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**Deposition of germanium dioxide films by the injection
of oxygen ion beam in conjunction with
hexamethyldigermane**

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18 **HIGHLIGHTS**

- 19 ● Germanium dioxide films have attracted much attention to its practical applications.
- 20 ● A methodology for fabricating germanium dioxide films was presented.
- 21 ● We used hexamethyldigermane (HMDG) as a source material.
- 22 ● Germanium dioxide films were formed when O^+ ions were irradiated in conjunction
- 23 with HMDG.

24

25 **Keywords:**

26

27 ion beam induced deposition;

28 germanium dioxide;

29 oxygen ion beam;

30 hexamethyldigermane

ABSTRACT

We proposed a methodology for fabricating a film by the simultaneous injections of O^+ ions and hexamethyldigermane (HMDG) to a substrate. The O^+ -ion energy was set at 50 eV. After the experimental trial, we found a film deposited on the substrate. The analyses of the film with X-ray photoelectron spectroscopy and Fourier transform infrared spectroscopy showed that the deposited film was germanium dioxide (GeO_2). It was also noted that no film deposition occurred when HMDG was supplied to substrates without O^+ -ion beam injections. In conclusion, the low-energy O^+ -ion beam induced deposition using HMDG was found to be useful for the deposition of GeO_2 films.

1. Introduction

The ion-beam-induced chemical vapor deposition (IBICVD) method is useful for fabricating nanometer-scale 3D structures [1] and ferromagnetic materials [2]. It can also be used to prepare films of metal oxides such as TiO_2 [3, 4], Al_2O_3 [4], ZrO_2 [5], and SiO_2 [6-9]. The IBICVD method enables films to be deposited at the exact location where both the source material and ion beam contact the substrate.

Germanium oxide films have attracted much attention because of their suitability for use in practical applications such as optical waveguides [10], semiconductors [11], passivation layers [12], rutile substrates [13], and optical memory devices [14]. Germanium oxide films can be produced via various methods such as radio-frequency sputtering [15-18], vacuum evaporation [19], atomic-layer deposition [20], thermal oxidation [21], anodic oxidation [22], ozone oxidation [23], and plasma oxidation [24].

Although IBICVD is a powerful tool for producing metal oxide films [3-9], the literature contains no reports concerning its use for the fabrication of germanium oxide films. In most cases, germanium oxide films formed by the aforementioned methods [15-24] were

deposited uniformly over the substrates. By contrast, films deposited by the IBICVD method can, in principle, be deposited at the exact location where both the source material and ion beam contact the substrate.

For the production of germanium oxide films by IBICVD, a suitable source material must first be selected. Hexamethyldigermane [HMDG, $(\text{CH}_3)_3\text{GeGe}(\text{CH}_3)_3$] is a molecular substance that contains two Ge atoms per molecule. We therefore selected HMDG as a source material for IBICVD. In addition, we selected an O^+ -ion beam. In the present study, O^+ -ion beams with various energies were used to irradiate a substrate, in conjunction with a flow of HMDG. After the experiments, we observed films deposited on the substrates. The deposited films were subsequently analyzed by X-ray photoelectron spectroscopy (XPS), X-ray diffraction (XRD), X-ray reflectometry (XRR), and Fourier-transform infrared (FTIR) spectroscopy.

2. Materials and Method

The film formation trials were carried out using an ion-beam injection system. A schematic view of the system is provided elsewhere [25]. In the present study, pure carbon

dioxide (CO₂) gas was used for O⁺-ion production. Specifically, O⁺ ions were produced by the decomposition of CO₂ gas in a Freeman-type ion source of the beam system and were then extracted. After mass-selection, the O⁺ ions were guided to the processing chamber of the beam system, where they irradiated a substrate at normal incidence.

