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Article

Magnetoelectric Induced Switching of Perpendicular Exchange Bias Using 30-nm-Thick Cr_2O_3 Thin Film

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Abstract: Magnetoelectric (ME) effect is a result of the interplay between magnetism and electric field and now, it is regarded as a principle that can be applied to the technique of controlling the antiferromagnetic (AFM) domain state. The ME-controlled AFM domain state can be read out by the magnetization of the adjacent ferromagnetic layer coupled with the ME AFM layer via exchange bias. In this technique, the reduction in the ME layer thickness is an ongoing challenge. In this paper, we demonstrate the ME-induced switching of exchange bias polarity using the 30-nm thick ME Cr_2O_3 thin film. Two typical switching processes, the ME field cooling (MEFC) and isothermal modes, are both explored. The required ME field for the switching in the MEFC mode suggests that the ME susceptibility (α_{33}) is not deteriorated at 30 nm thickness regime. The isothermal change of the exchange bias shows the hysteresis with respect to the electric field, and there is an asymmetry of the switching field depending on the switching direction. The quantitative analysis of this asymmetry yields α_{33} at 273 K of $3.7 \pm 0.5 \text{ ps/m}$, which is comparable to the reported value for the bulk Cr_2O_3 .



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1. Introduction

Controlling antiferromagnetic (AFM) domain has been an active pursuit because of the possible applications, such as ultrahigh density storage and THz devices. The difficulty is mainly in the control and detection of the AFM domain state because no net magnetization emerges from AFM materials. So far, some techniques to control the AFM domain state have been proposed, such as spin orbit torque [1]. The magnetoelectric (ME) effect, an induction of magnetization (M) by an electric field (E) or an induction of electric polarization (P) by a magnetic field (H), is also one root. The ME effect appears in some insulating antiferromagnets as a result of the simultaneous breakings of time- and spatial-inversion symmetries. The strength of the ME effect is quantified by the ME susceptibility α ($= dM/dE = dP/dH$). The linear ME effect was experimentally observed in the bulk Cr_2O_3 crystal in the early of 1960s [2,3]. In 1966, Martin and Anderson revealed, based on the symmetrical argument, that the sign of α depends on the orientation of Néel vector and the AFM domain state was consequently controllable [4]. Now, the linear ME effect of Cr_2O_3 has been recognized renewably as the ferroic feature in the presence of the finite E or H [5], and it was confirmed by the Cr_2O_3 thin film [6].

For the detection of the AFM domain state, the usage of exchange bias at the ferromagnetic (FM)/AFM interface [7] or the anomalous Hall effect (AHE) in the heavy metal such as Pt on AFM [8] has been proposed. The device architecture based on the former scenario was proposed by Chen et al., as ME-random access memory (ME-RAM) [9]. The former scenario is based on the fact that the ME-controlled AFM domain state is detectable via the FM magnetization with the assumption that the exchange bias polarity is coupled with the AFM domain state. The detection of ME-controlled Cr_2O_3 domain state was first demonstrated using the bulk Cr_2O_3 substrate [7] and it has been developed to all-thin-film

system with the Cr_2O_3 layer [10,11]. Notably, the above prerequisite has also been proven experimentally by means of the element-specific magnetic domain observation [12]. In this approach, the large output signal is expected so that the output voltage is determined by the FM magnetization direction. Another approach based on the AHE of the heavy metal has the benefit that the stacking structure is very simple and the switching energy can be reduced compared with the former scenario because the interfacial exchange coupling with the FM spin is absent. Instead, the output voltage could be small. At the first stage for both attempts, the Cr_2O_3 thickness is high, typically above 200 nm [10,11,13–15], and the reduction in the Cr_2O_3 thickness is an ongoing demand. Until now, the switching of the exchange bias polarity was realized using the 50-nm thick Cr_2O_3 [16] and the AHE detection was confirmed down to 20-nm thick Cr_2O_3 regime [17]. In particular, the former scheme has a prerequisite that the exchange bias has to maintain the low Cr_2O_3 thickness. It was reported that the critical thickness of the appearance of the exchange bias was relevant to the AFM domain wall width [18]. For the case of Cr_2O_3 , the AFM domain wall width is reported to be 20–60 nm depending on the lattice deformation [19]. Hence, it is important to investigate the applicable thickness of former scheme below 50 nm. In this paper, we explored the reduction in Cr_2O_3 thickness in the former approach and demonstrate the ME-induced switching of the exchange bias polarity using the 30-nm thick Cr_2O_3 layer.

