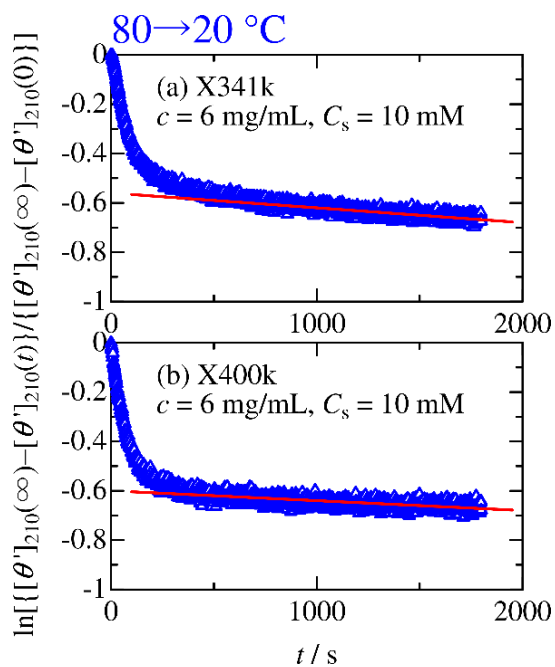
**Fig. S2.** Specific viscosity  $\eta_{sp}$  plotted against the time  $t$  after quenching for **X341k** in 10 mM aqueous NaCl at 20 °C. The denatured ( $c_d$ ) and measured ( $c$ ) concentration were 6 mg/mL and 1 mg/mL, respectively.



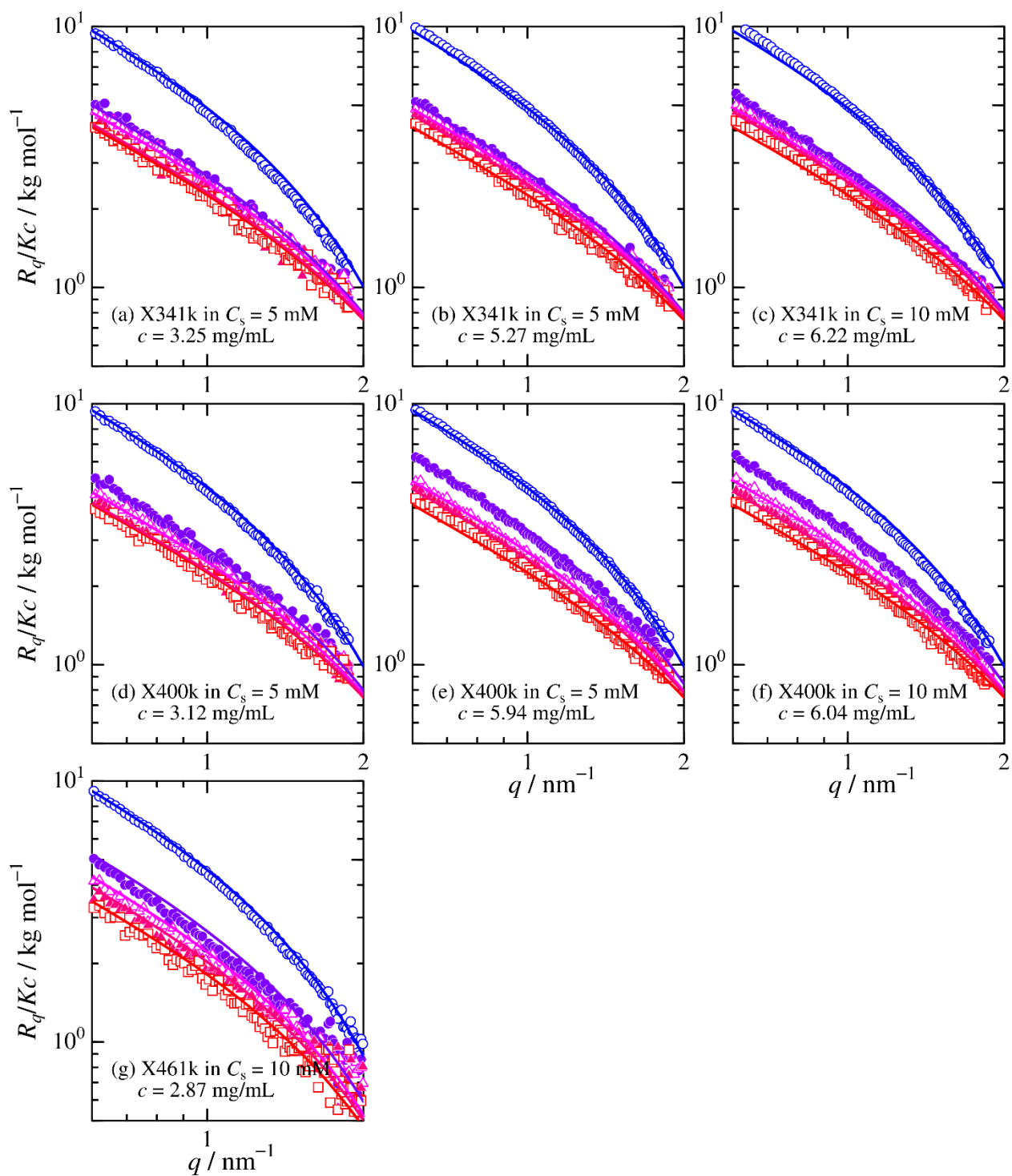
**Fig. S3.** Temperature dependence of  $[\theta]_{210}$  for (a) **X341k** and (b) **X461k** in aqueous NaCl and at  $\lambda_0 = 210$  nm at indicated  $C_s$ . Temperature-raising rate and  $c$  were 5 °C min<sup>-1</sup> and 3 mg mL<sup>-1</sup>, respectively.