A schematic of the process chamber is shown in Fig. 1. The base pressure in the chamber was 1×10^{-6} Pa. Liquid HMDG (Sigma-Aldrich Co. LLC) in a stainless steel container was used as a source material. The HMDG vapor was extracted from the container using Ar as a carrier gas. The mixed gas (HMDG + Ar) was supplied to the substrate surface at 0.7 sccm, in conjunction with the incident O⁺-ion beam. The pressure was 1×10^{-3} Pa during the trial.

An Au-coated quartz crystal microbalance (QCM) substrate (ULVAC, CRTS-0) or an untreated Si substrate (15 × 15 mm) was used as a substrate for O⁺-ion beam irradiation. The diameter of the film deposition area on the QCM substrate was 7.5 mm. Both substrates were used at room temperature. The change in the mass of the film deposited on the QCM substrate was measured using a QCM controller (ULVAC, CRTM-9000). After the experimental trials, the deposited films were analyzed by XPS (ULVAC-PHI,

ESCA-3057), stylus profilometry (KLA-Tencor, P-15), XRD (RIGAKU, RINT2200),
XRR (RIGAKU, Smart Lab), and FTIR spectroscopy (Jasco, FT/IR-410).

3. Results and discussion

Prior to the film formation experiments, the mass of the incident ions was measured using
a mass and energy analyzer (barzers, PPM-421), and the results are shown in Fig. 2. The
mass of the incident ions was 16 u (Fig. 2). Therefore, the incident ions were O^+ ions
without impurity ions. The results of ion-energy measurements using the PPM-421
analyzer show that the peak energy in the ion energy distribution was 50 eV (Fig. 3). The
profile of the ion beam was also acquired (Fig. 4).

During experiments to form germanium oxide films via O^+ -ion beam irradiation of Si
substrates in the presence of HMDG, we could not determine whether a film was actually
formed on the substrate surface unless we removed the substrate from the process
chamber and analyzed its surface using an instrument such as a stylus profiler. Therefore,
in the first trial, we irradiated a QCM substrate with an O^+ -ion beam in the presence of
HMDG. We could then recognize that a film was formed without removing the substrate

from the process chamber. The mass of the deposited film was measured *in situ* using the CRTM-9000 controller.

Before the IBICVD trial, we irradiated the QCM substrate with an O^+ -ion beam without supplying HMDG, and found that no film was deposited. Trials in which HMDG was supplied to the substrate without O^+ -ion irradiation also showed no film deposition.

The QCM substrate was subsequently irradiated with O^+ -ion beams of various energy while HMDG was supplied to the substrate. We performed five trials in which the peak O^+ energy was 20, 50, 75, 100, or 200 eV, corresponding to current densities of 0.5, 1.3, 1.6, 1.4, and 1.1 $\mu A/cm^2$, respectively. After each trial, a film was found to have been deposited onto the substrate. The masses of the deposited films were measured using the CRTM-9000 controller, and the results are shown in Fig. 5. We found that the film mass in the 50 eV case was greater than those in the 20, 75, 100, and 200 eV cases. Therefore, subsequent film formation experiments were carried out using a 50 eV O^+ -ion beam.

We irradiated a QCM substrate with a 50 eV O^+ -ion beam while supplying HMDG to the substrate. The duration of the ion-beam injection was 800 min. After the trial, we analyzed the deposited film using the CRTM-9000 controller, revealing the film mass to be 5 μg .

133

134 The deposited film was further analyzed by XPS using an ESCA-3057 X-ray
135 photoelectron spectrometer equipped with an Al K_{α} radiation source. In the present study,
136 the operating parameters for the ESCA-3057 spectrometer were an energy step of 0.2 eV
137 and a dwell time of 20 ms; the number of scans per spectrum was 10. Energy calibration
138 of the ESCA-3057 spectrometer was regularly performed using standard reference
139 specimens. The acquired XPS spectra are shown in Fig. 6. The binding energies in the
140 $Ge3d$ [Fig. 6(a)] and $O1s$ [Fig. 6(b)] spectra indicate that the deposited film was
141 germanium oxide. The O/Ge atomic ratio calculated from the XPS data was 2 (i.e., GeO_2).
142 The $C1s$ XPS spectrum [Fig. 6(c)] of the film shows that no carbon atoms were included
143 in the film.