There are two typical ME field application processes: ME field cooling (MEFC) and isothermal switching. In the previous reports, mainly the ME field cooling (MEFC) process was done. The isothermal mode was not as much because of the difficulty despite its importance for practical use. This is partly because for the isothermal switching, the high dielectric resistance is required because of the high required ME energy compared with the MEFC mode [15,20]. In this paper, we present both types of switching. Based on the required field condition, we show that the ME susceptibility (α) is not deteriorated in the 30-nm-thickness regime.

2. Materials and Methods

Pt 2 nm/Co 0.25 nm/Au 1.0 nm/ Cr_2O_3 30 nm/Pt 20 nm stacked film was prepared on an $\alpha\text{-Al}_2\text{O}_3(0001)$ substrate. The film preparation was done by the DC magnetron sputtering system with the base pressure below 1×10^{-6} Pa. The 20-nm-thick Pt layer was deposited at 873 K on the ultrasonically cleaned substrate as a buffer layer to align the crystallographic orientation of the Cr_2O_3 layer. The Pt-buffer layer also works as the bottom electrode to apply E to the Cr_2O_3 layer. The Cr_2O_3 was formed by sputtering of a pure Cr target in $\text{Ar} + \text{O}_2$ gas mixture at the substrate temperature of 773 K. The 1.0-nm thick Au layer was used to tune the strength of the interfacial exchange coupling J_{INT} between Co and Cr_2O_3 [21]. Unless the suitable spacer layer was inserted, the exchange bias cannot be maintained in the temperature regime where the ME susceptibility is high [15]. The Co and Pt top layers were deposited at room temperature. The 2-nm thick Pt layer prevents the oxidization of ultrathin FM Co layer and also acts as the induction of the perpendicular magnetic anisotropy. The crystallographic orientation of each layer was characterized by using a reflection high-energy electron diffraction (RHEED). The RHEED chamber is directly connected to the sputtering chamber, and hence the RHEED observations could be done without exposing the sample to air. As shown in Figure 1, the RHEED pattern on the Cr_2O_3 layer is streaky, which shows the flat surface. The diffraction pattern indicates that the Cr_2O_3 layer grows with the c -axis along the growth direction and that the twin boundary is included along the [1120] direction.

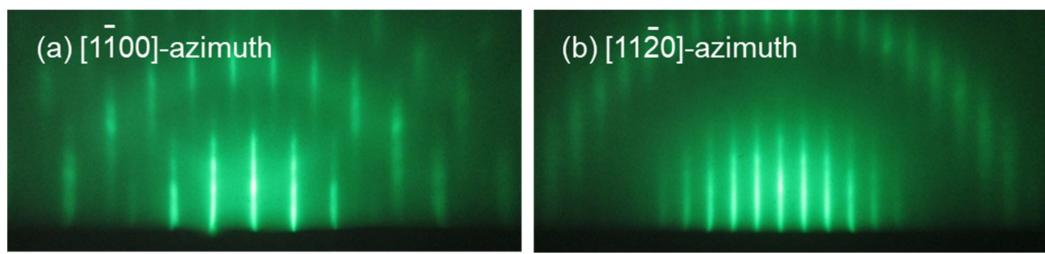


Figure 1. Reflection high-energy electron diffraction RHEED images of Cr_2O_3 layer with (a) $[1\bar{1}00]$ and (b) $[11\bar{2}0]$ azimuth.

X-ray reflection (XRR) measurement was carried out to confirm the well-defined stacking structure. The XRR profile shown in Figure 2a shows the clear oscillation above $2\theta/\omega$ of 14° , which proves the sharp interfaces and the well-defined stacking structure. The FFT analysis of the oscillation gives the actual Cr_2O_3 thickness as 31.6 nm. In the high-angle X-ray diffraction (XRD) profile shown in Figure 2b, any diffraction peaks other than $\text{Pt}(111)$ and $\alpha\text{-Al}_2\text{O}_3(0001)$ substrate were not observed. Many Laue fringes were observed around the $\text{Pt}(111)$ diffraction peaks (Figure 2c), which indicate the good crystalline quality of the Pt buffer layer. It should be noted that the diffraction peaks from the Cr_2O_3 layer are overlapped with those from the $\text{Pt}(111)/\text{Pt}(222)$ diffraction peaks. The diffractions from the Cr_2O_3 layer could be masked because of the larger atomic scattering factor of Pt compared with Cr^{3+} and O^{2-} .