**Fig. S4.** Time course of  $[\theta']_{200}$  for (a) **X341k** and (b) **X400k** in 10 mM aqueous NaCl. Left, abruptly heated from 20 to 80 °C. Right, abruptly cooled from 80 to 20 °C. Solid blue lines, the  $[\theta']_{200}$  value for the as-prepared solution at 20 °C.

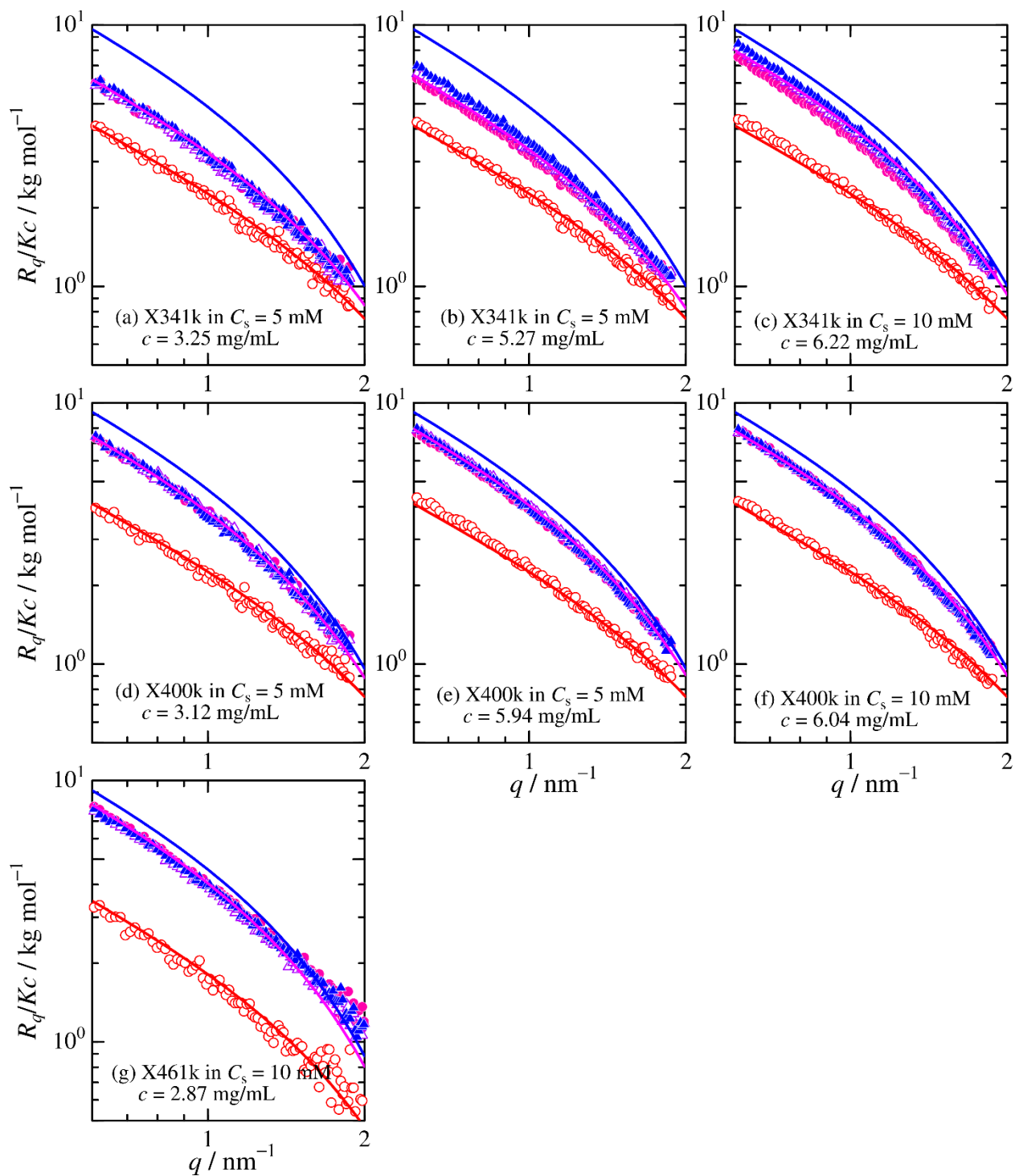


**Fig. S5.** Plots of  $\ln\left[\frac{[\theta']_{210}(\infty) - [\theta']_{210}(t)}{[\theta']_{210}(\infty) - [\theta']_{210}(0)}\right]$  vs  $t$  for (a) **X341k** and (b) **X400k** in 10 mM aqueous NaCl. The first-order rate constant was estimated from the solid red line.

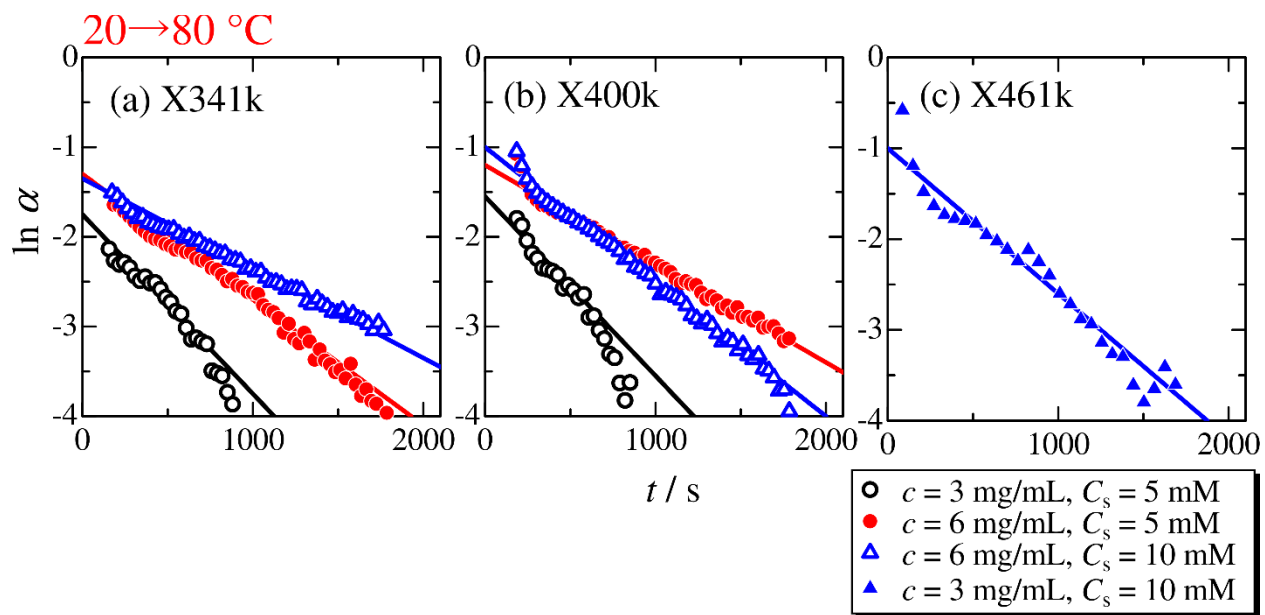


**Fig. S6.** Enlarge of the middle  $q$  region of Fig. 5.





**Fig. S7.** Enlarge of the middle  $q$  region of Fig. 6.



**Fig. S8.** Plots of  $\ln \alpha$  vs  $t$  for (a) X341k, (b) X400k, and (c) X461k in aqueous NaCl. The first-order rate constant was estimated from the solid lines.

1. Sato, T.; Norisuye, T.; Fujita, H., Double-Stranded Helix of Xanthan - Dimensional and Hydrodynamic Properties in 0.1-M Aqueous Sodium-Chloride. *Macromolecules* **1984**, *17* (12), 2696-2700.