144

145 We next attempted to obtain an FTIR spectrum of the deposited film. However, the film
146 deposited on the QCM substrate was not suitable for FTIR analysis because infrared
147 radiation did not penetrate the substrate. Therefore, to obtain a film suitable for FTIR
148 analysis, a 50 eV O^+ -ion beam was used to irradiate a Si substrate in the presence of
149 HMDG. The experimental parameters were the same as those in the experiments using
150 the QCM substrates. The duration of ion-beam irradiation was 2950 min. After the
151 IBICVD experiment, we found that a film had formed on the Si substrate.

152

153 We first measured the thickness of the film using a stylus profiler. The measurement
154 showed that the film was ~ 100 nm thick.

155

156 The FTIR spectrum of the film (Fig. 7) was obtained using an FT/IR-410 spectrometer in
157 transmission mode with subtraction of the contribution of the Si substrate. The bands at
158 ~ 2350 cm^{-1} in the FTIR spectrum are attributed to CO_2 gas remaining inside the
159 measurement chamber of the spectrometer. The strong peak observed at ~ 900 cm^{-1} is
160 known to correspond to GeO_2 [26].

161

162 The deposited film on the Si substrate was analyzed by XRR. XRR measurements of the
163 film showed that the thickness, density, and roughness of the film were 101 nm, 3.9 g/cm^3 ,
164 and 11 nm, respectively.

165

166 The deposited film on the Si substrate was subsequently analyzed by XRD using an
167 RINT2200 X-ray diffractometer. The wavelength of the X-rays was $\lambda = 1.78892$ Å (Co
168 $\text{K}_{\alpha 1}$), and the X-ray beam was incident to the film surface at an angle θ . The obtained
169 XRD pattern (θ - 2θ method) shows no obvious peaks (Fig. 8), suggesting that no
170 crystalline structures were present in the film.

171

172 Matsutani et al. pioneered the IBICVD method using HMDG as a source material [27].

173 In their experiments, a Si substrate was irradiated with an Ar^+ -ion beam while HMDG

was simultaneously supplied to the substrate. They reported the deposition of amorphous germanium carbide (GeC) films. No experiments have been reported concerning the simultaneous use of an O^+ -ion beam and HMDG, as conducted in the present study. Irrespective of the variety of available experimental methods, no previous experiments using HMDG as a source material for GeO_2 film deposition have been reported.

Yoshimura et al. attempted to use HMDG as a source material in a low-energy molecular-ion-beam deposition method and proposed a method for fabricating amorphous GeC films by injecting $GeCH_x$ molecular ions produced by the decomposition of HMDG into substrates [28]. They reported that amorphous GeC films were formed only when the $GeCH_x$ ion energy was less than 30 eV. They also found that, when $GeCH_x$ ions were injected into substrates with an energy of 30 eV or more, the bonds between Ge and C in the $GeCH_x$ ions were broken and Ge and C failed to recombine on the substrate. By contrast, we found that, when O^+ -ion beams were used to irradiate a substrate while HMDG was simultaneously sprayed onto the substrate, the bonds between Ge and C in the HMDG molecules were broken and then Ge atoms combined with O atoms to form GeO_2 films on the substrate.

Sigma-Aldrich Co. LLC produces a very small quantity of HMDG and sells it only for research purposes. For this reason, only a few groups have reported studies related to HMDG, such as electron spin resonance measurements [29], identification of fragment ions [30, 31], and chemical applications [32]. HMDG has attracted little attention; thus, little interest has been expressed in its industrial applications. There have been no attempts thus far to apply HMDG in industrial fields. HMDG is an expensive reagent

(~\$100 USD per gram). Because our proposed method uses HMDG as a source material, the current methodology is more expensive than other reported germanium oxide film-forming methods [15-24]. However, we emphasize that, if applications of HMDG (e.g., the GeO₂ deposition method proposed in the present study) become widespread in industrial and technological fields, chemical companies such as Sigma-Aldrich Co. LLC might produce and sell HMDG in larger quantities, drastically reducing its cost.