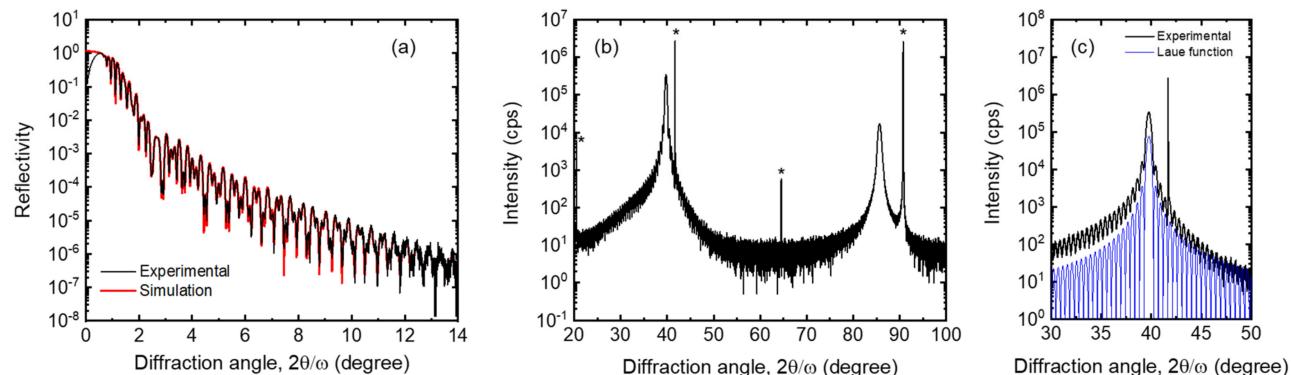


Figure 2. (a) X-ray reflection (XRR), (b) high-angle $2\theta/\omega$ profile of the film. (c) represents the enlarged profile of (b) around $2\theta/\omega = 30\text{--}50^\circ$. Black and red lines in (a) represent the experimental result and the fitted result, respectively. Blue line in (c) represents the calculated profile using the Laue function.

Magnetic characterizations were carried out mainly by means of the AHE measurements after applying the ME field. For the AHE measurements, the film was patterned into the microdot with $15\text{ }\mu\text{m}$ diameter. On top of the microdot, the four top electrode (Cr/Au) was prepared by the lift-off technique. The optical microscope image with the electrical circuit is shown in Figure 3a. In our sample structure, the current flows in the $\text{Pt}/\text{Co}/\text{Au}$ layer wherein the AHE detects the Co magnetization perpendicular to the film plane. This is suitable in this work because the studied film has the perpendicular magnetic anisotropy. In this work, the AHE loops were measured as a function of H applied to the direction perpendicular to the film. From the AHE loop, the exchange bias field, H_{ex} , a shift of the magnetization curve along the H axis was evaluated. H_{ex} was estimated as $\Delta H_{\text{C}}/2$ where H_{C} is the switching field (coercivity) for up-to-down and down-to-up switching of the FM magnetization. E was applied between top ($\text{Pt}/\text{Co}/\text{Au}$) and bottom (Pt-buffer layer) electrodes, i.e., the direction perpendicular to film. The positive directions of H and E are defined as the direction from bottom (substrate side) to top (film side) of the film.

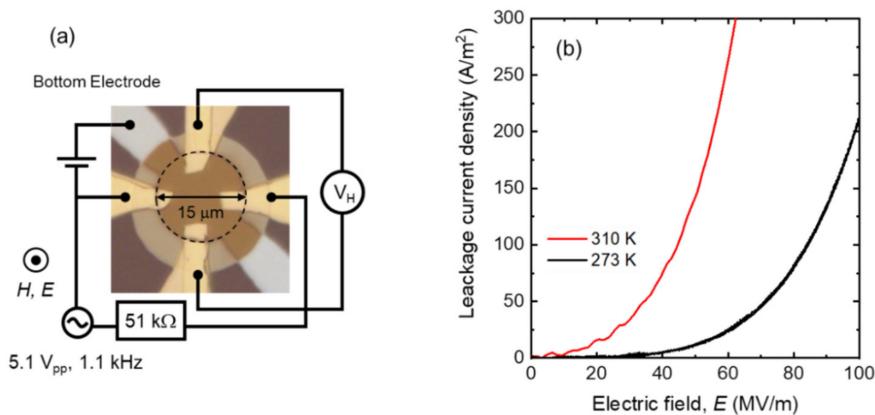


Figure 3. (a) Optical microscope image of the device used for the test and the electric circuitry. (b) *I*-*V* curves of the Cr₂O₃ layer measured at 310 K (red) and 273 K (black).

Here, we denote the ME field as the simultaneously applied H and E because the energy gain by the ME effect ΔF is expressed as

$$\Delta F = \alpha_{ij} \cdot E_i H_j \quad (1)$$

Together with the crystallographic characterization of Cr₂O₃, the ME susceptibility evaluated in this paper is α_{33} . As mentioned above, there are two typical processes to apply to the ME field: MEFC and isothermal modes. In the MEFC mode, the sample was once heated to above the Néel temperature of Cr₂O₃ (~307 K for bulk Cr₂O₃ [22]), 310 K. Then, H_{MEFC} ($\mu_0 H_{\text{MEFC}} = 5, 6, 7$ T) and E (0–47 MV/m) were applied. In maintaining both fields, the sample was cooled to the AHE measurement temperature, 250 K at which the exchange bias polarity was checked. After measuring the AHE loop, the above heating and cooling processes were repeated with the different ME field condition. More details of the MEFC process can be found in [7,12].

In the isothermal process, the temperature kept constant during the ME field application and the AHE measurements. In this process, the sample was cooled from 310 K to 273 K under the H ($\mu_0 H = 0.6$ T) application during the cooling: the conventional field-cooling (FC) to induce the exchange bias. At the constant temperature (273 K), $H_{\text{isothermal}}$ ($\mu_0 H_{\text{isothermal}} = 6$ T) and E (−60 – +100 MV/m) were simultaneously applied typically for 30 s. After that, E was removed and the AHE as a function of H measured. In the isothermal process, the sequential ME field application is important so that the exchange bias switching occurs accompanied by the hysteresis. In this work, the ME fields were applied in the following sequences.

- ME field application: for example, $\mu_0 H_{\text{isothermal}} = 6$ T and $E = +60$ MV/m;
- Removing E and AHE measurement as a function of $\mu_0 H$;
- ME field application: for example, $\mu_0 H_{\text{isothermal}} = 6$ T and $E = +80$ MV/m;
- Removing E and AHE measurement as a function of $\mu_0 H$;

These processes were repeated with increasing E until the exchange bias polarity is fully switched. Finally, after the exchange bias polarity was switched, the sign of the ME field was reversed, and the similar processes were repeated with the negative E .

- ME field application: for example, $\mu_0 H_{\text{isothermal}} = 6$ T and $E = 0$ MV/m;
- Removing E and AHE measurement as a function of $\mu_0 H$;
- ME field application, for example, $\mu_0 H_{\text{isothermal}} = 6$ T and $E = -18$ MV/m;
- Removing E and AHE measurement as a function of $\mu_0 H$;
- ME field application: for example, $\mu_0 H_{\text{isothermal}} = 6$ T and $E = -23$ MV/m;
- Removing E and AHE measurement as a function of $\mu_0 H$;

These processes were repeated with decreasing E until the exchange bias polarity is fully switched again. Details of the isothermal switching protocol can be found in

our previous report [20]. Note that for both processes, the highest μ_0H during the AHE loops was ± 700 mT, which is low enough to switch the exchange bias polarity by H alone ($|\mu_0H| > 8$ T) [23].

3. Results and Discussions

Prior to showing the results on the ME-induced switching, it is helpful to show the electric resistance of the Cr_2O_3 layer. Using the same device used for the AHE measurements, the electric resistance was measured. Typical I - V curves measured at 310 K and 273 K are shown in Figure 3b. The I - V curves show the non-linear increase with respect to the voltage as is generally observed in an insulator thin film. Even at the most severe conditions adopted in this work (at highest temperature and at the highest E adopted), the current density is in the range of 10^2 A/m², which is low enough compared with the current-induced magnetization switching, such as the spin-orbit torque mechanism, typically above 10^9 A/m² [24,25].