4. Conclusion

An experimental methodology for GeO₂ film formations using HMDG as a source material was presented. HMDG was supplied onto the substrate surface together with the O⁺-ion beam. The O⁺-ion energy was 50 eV. The substrate was set at room temperature. We found a film formed on the substrate after the trial. The XPS and FTIR results showed that the film was GeO₂. In addition, the film mass obtained at five different O⁺ energy levels (20, 50, 75, 100, and 200 eV) was evaluated. The mass of the deposited film obtained following 50 eV injection was larger than those obtained following 20, 75, 100, and 200 eV injections. In conclusion, the 50 eV O⁺-ion beam induced deposition using HMDG was found to be useful for the formation of the GeO₂ film.

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218

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222

223 **Declaration**

224 **Author Contributions**

225 Satoru Yoshimura: Conceived and designed the experiments; Performed the experiments;
226 Wrote the paper.

227 Satoshi Sugimoto: Performed the experiments; Contributed reagents, materials, analysis
228 tools or data.

229 Takae Takeuchi: Analyzed and interpreted the data; Contributed reagents, materials,
230 analysis tools or data.

231 Kensuke Murai: Performed the experiments; Contributed reagents, materials, analysis
232 tools or data.

233 Masato Kiuchi: Analyzed and interpreted the data; Contributed reagents, materials,
234 analysis tools or data.

235

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Data Availability Statement

Data will be made available on request.

Declaration of interest's statement

The authors declare no conflict of interest.

Additional information

No additional information is available for this paper.

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Figure Captions

345

346 **Fig 1.** Schematic drawing of the process chamber of the ion beam system.

347

348 **Fig 2.** The mass spectrum of the ion beam.

349

350 **Fig 3.** The energy spectrum of the O^+ -ion beam.

351

352 **Fig 4.** Typical intensity profile for the O^+ -ion beam. Horizontal axis represents the
353 distance in the vertical direction.

354

355 **Fig 5.** The dependence of deposited film mass on the O^+ -ion energy.

356

357 **Fig 6.** (a) Ge3d, (b) O1s, and (c) C1s X-ray photoelectron spectroscopy spectra of a film
358 deposited following the injection of O^+ ions to a quartz crystal microbalance substrate in
359 conjunction with hexamethyldigermane.

360

361 **Fig 7.** Fourier transform infrared spectrum of a film deposited on a Si substrate following
362 the injection of O^+ ions in conjunction with hexamethyldigermane.

363

364 **Fig 8.** X-ray diffraction pattern (θ - 2θ method) of a film deposited on a Si substrate
365 following the injection of O^+ ions in conjunction with hexamethyldigermane.