First, we show the switching of the exchange bias polarity by the MEFC process. Figure 4a shows the series of AHE loops measured after the MEFC with $\mu_0H_{\text{MEFC}} = 7$ T. When the electric field was not applied, e.g., the conventional FC, the negative exchange bias of $\mu_0H_{\text{ex}} = -108$ mT appears (black curve). The similar AHE loops were obtained for E below 16 MV/m (blue curve). With increasing E , the step at about +100 mT starts to be observed in the AHE loop, a signature of the appearance of positive exchange bias. The two-step AHE loops are observed in the E range of 24–43 MV/m. The two-step magnetization curve is attributed to the co-existence of the positive and negative exchange biased domains. The similar two-step exchange biased state was observed in the previously reported ME-induced switching [16] and the magnetization curve after zero-field cooling [26]. The positive exchange bias grows with increasing E , the step at about -100 mT suppresses and instead, that at about +100 mT enhances. Finally, above $E = 47$ MV/m, the AHE loop shows the full positive exchange bias. In Figure 4b, the change in μ_0H_{ex} as a function of E during the MEFC is shown. We find that the change in the exchange bias is common to every μ_0H_{MEFC} . It should be noted that the AHE loops show the tiny hysteresis, i.e., the low coercivity. This is attributed to the usage of the Au spacer layer instead of the Pt spacer layer as in [21]. The Au spacer layer is suitable to tune the interfacial magnetic anisotropy in maintaining high exchange bias with suppressing the coercivity enhancement. The details of the role on the spacer layer can be found in our previous paper [21].

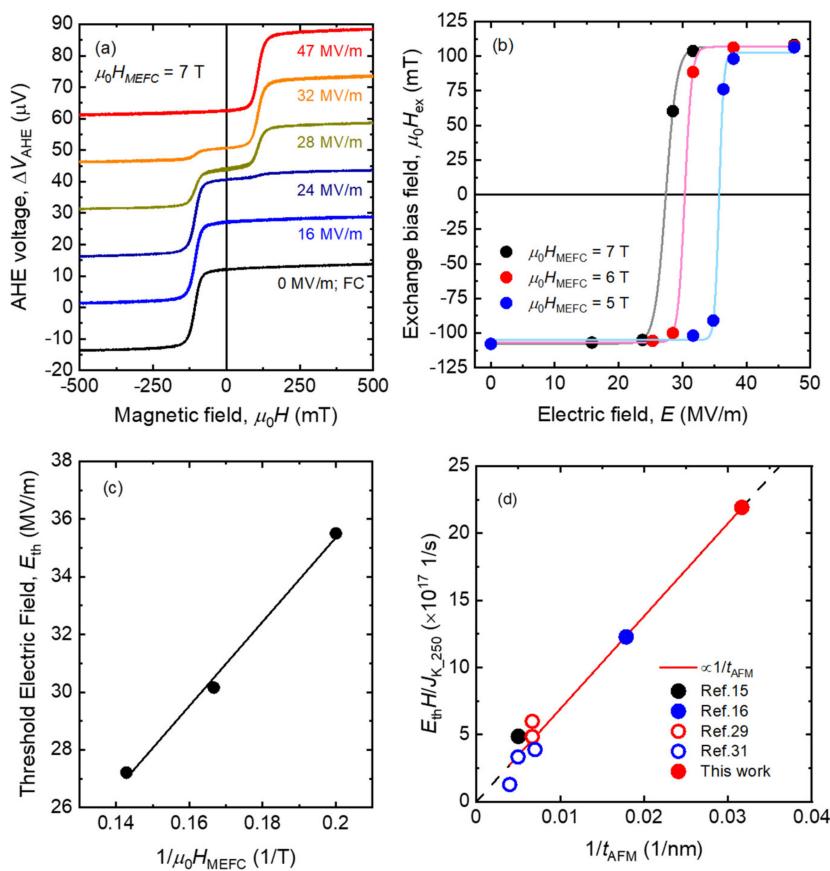


Figure 4. (a) Anomalous Hall effect (AHE) loops as a function of $\mu_0 H$ after the magnetoelectric field cooling (MEFC). The E value for the MEFC are 0 (black), 16 MV/m (blue), 24 MV/m (dark blue), 28 MV/m (yellow), 32 MV/m (orange), and 47 MV/m (red). For every case, $\mu_0 H_{MEFC}$ was 7 T. (b) Change in the exchange bias field with E . Black, red and blue point correspond to the case of $\mu_0 H_{MEFC} = 7$ T, 6 T and 5 T. Lines represent the fitted results using Equations (2) and (3). (c) Change in the threshold E with $1/\mu_0 H_{MEFC}$. Line represents the linearly fitted result. (d) $1/t_{AFM}$ dependence of the required EH product to switch the exchange bias. The EH values are normalized by the exchange anisotropy energy density at 250 K (see text).

The growth of the positive exchange-biased state by E is understood by the energy competition between J_{INT} and the energy gain by the ME effect (Equation (1)) [27]. The former is caused by the interfacial exchange coupling between FM (Co) and interfacial AFM (Cr) spins. Previously, we reported that Co and interfacial Cr spins couple antiferromagnetically; the spin orientation of Co and interfacial Cr is opposite [28]. Under the positive H_{MEFC} adopted in this paper, upward Co spin and downward Cr spin are the favorable spin alignments near the interface. When E during the MEFC is weak, this effect predominantly determines the interfacial spin alignment and yields the negative exchange-biased state. Conversely, the energy gain by the ME effect expressed by Equation (1) favors the upward Cr spin orientation favoring the positive exchange-biased state, which competes J_{INT} . The phenomenological expression of this energy competition is given by [27]

$$\Delta G = \left[(\alpha_{33} E + M_{AFM}) H_{MEFC} - \frac{J_{INT}}{t_{AFM}} \right] V_{AFM} \quad (2)$$

where J_{INT} denotes the interfacial exchange coupling energy (J/m^2), t_{AFM} is the AFM layer thickness (m), M_{AFM} is the uncompensated AFM moment in the Cr_2O_3 layer (Wb/m^2) and V_{AFM} is the activation volume (m^3). In the MEFC process, the spin configuration and the consequent exchange-biased state are determined during the cooling. Hence, the

thermal fluctuation is also taken into account. Assuming that the fraction of the negative and positive exchange-biased state obeys the Boltzmann distribution, the change in the exchange bias should be expressed as [27]

$$\mu_0 H_{\text{ex}}(E) \propto \tanh\left(\frac{\Delta G}{k_B T}\right) \quad (3)$$

where k_B is a Boltzmann constant (J/K) and T is an absolute temperature (K). In Figure 4b, the fitted results using Equation (3) are shown as the solid lines. The results for all adopted MEFC conditions follow the same mathematical form. When H_{MEFC} decreases, the curve shifts toward the high E . This shift can be also understood based on Equation (2). For the weaker H_{MEFC} , the higher E is required to compensate $J_{\text{INT}}/t_{\text{AFM}}$. According to this argument, the required E to switch the exchange bias should be inversely proportional to H_{MEFC} . Defining the threshold E , E_{th} to switch the exchange bias as E at which $\mu_0 H_{\text{ex}}$ becomes zero, we plot the E_{th} as a function of $1/\mu_0 H_{\text{MEFC}}$ in Figure 4c. The E_{th} linearly increases with $1/H$, which is in agreement with the above argument that the driving force for switching derives from the energy difference shown in Equation (1). The slope of Figure 4c gives the required EH product, which has been used as a measure of the ME-induced switching [15,27,29], as $1.24 \times 10^{14} \text{ V}\cdot\text{A}/\text{m}^2$.

In other multiferroic systems such as BiFeO_3 , the cross-correlation coefficient such as the piezoelectric coefficient (d_{33}) can be deteriorated with decreasing thickness because of the lattice confinement by the epitaxial strain [30]. Here, we discuss the deterioration/enhancement of α_{33} with reducing t_{AFM} by comparing the obtained value with the previous reports for the similar FM/ Cr_2O_3 stacked system [15,16,29,31]. To compare the EH product, J_{INT} in Equation (2) has to be taken into account. Although the direct evaluation of J_{INT} is difficult, the exchange anisotropy energy density J_K ($= H_{\text{ex}} \cdot M_S \cdot t_{\text{FM}}$) can be used as a measure of J_{INT} ; in the simple pinned spin model [32] or the weak J_{INT} limit in the domain wall model [33], J_{INT} and J_K become equal to each other. Because J_K depends on the temperature, J_K measured at 250 K, $J_{K,250}$ is used as a measure of J_{INT} as in the previous paper [16]. To evaluate J_K , the saturation magnetization per unit area, $M_S \cdot t_{\text{FM}}$ was measured based on the magnetization curve (M - H curve). The $M_{S,\text{FM}} \cdot t_{\text{FM}}$ value is $(5.7 \pm 0.7) \times 10^{-10} \text{ Wb}/\text{m}$, which is higher than the bulk Co because of the sizable spin polarization of Pt and Au attached with Co [21]. In Figure 4d, the relationship between $E_{\text{th}} \cdot H_{\text{MEFC}}/J_K$ and t_{AFM} is shown. As expected from Equation (2), $E_{\text{th}} \cdot H_{\text{MEFC}}/J_K$ roughly proportional to $1/t_{\text{AFM}}$. Notably, the $E_{\text{th}} \cdot H/J_K - 1/t_{\text{AFM}}$ relationship is maintained up to the 30-nm-thickness regime, which suggests that α_{33} does not deteriorate in this thickness regime. This finding is in agreement that the switching is not highly influenced by the strain effect discussed above.