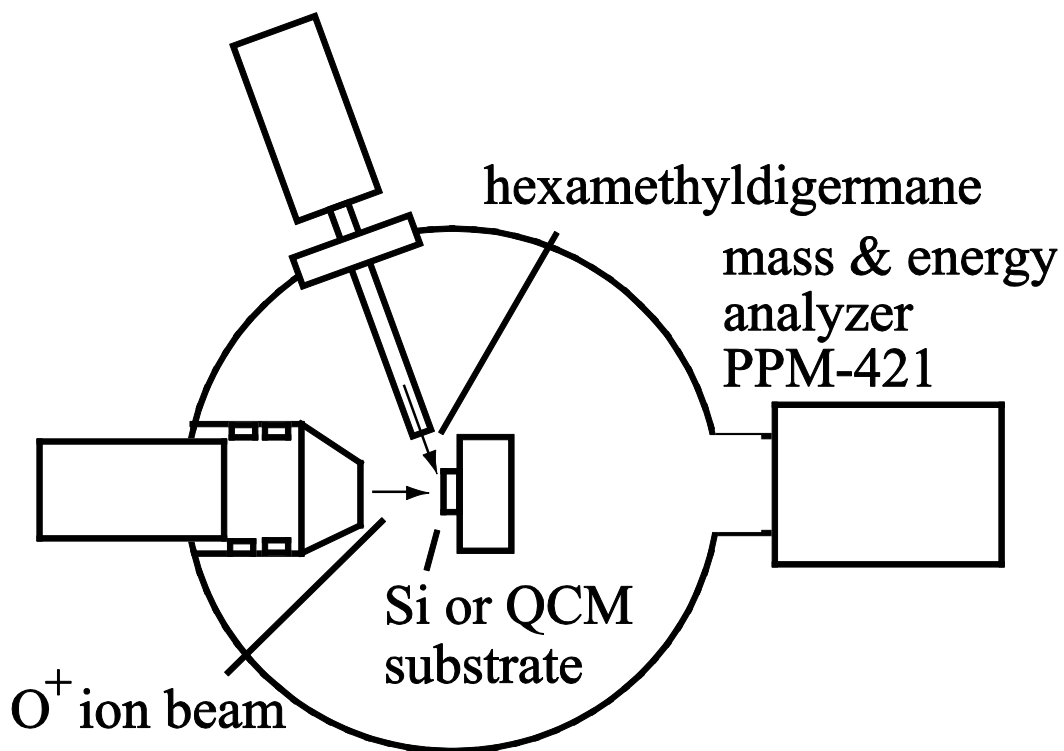


Fig. 1

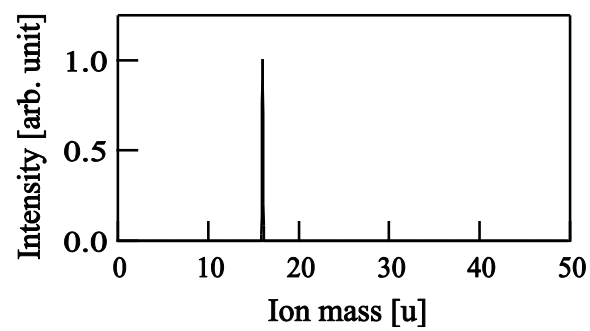


Fig. 2

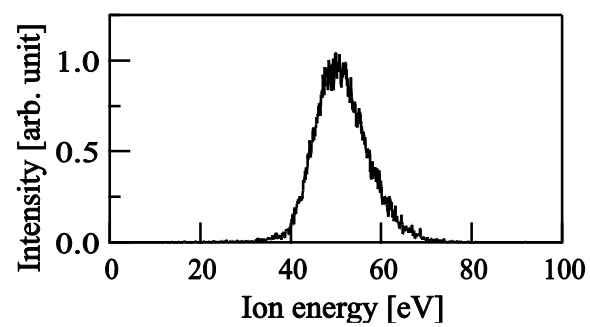


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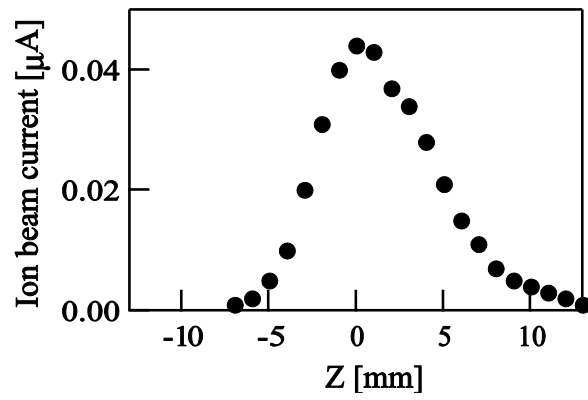