We further evaluate the α_{33} value based on the isothermal switching mode. Figure 5a shows the AHE loops after applying the ME fields, $\mu_0 H_{\text{isothermal}} = 6 \text{ T}$ and $E = +98 \text{ MV}/\text{m}$ (red curve) and $\mu_0 H_{\text{isothermal}} = 6 \text{ T}$ and $E = -41 \text{ MV}/\text{m}$ (blue curve), which correspond to the positive and the negative exchange-biased states, respectively. We find that the switching between two states reversibly occurs. Figure 5b shows the change in $\mu_0 H_{\text{ex}}$ as a function of E . The clear hysteresis is observed, which indicates the presence of the energy barrier to switch the exchange bias polarity. We also find that the hysteresis shifts along the E -axis toward the positive direction, resembling the exchange bias in the M - H curve.

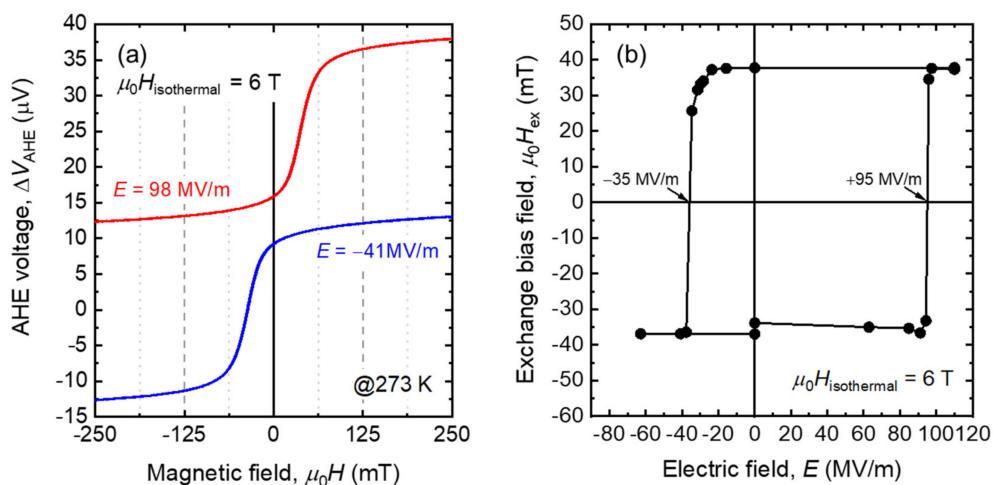


Figure 5. (a) AHE loops as a function of $\mu_0 H$ after the applying the magnetoelectric (ME) fields in an isothermal way. Blue and red curves represent the AHE loop after applying $E = +98 \text{ MV/m}$ and -41 MV/m with $\mu_0 H_{\text{isothermal}} = 6 \text{ T}$, respectively. (b) Change in the exchange bias field as a function of E with $\mu_0 H_{\text{isothermal}} = 6 \text{ T}$. In (b), the threshold electric field is shown by text.

We analyze the shift of the hysteresis based on the phenomenological expression of the switching energy assuming the coherent rotation of the AFM spin. The switching ME condition is expressed as [20]

$$(\alpha_{33} E_{\text{th}} + M_{\text{AFM}})H = -2K_{\text{AFM}} \mp \frac{J_{\text{INT}}}{t_{\text{AFM}}} \quad (4)$$

where K_{AFM} denote the magnetic anisotropy energy density of the AFM layer (J/m^3). In this expression, the change in the sign of α_{33} was taken into account. The sign of the second term depends on the switching direction: the negative-to-positive switching and the positive-to-negative switching. According to Equation (4), the asymmetry of the coercive E in the $\mu_0 H_{\text{ex}}-E$ hysteresis is relevant to the difference in the switching energy caused by the unidirectional nature of J_{INT} . Hence, the shift of the $\mu_0 H_{\text{ex}}-E$ hysteresis is essentially same as J_K . J_K is quantified as

$$J_K = \frac{M_{\text{S_FM}} \cdot t_{\text{FM}} \cdot \Delta H_C}{2} = \frac{\alpha_{33} H \cdot t_{\text{AFM}} \cdot \Delta E_{\text{th}}}{2} \quad (5)$$

The first expression is the difference in the Zeeman energy for up-to-down and down-to-up switching of the FM magnetization. Values of $M_{\text{S_FM}} \cdot t_{\text{FM}}$, $(5.7 \pm 0.7) \times 10^{-10} \text{ Wb/m}$, and the exchange bias field $H_{\text{ex}} = \Delta H_C / 2$ obtained from the AHE loop yields J_K at 273 K of $0.017 \pm 0.002 \text{ mJ/m}^2$. Because the exchange bias polarity is determined by the interfacial AFM spin direction [12], the $\mu_0 H_{\text{ex}}-E$ hysteresis represents that of the interfacial AFM spin as a function of E . The second expression relies on this fact. Using J_K , $\mu_0 H_{\text{isothermal}}$ (6 T), t_{AFM} (31.6 nm, determined by XRR, see above) and ΔE_{th} ($95 - 35 = 60 \text{ MV/m}$), α_{33} is yielded as $3.7 \pm 0.5 \text{ ps/m}$. This value is in good agreement with the reported values at the same temperature for the bulk Cr_2O_3 and 500-nm-thick Cr_2O_3 film [6], $\sim 4 \text{ ps/m}$.

One may imagine that the magnetic anisotropy energy of Cr_2O_3 can be evaluated from the K_{AFM} value. In our experiments, the exchange bias was checked at zero E , i.e., after removing the ME field. Besides, the exchange bias polarity is determined by the interfacial AFM spin direction [12]. Considering them, the change in the exchange bias reflects the interfacial antiferromagnetic spin direction at the remanent state after applying the ME field. Hence, the physical meaning of the $\mu_0 H_{\text{ex}}-E$ curve is similar to the remanent magnetization curve for the interfacial antiferromagnetic spin. The remanent magnetization curve is often used in the field of magnetic recording/storage and the details of the remanent magnetization curve can be found in [34]. Equation (4) assumes the

coherent rotation whereas the switching occurs with the AFM domain wall motion [12]. In this case, the K_{AFM} evaluated using Figure 5b and Equation (4) corresponds the ME energy equivalent to the remanent coercive E , which is typically two or three orders lower than the magnetocrystalline anisotropy energy density. This is similar to the fact that the switching H (the coercivity) in the M - H loop is different from the magnetic anisotropy field, $2K_{\text{FM}}/M_{\text{S_FM}}$ of the FM layer (K_{FM} is a magnetic anisotropy energy density of FM layer).

Finally, we discuss the role of M_{AFM} and the approach to decrease the switching energy using M_{AFM} . As shown in Figure 4d, the reduction in t_{AFM} enhances the switching energy, and this tendency is common to the isothermal mode. According to Equations (2) and (4), if M_{AFM} is parallel to $\alpha_{33}H$, it would assist the switching. M_{AFM} can be attributed to the finite magnetization at the bulk site [35,36] and/or the interfacial uncompensated moment [20]. The former relies on the defect-induced magnetization [35] and/or the selective substitution of the non-magnetic element such as Al to one sub-lattice [36]. Although the reduction effect using M_{AFM} has been actually reported for the MEFC mode [28], the validity for the isothermal mode has not been proven, which will be investigated in the near future.

4. Conclusions

In summary, we investigate the ME-induced switching of the exchange bias, using especially the Cr_2O_3 thin film with the 30-nm thickness regime. While the previous reports seem to pay less attention to the isothermal reversible switching because of the difficulty, this switching mode as well as the MEFC-driven switching are demonstrated. Based on the argument on the switching energy, the non-deterioration of the ME susceptibility α_{33} in the studied thickness regime is found. The quantitative analysis based on the asymmetric switching energy in the isothermal switching gives the similar α_{33} value to the reported one for the bulk Cr_2O_3 . While the cross-correlation functionality can deteriorate with decreasing thickness in some multiferroic materials, the finding in this paper could be a light for the further reduction in the ME-based all-thin-film system, which may be useful in spintronic applications.

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