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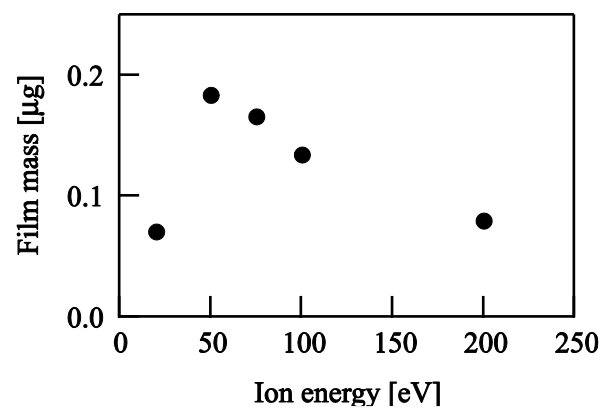


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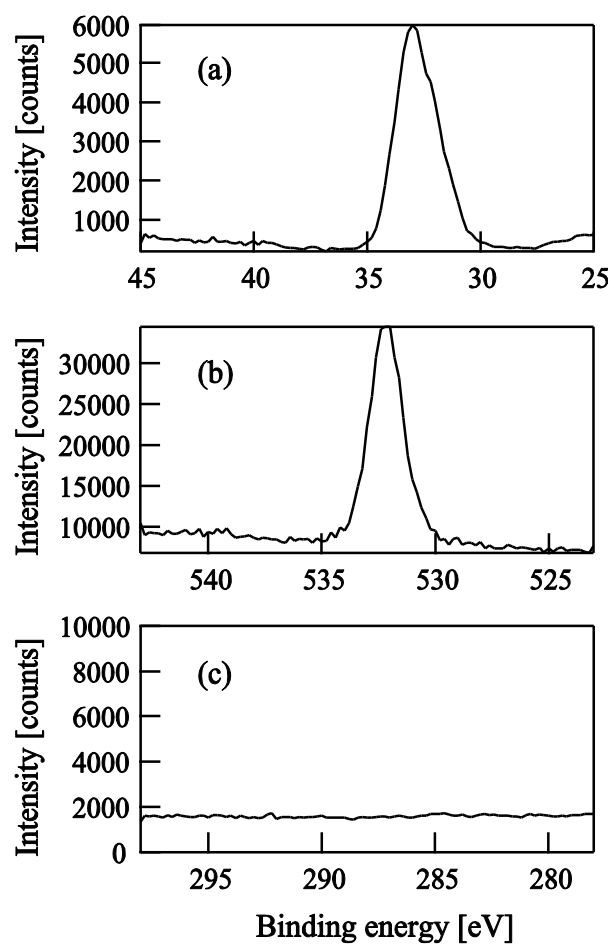


Fig. 6

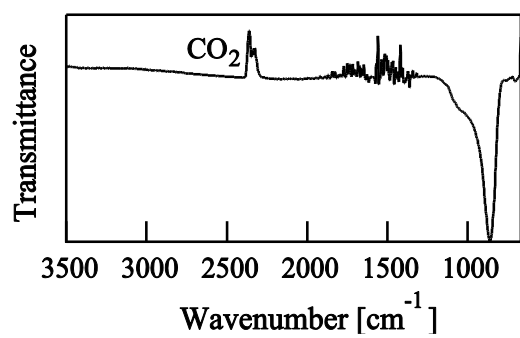


Fig. 7

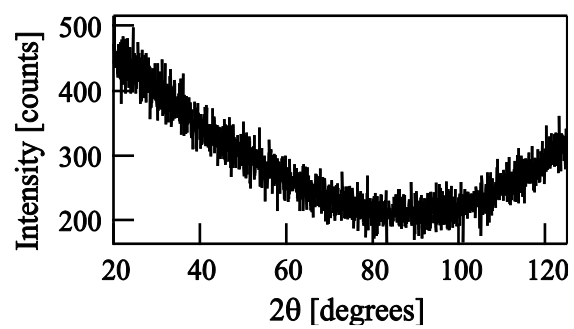


Fig. 8